



INSA | INSTITUT NATIONAL
DES SCIENCES
APPLIQUÉES
LYON



UMONS
Université de Mons

materials
UMONS RESEARCH INSTITUTE
FOR MATERIALS SCIENCE
AND ENGINEERING



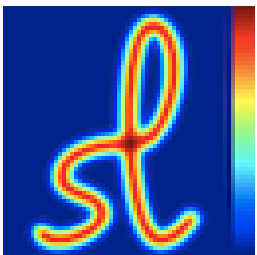
Mesures des propriétés viscoélastiques des matériaux : des mesures macroscopiques à la cartographie quantitative à la nanoéchelle

Dr. Florent DALMAS

florent.dalmas@insa-lyon.fr

Prof. Dr. Philippe LECLERE

philippe.leclere@umons.ac.be



Forum des Microscopies à Sonde Locale 2024
22 – 26 avril 2024, Ecully

Plan

La viscoélasticité, qu'est-ce que c'est ?

La spectroscopie mécanique ou DMA

L'équivalence temps-température

Limites et perspectives : vers les mesures locales

Extension à la nanoéchelle

- nanoDMA
- CR-AFM
- Intermodulation AFM
- ...

Conclusions et Perspectives



Plan

La viscoélasticité, qu'est-ce que c'est ?

La spectroscopie mécanique ou DMA

L'équivalence temps-température

Limites et perspectives : vers les mesures locales

Extension à la nanoéchelle

- nanoDMA
- CR-AFM
- Intermodulation AFM
- ...

Conclusions et Perspectives



La viscoélasticité

Elasticité



Viscosité



L'expérience de la goutte de poix

Fluide visqueux

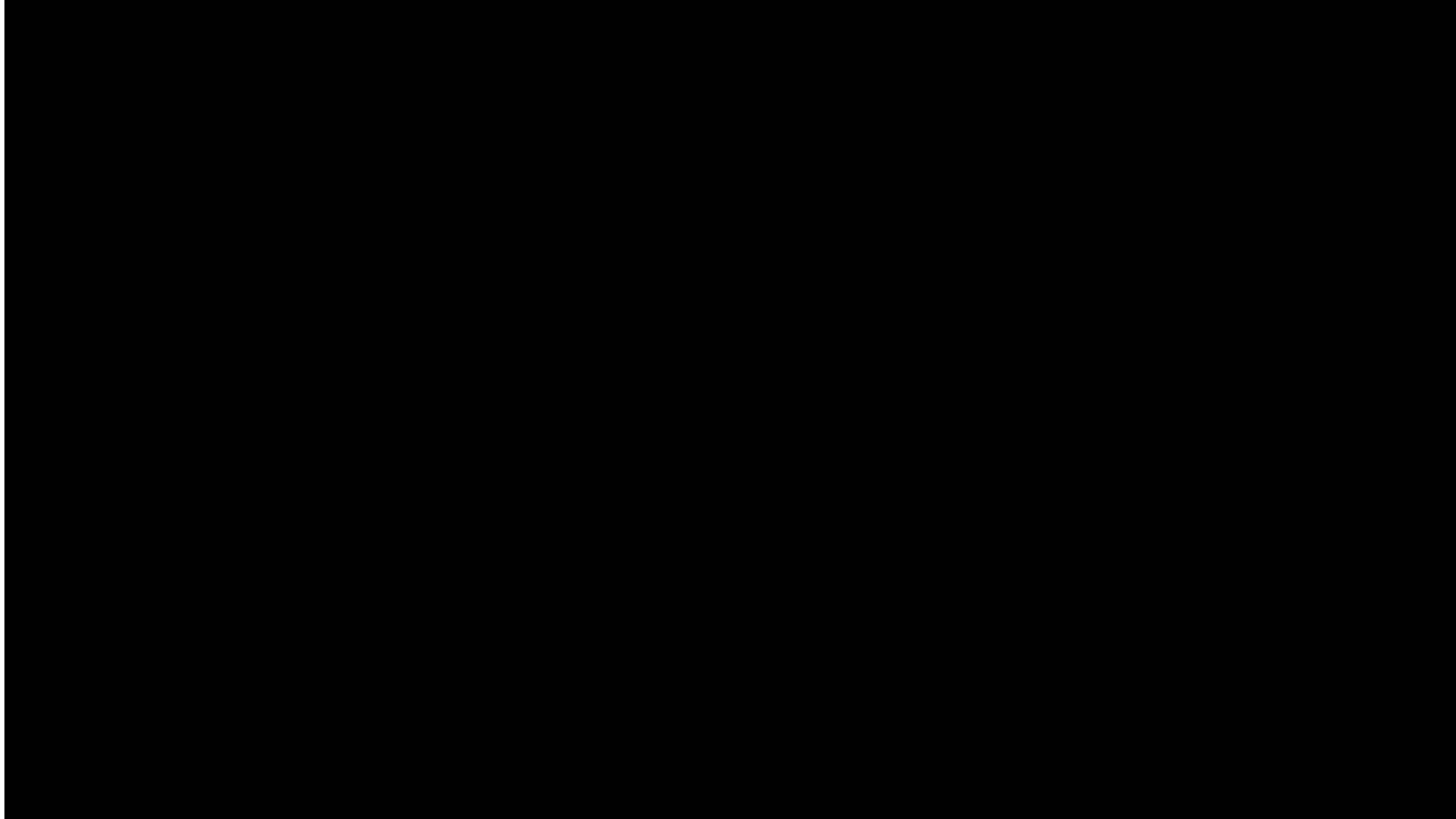


Solide fragile



La viscoélasticité

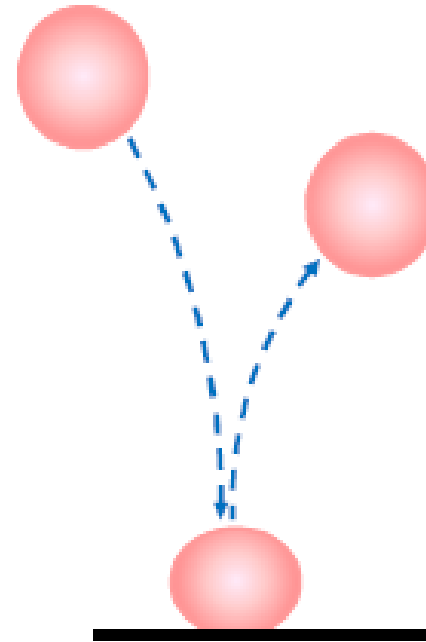
La maïzena dans l'eau



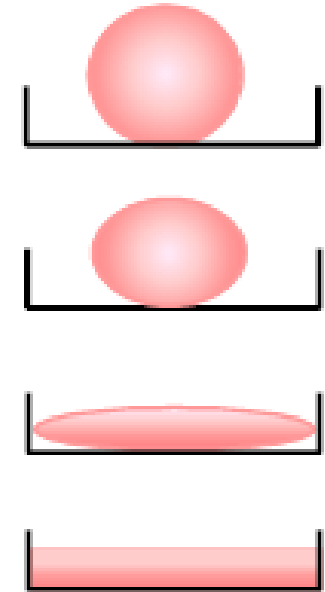
La viscoélasticité

Un matériau peut avoir l'apparence d'un « solide » si :

- 1- il a un **temps de relaxation** caractéristique très long
- 2- le processus de déformation correspondant est très rapide



T is short [$< 1s$]



T is long [24 hours]

La viscoélasticité

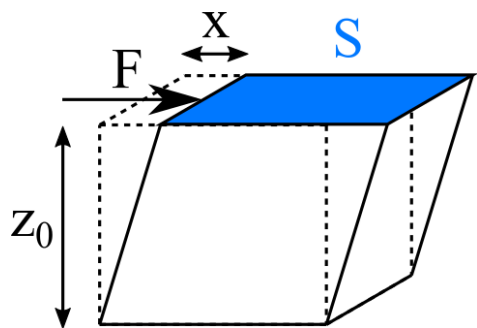


La viscoélasticité

Modes de sollicitation

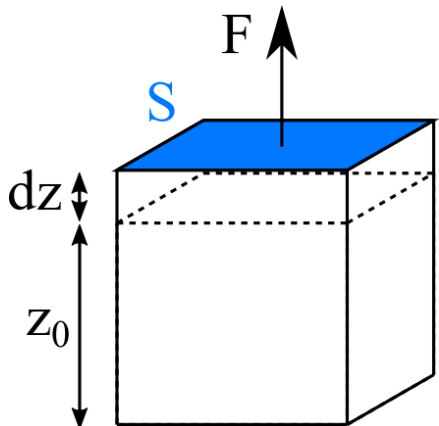
Mesurer les propriétés mécaniques

Contrainte σ , déformation ε



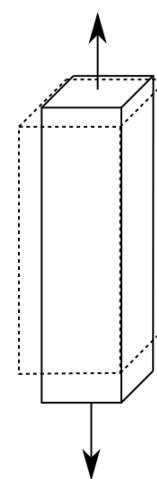
Cisaillement

$$\sigma = \frac{F}{S} \quad \gamma = \frac{x}{z_0} \quad \dot{\gamma} = \frac{1}{z_0} \frac{dx}{dt}$$

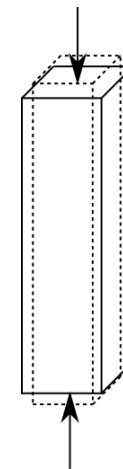


Traction

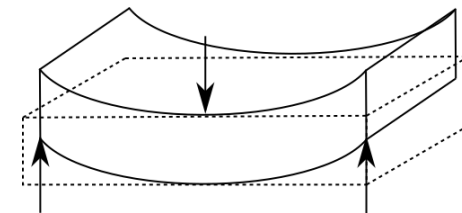
$$\sigma = \frac{F}{S} \quad d\varepsilon = \frac{dz}{z_0} \quad \dot{\varepsilon} = \frac{1}{z_0} \frac{dz}{dt}$$



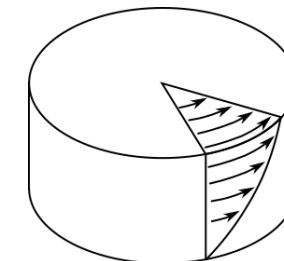
Traction



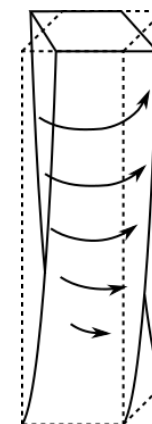
Compression



Flexion



Cisaillement

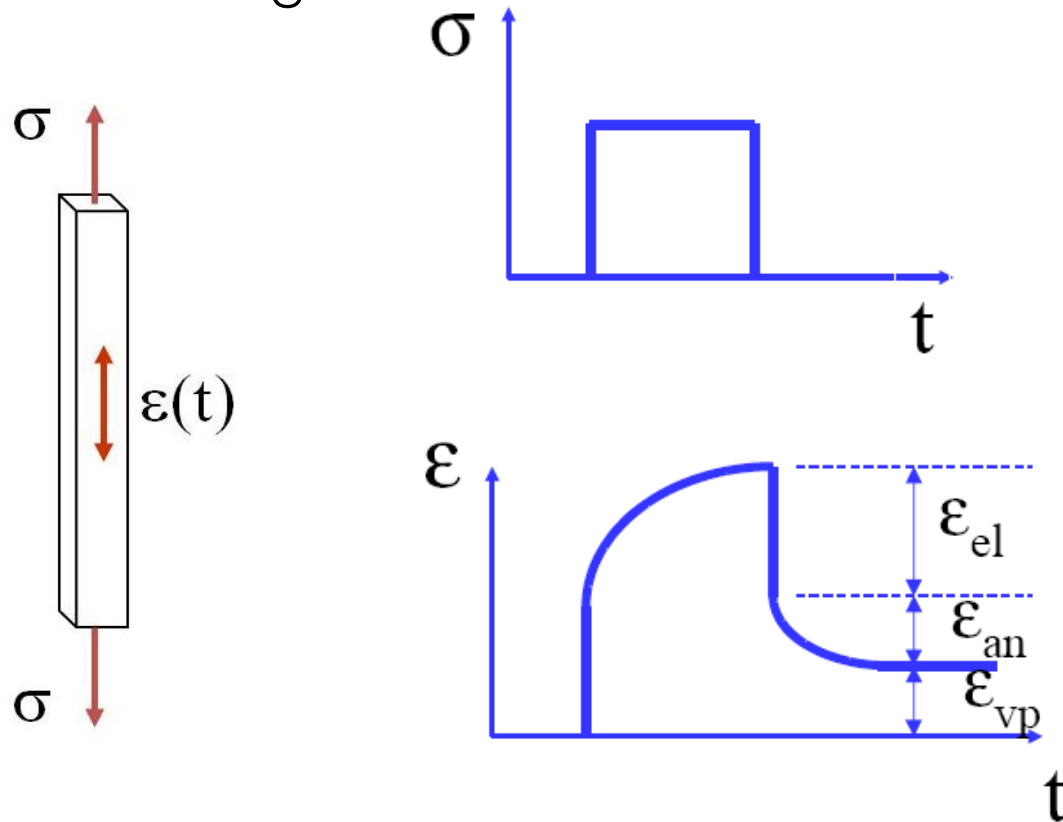


Torsion

La viscoélasticité

Manifestation expérimentale de la viscoélasticité

Essai de fluage



el : élastique; retour réversible, immédiat

an : anélastique ou viscoélastique; retour réversible, différé (f(temps))

vp : viscoplastique; irréversible, pas de retour (ε_{vp} peut être recouvrée à $T > T_g$)

La viscoélasticité

L'élasticité



Loi de Hook

En traction : $\sigma = E\varepsilon$

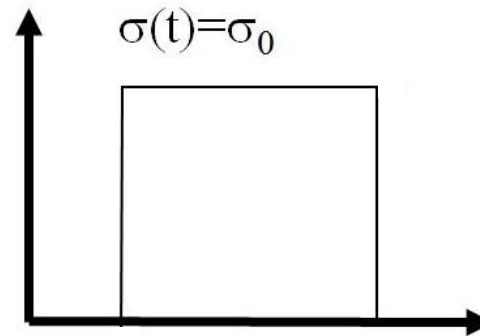
En cisaillement : $\sigma = G\gamma$

Module d'élasticité :

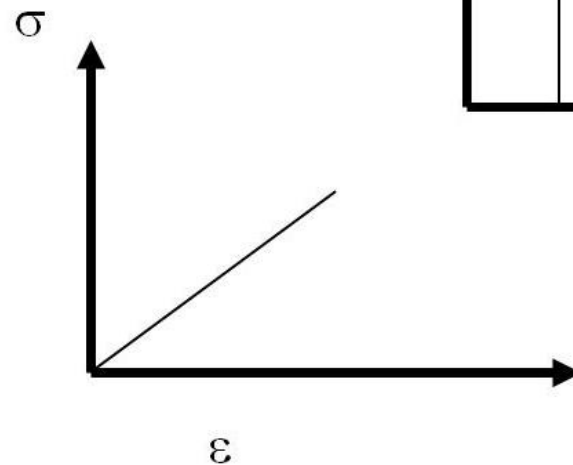
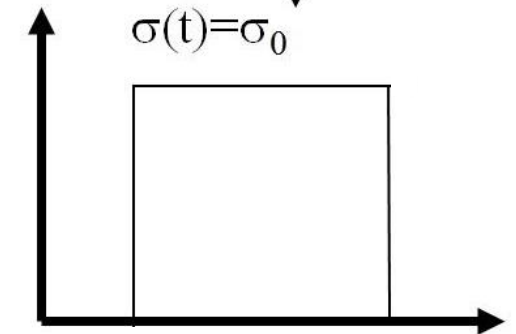
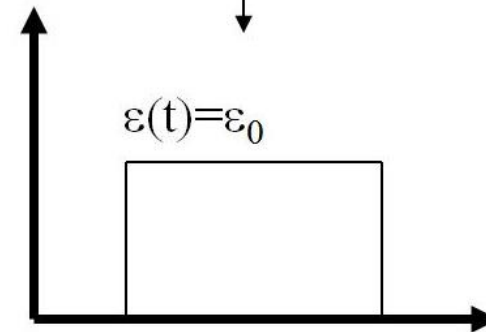
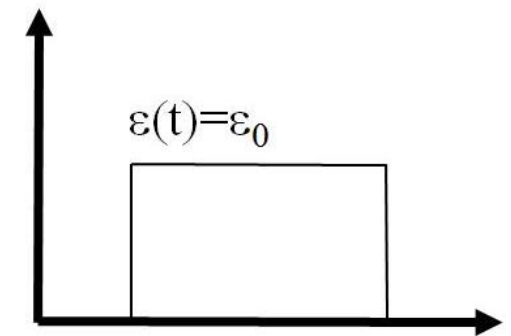
$$E = 2(1 + \nu)G$$



Essai de fluage



Essai de relaxation



Essai de traction monotone

La viscoélasticité

La viscosité



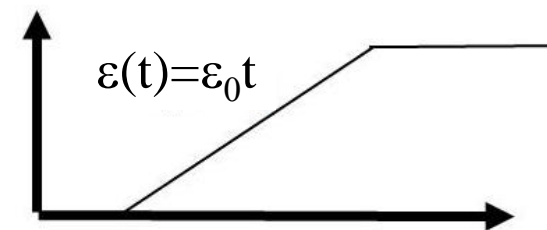
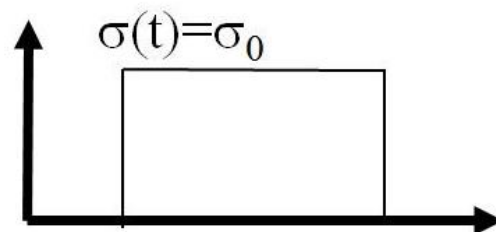
Loi de Newton

En traction : $\sigma = \eta \dot{\epsilon}$

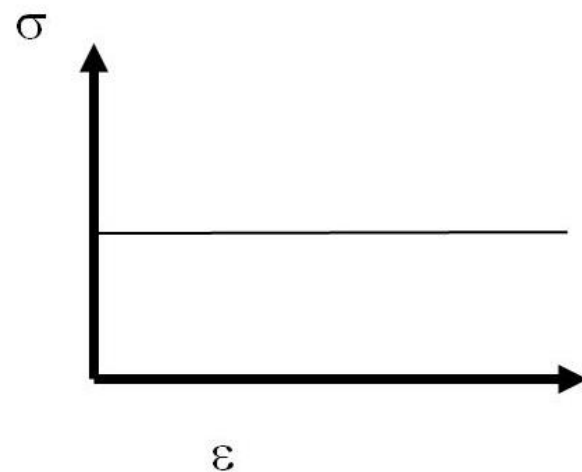
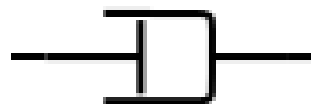
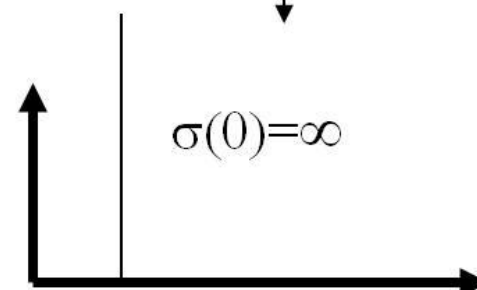
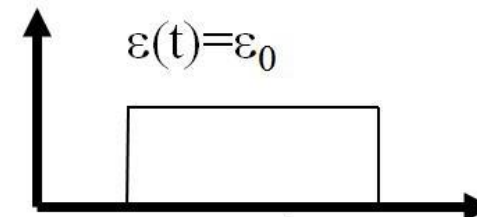
En cisaillement : $\sigma = \eta \dot{\gamma}$

η : Viscosité

Essai de fluage



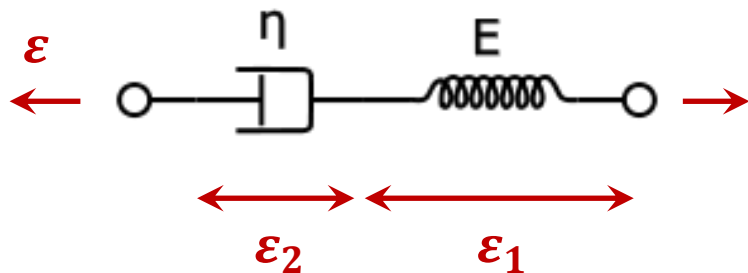
Essai de relaxation



Essai de traction monotone

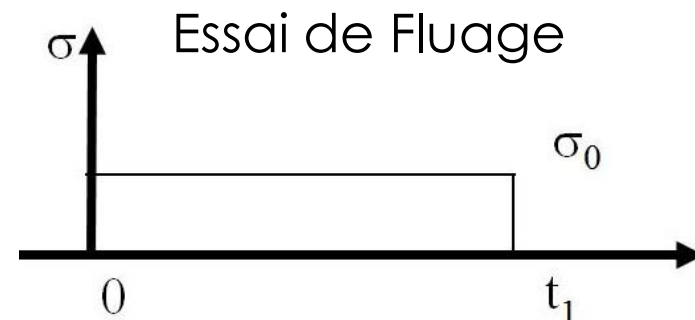
La viscoélasticité

Et la viscoélasticité alors ?



$$\begin{array}{l|l} \sigma = \sigma_1 = \sigma_2 & \sigma_1 = E \varepsilon_1 \\ \varepsilon = \varepsilon_1 + \varepsilon_2 & \sigma_2 = \eta \dot{\varepsilon}_2 \end{array}$$

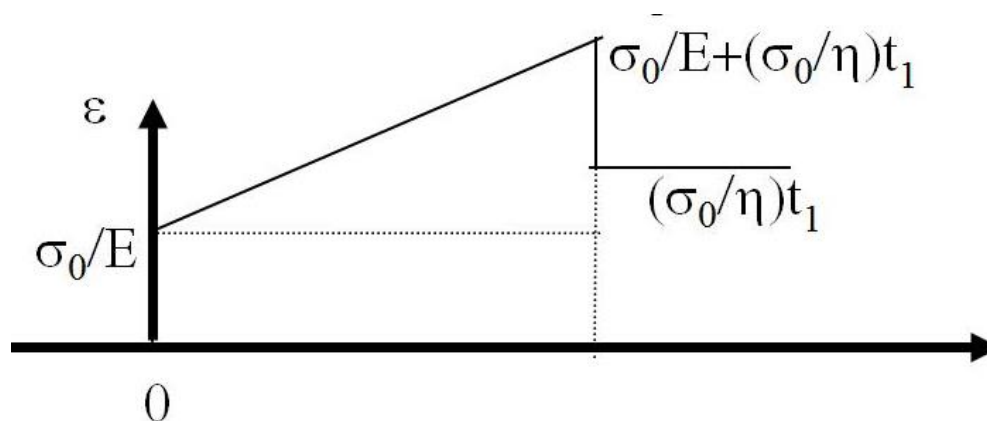
$$\dot{\varepsilon} = \frac{d\varepsilon}{dt} = \frac{1}{E} \frac{d\sigma}{dt} + \frac{1}{\eta} \sigma$$



À $t = 0, \sigma = \sigma_0$ et $\varepsilon = \frac{\sigma_0}{E}$

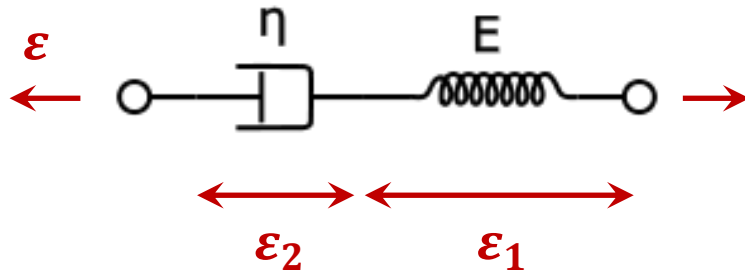
À $0 < t < t_1$ $\varepsilon(t) = \frac{\sigma_0}{\eta} t + \frac{\sigma_0}{E}$

À $t > t_1$ $\varepsilon_1 = 0$ $\varepsilon_2 = \frac{\sigma_0}{\eta} t_1$



La viscoélasticité

Et la viscoélasticité alors ?



$$\begin{array}{l|l} \sigma = \sigma_1 = \sigma_2 & \sigma_1 = E \varepsilon_1 \\ \varepsilon = \varepsilon_1 + \varepsilon_2 & \sigma_2 = \eta \dot{\varepsilon}_2 \end{array}$$

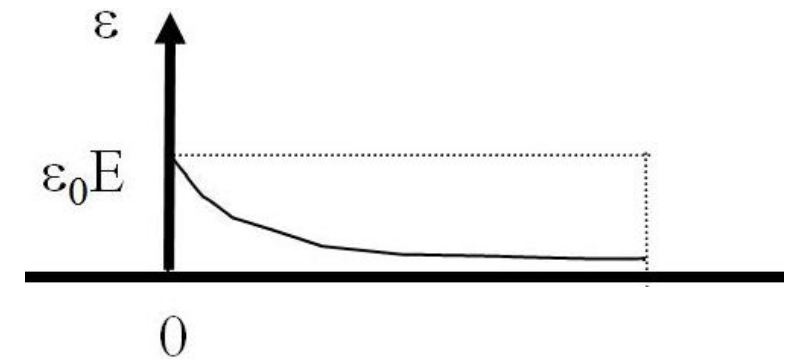
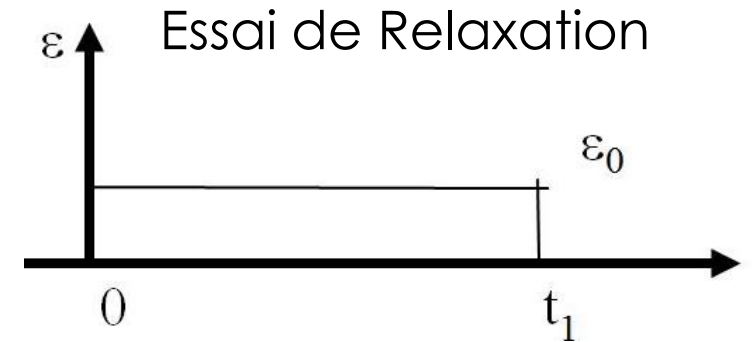
$$\dot{\varepsilon} = \frac{d\varepsilon}{dt} = \frac{1}{E} \frac{d\sigma}{dt} + \frac{1}{\eta} \sigma$$

Avec $\frac{d\varepsilon}{dt} = 0 \rightarrow \sigma(t) = A e^{-\frac{t}{\tau}}$

À $t = 0, \sigma = \varepsilon_0 E \rightarrow A = \varepsilon_0 E$

$$E_{eq}(t) = \frac{\sigma_0}{\varepsilon} = E e^{-\frac{t}{\tau}}$$

$\tau = \frac{\eta}{E}$: temps de relaxation



La viscoélasticité

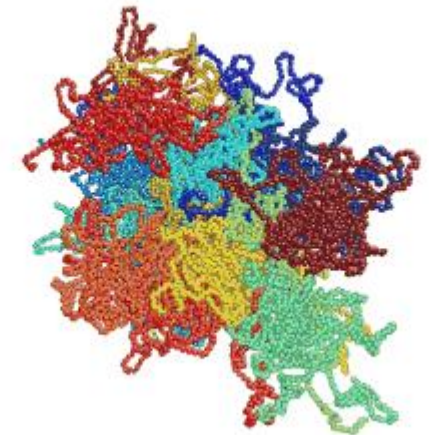
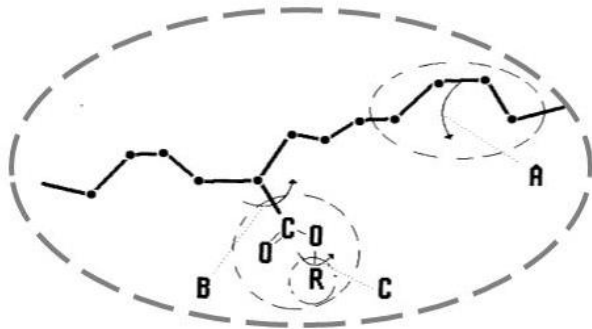
$$E(t) = E_0 e^{-\frac{t}{\tau}}$$

Et la viscoélasticité alors ?

Temps de relaxation → « mécanisme relaxationnel » ou relaxation

Lorsque le matériau est soumis à une sollicitation mécanique, les phénomènes intervenant au sein du matériau ont besoin d'un certain temps pour accommoder la contrainte.

Pour quoi du temps ?

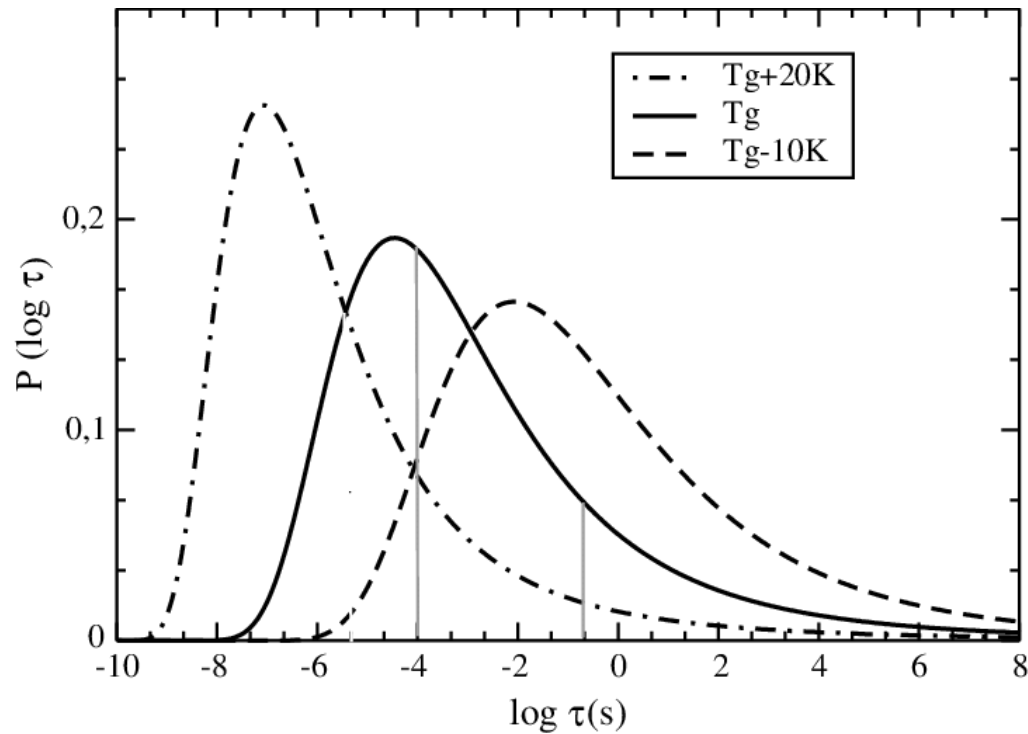


Le concept de relaxation est rencontré dès qu'un temps est nécessaire avant le retour à un équilibre.

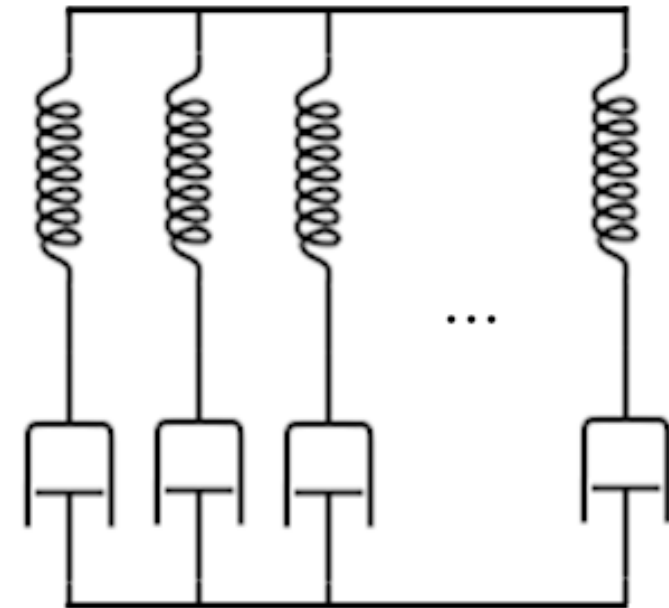
La viscoélasticité

Comment décrire ce plat de serpent viscoélastique ?

Spectre de relaxation d'un polystyrène



Modèle de Maxwell généralisé



La spectrométrie mécanique ou DMA

Comment mesurer les propriétés viscoélastiques ?



S/D Cantilever



Film/Fiber Tension



3-Point Bending



Compression



Analyse
Mécanique
Dynamique
(DMA en anglais)

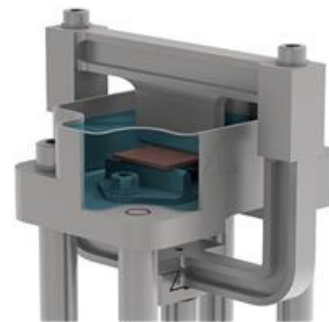
Shear Sandwich



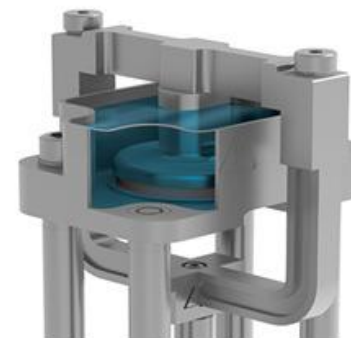
Submersible Tension



Submersible Bending



Submersible Compression



La spectrométrie mécanique ou DMA

- Déformation sinusoïdale imposée : $\varepsilon^* = \varepsilon_0 e^{i\omega t}$
- Contrainte déphasée : $\sigma^* = \sigma_0 e^{i(\omega t + \delta)}$

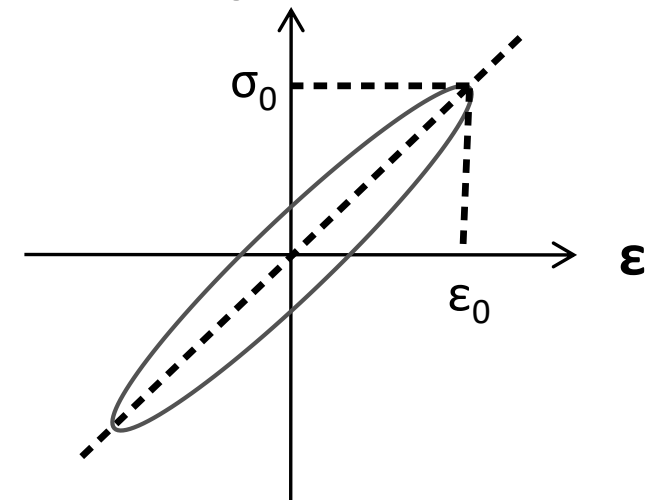
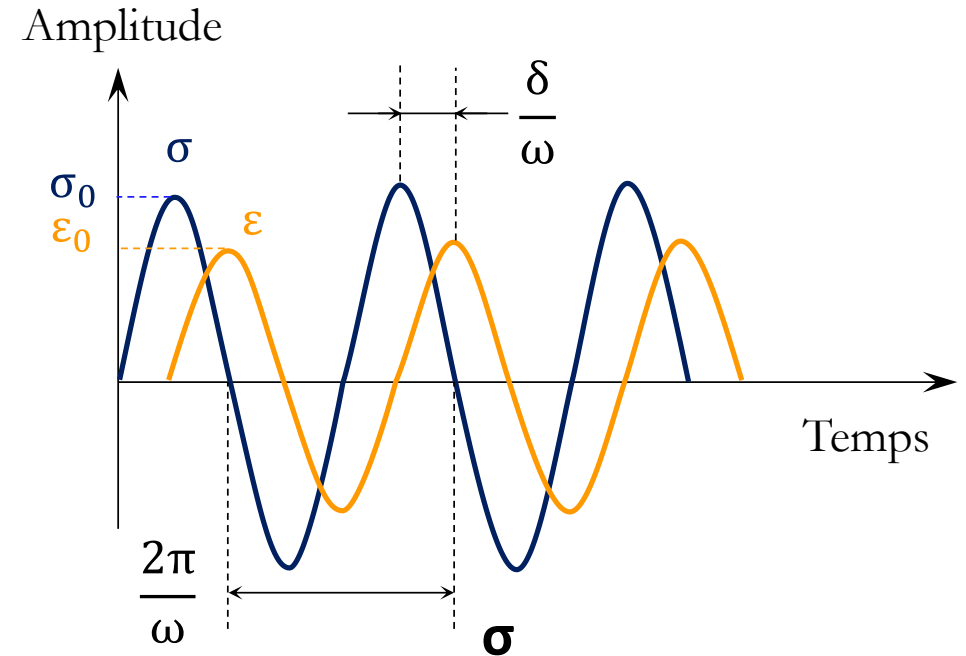
⇒ **Module d'Young complexe** : $E^* = \frac{\sigma^*}{\varepsilon^*} = E' + iE''$

{ E' = Module de conservation (élasticité)
 E'' = Module de perte (anélasticité)

⇒ **Facteur de perte** : $\tan(\delta) = \frac{E''}{E'}$

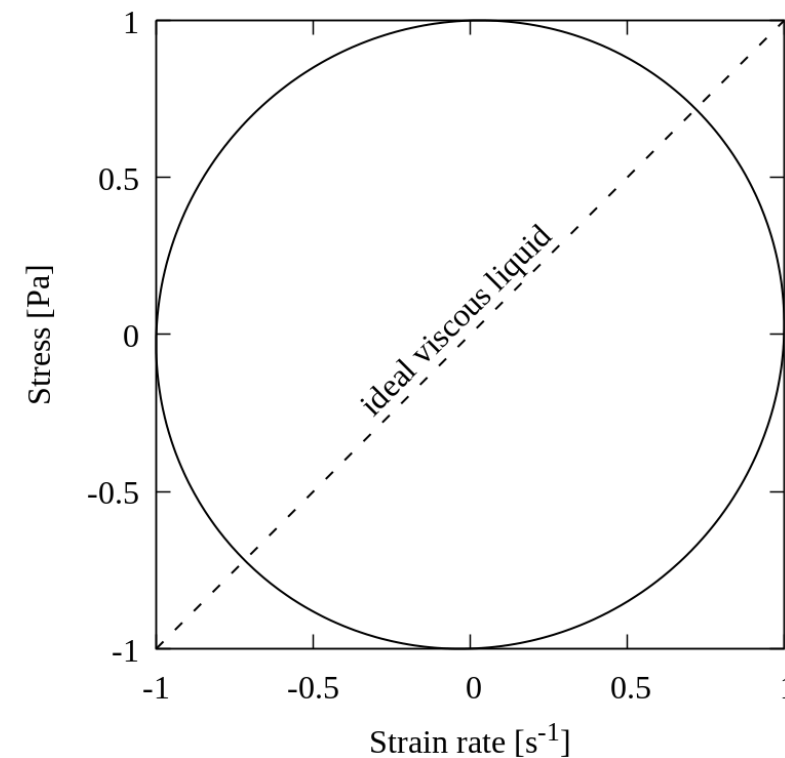
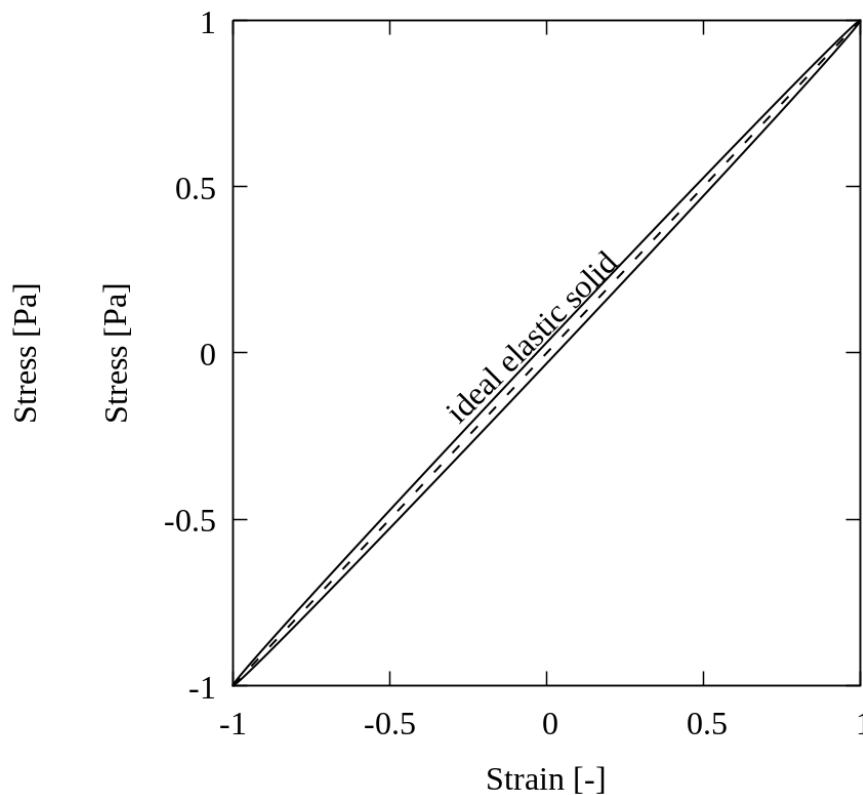
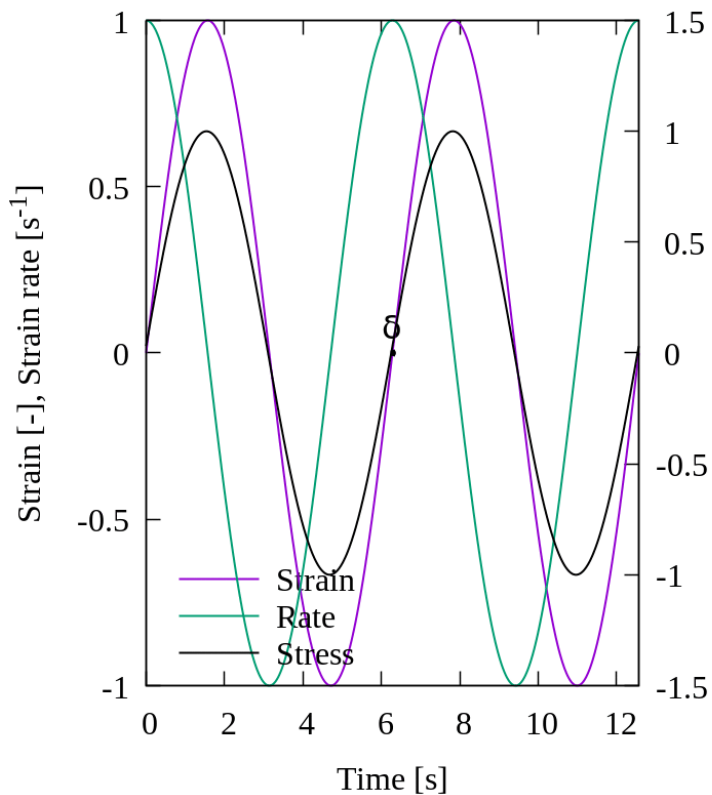
Module complexe en fonction de la **température** (isochrone) ou de la **fréquence** (isotherme) de sollicitation

$f : 10^{-2}\text{Hz} \rightarrow 10^2\text{Hz}$

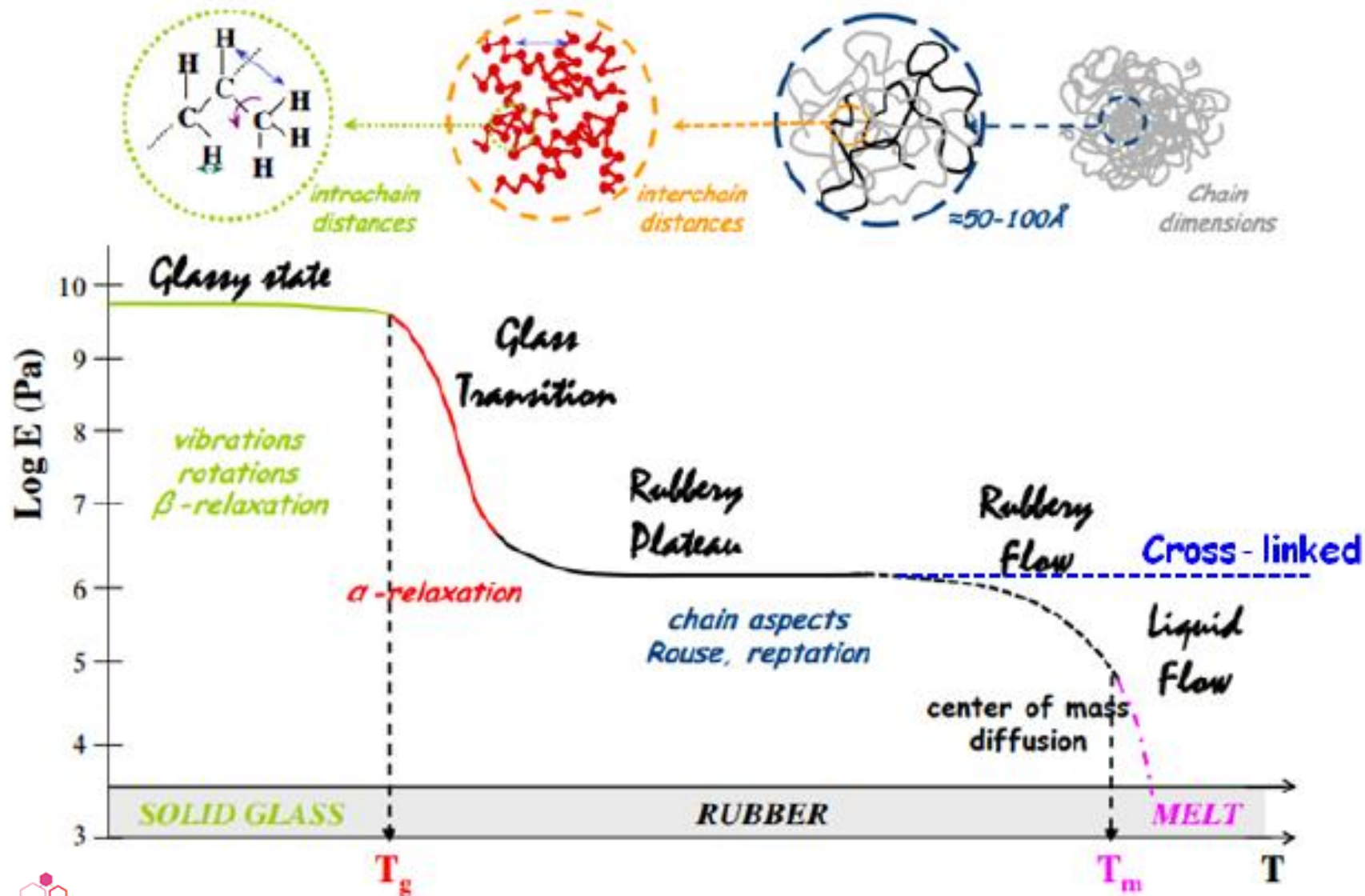


La spectrométrie mécanique ou DMA

$\tan \delta = G''/G' = 0.03$
more elastic



La spectrométrie mécanique ou DMA

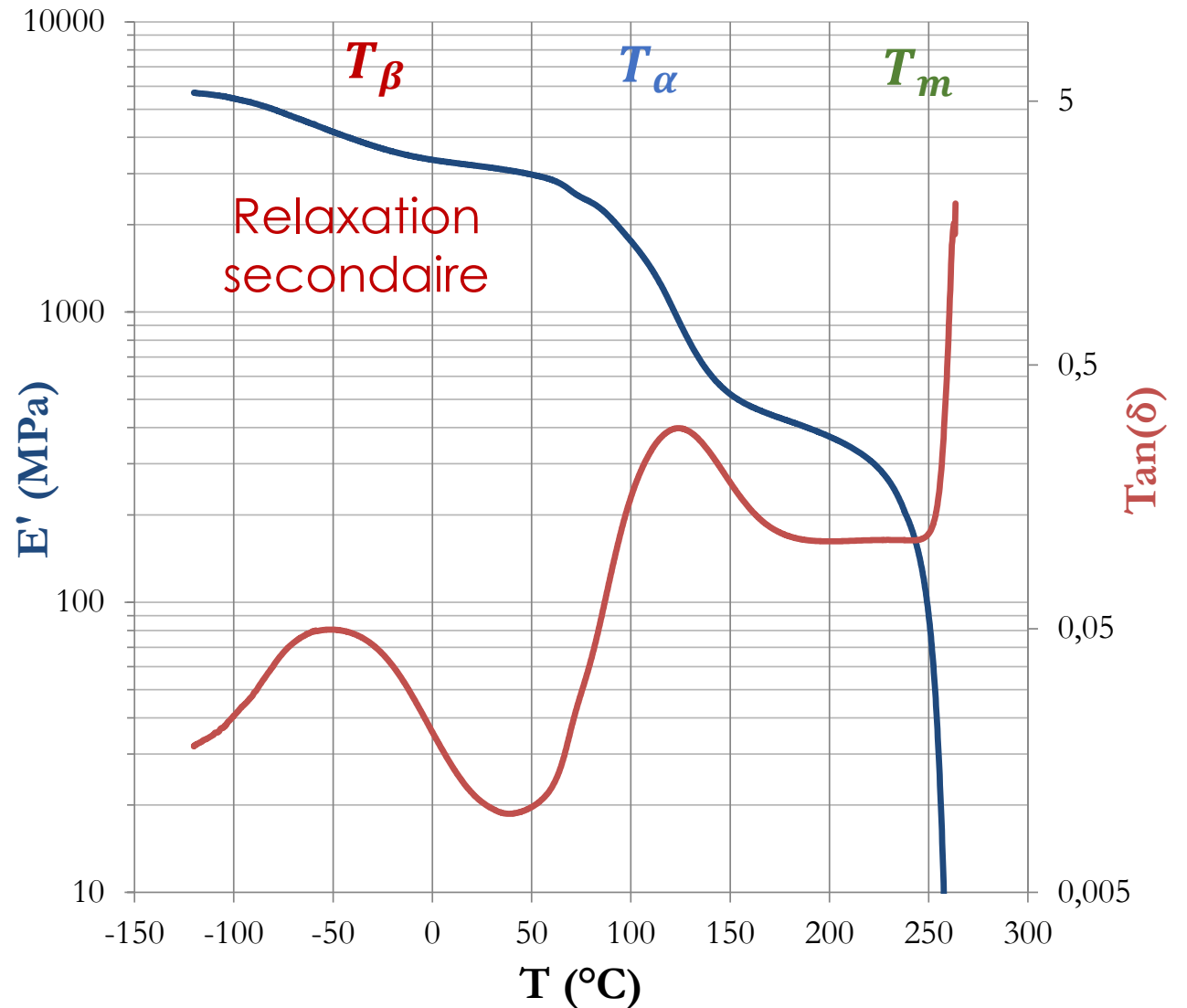
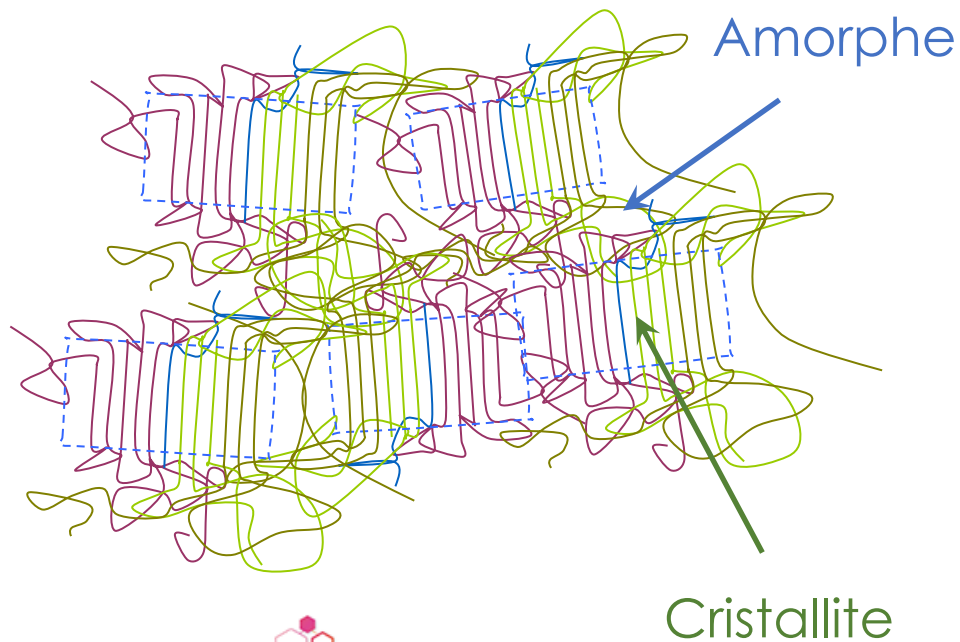
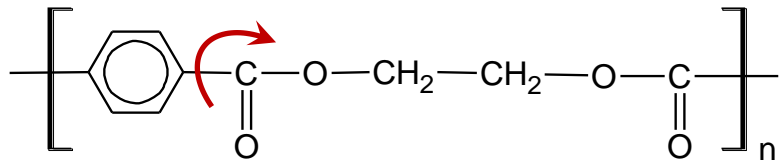


La spectrométrie mécanique ou DMA

Relaxation principale

Fusion

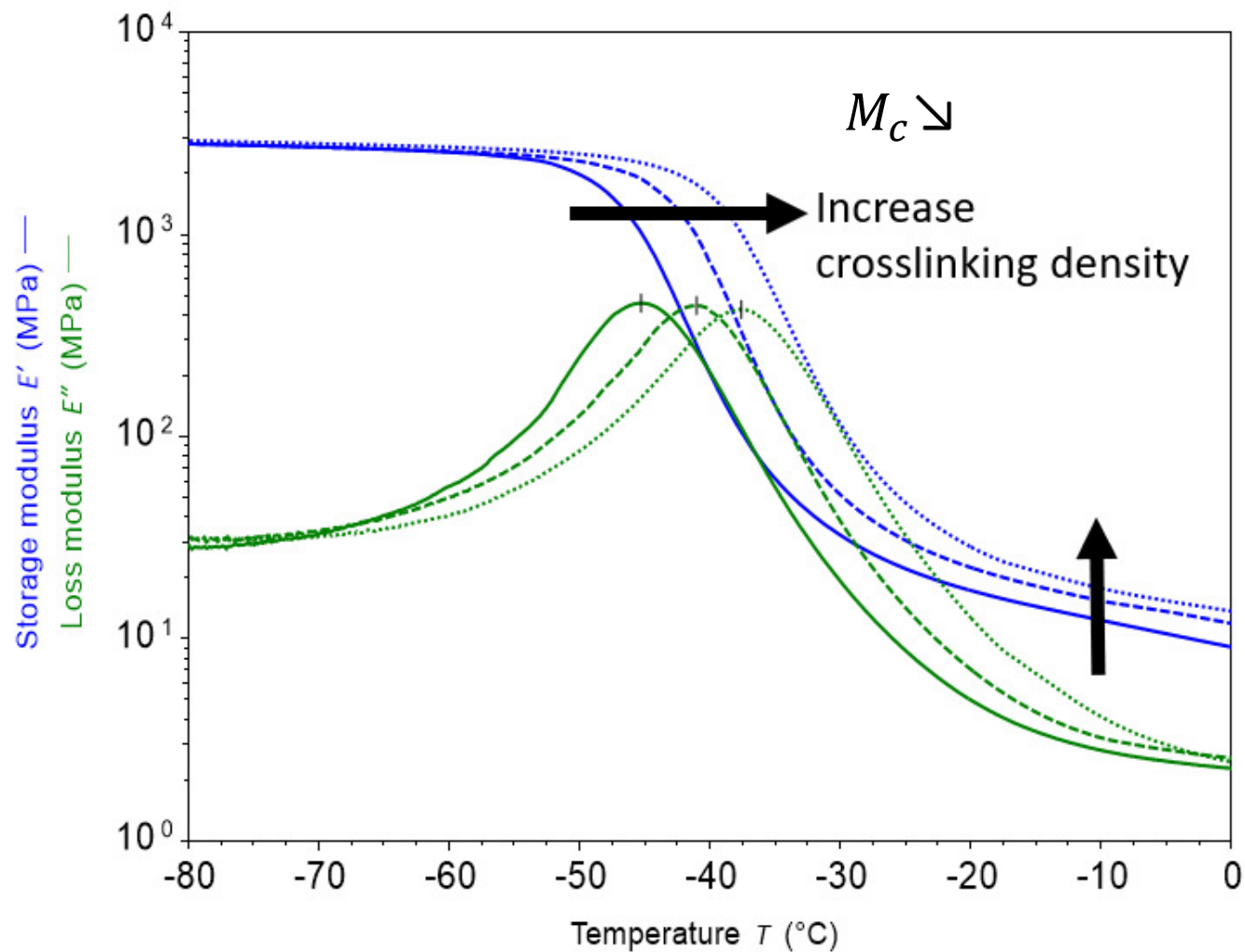
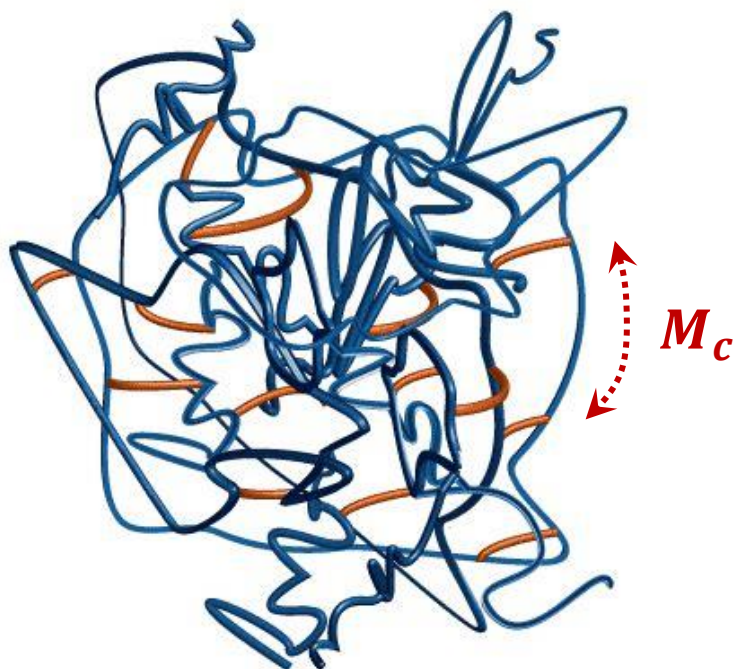
Relaxations dans le polyéthylène téréphthalate (PET)



La spectrométrie mécanique ou DMA

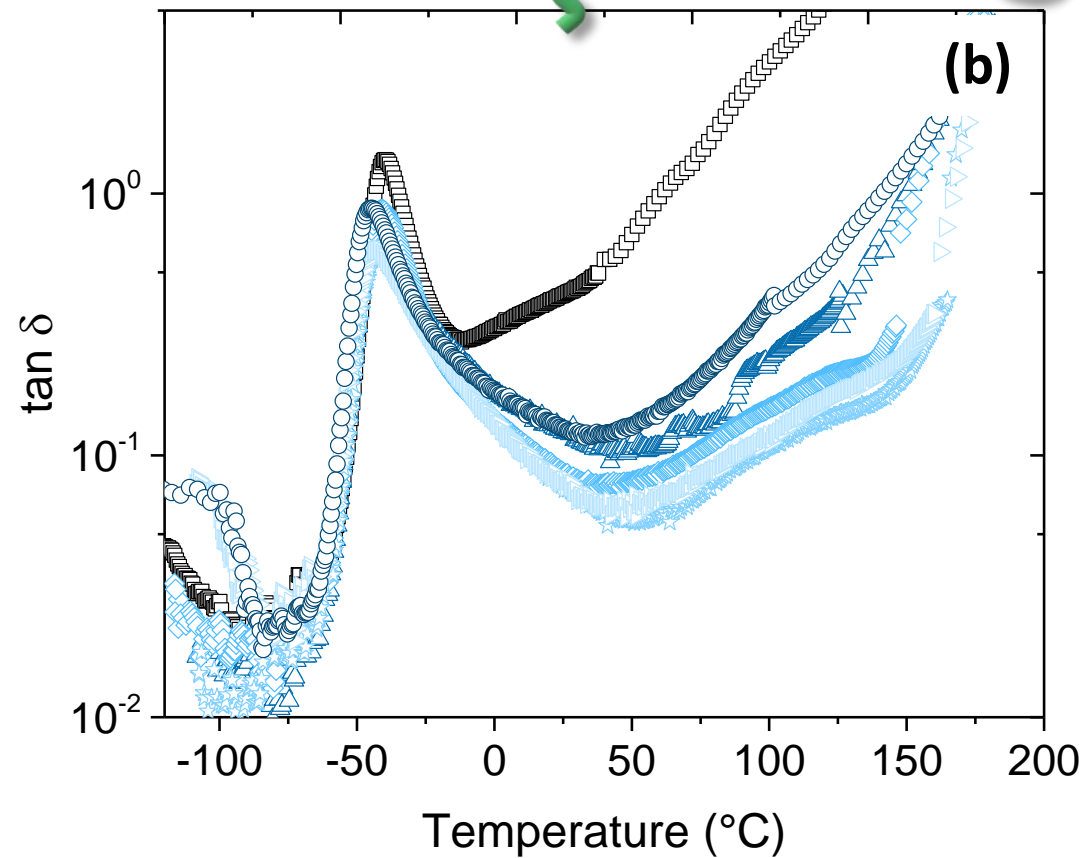
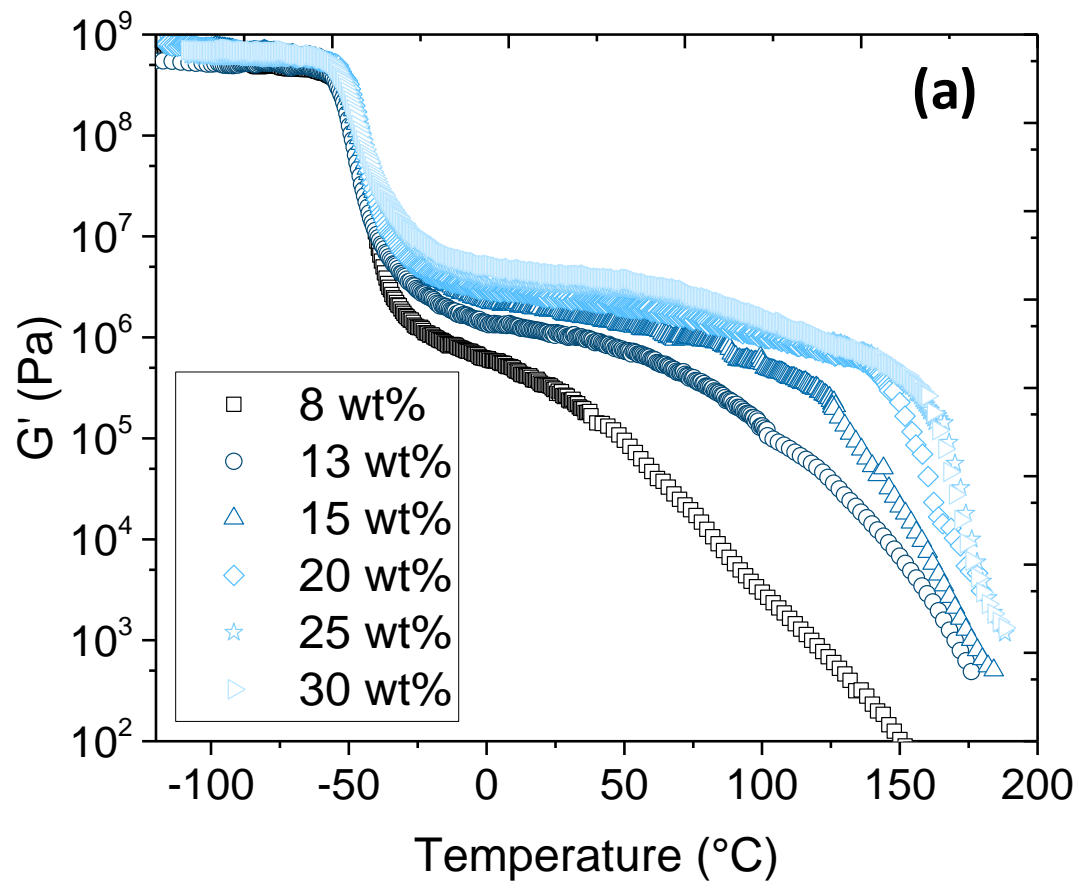
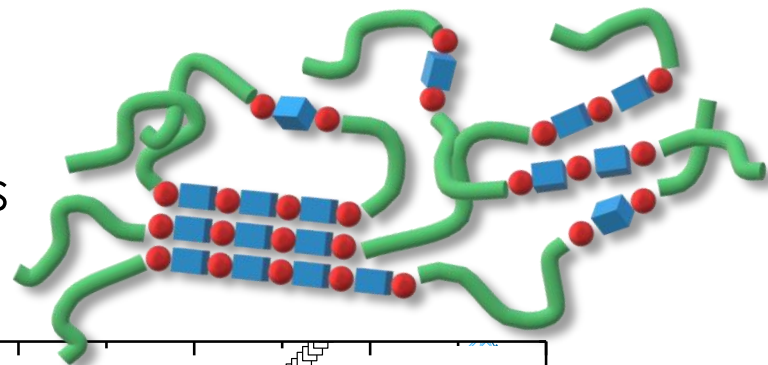
Quelques exemples

Elastomère réticulé : effet de la densité de réticulation

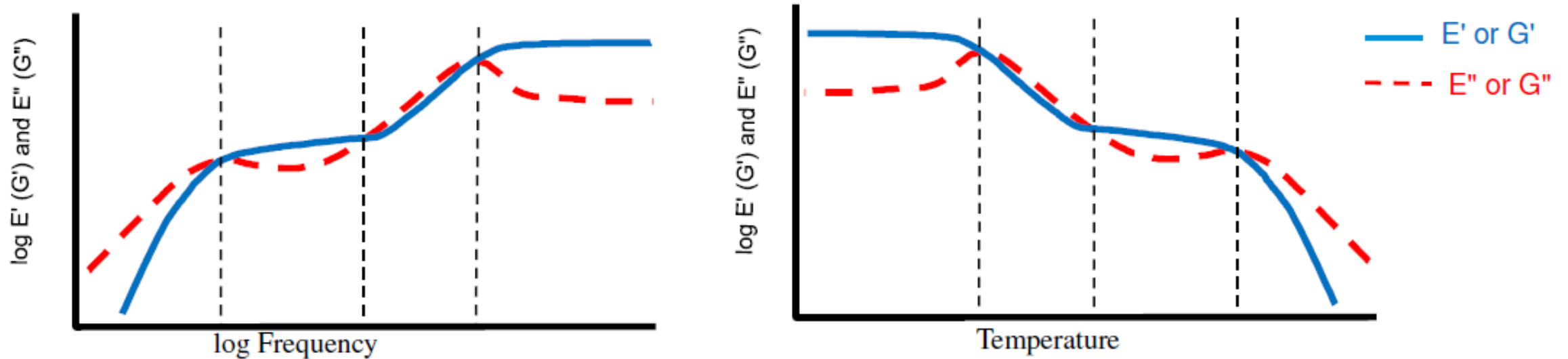


La spectrométrie mécanique ou DMA

Polyuréthane thermoplastique : effet du taux de segments rigides



Et l'équivalence temps-température ?

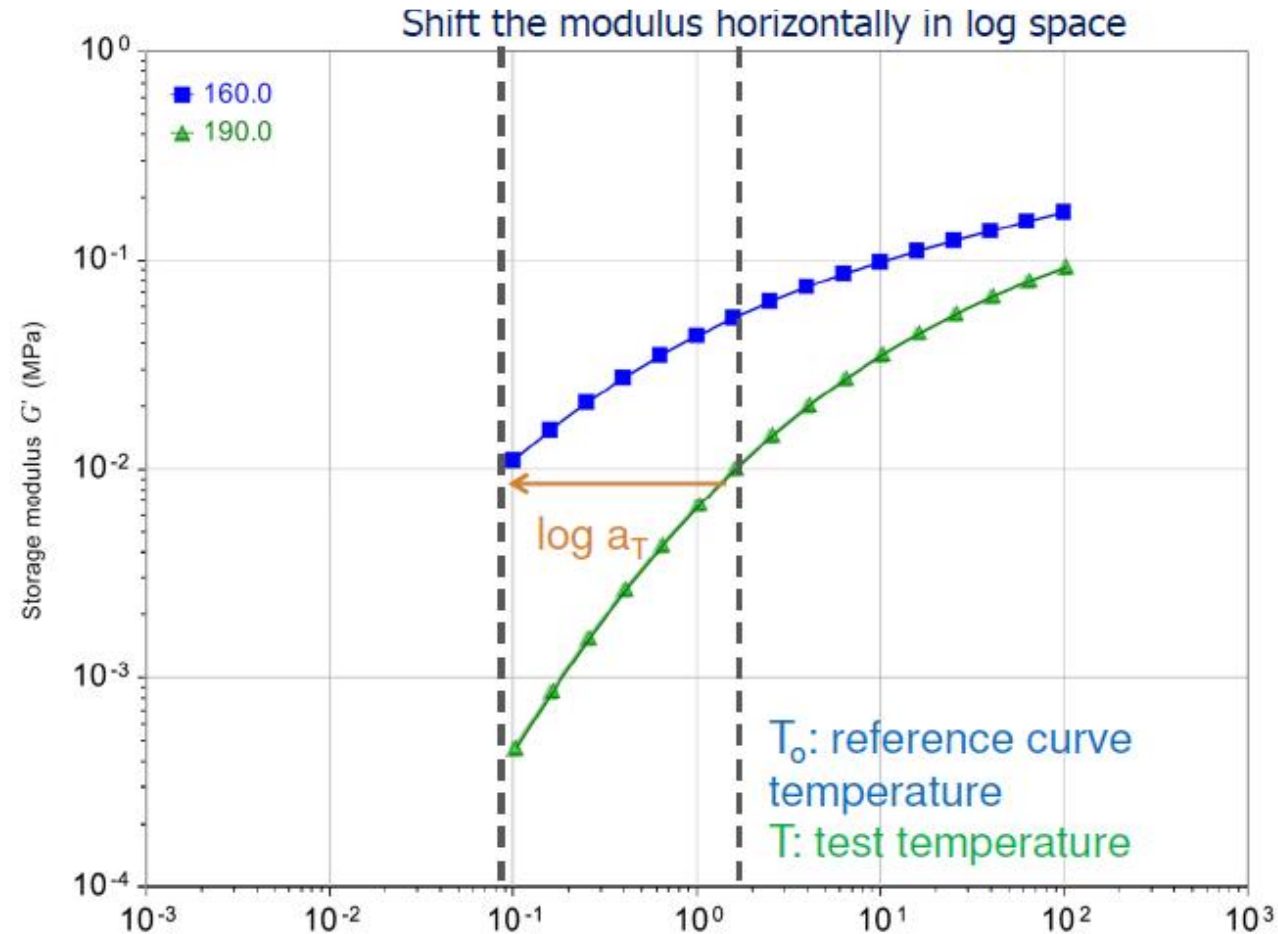


Certains matériaux présentent une dépendance temporelle proportionnelle à la dépendance thermique.

- ➔ Diminuer la température \equiv Augmenter la fréquence et vice versa
- ➔ Pour ces matériaux, les changements de température peuvent être utilisés pour « rééchelonner » le temps et **prédire le comportement sur des échelles de temps qui ne sont pas facilement mesurables.**

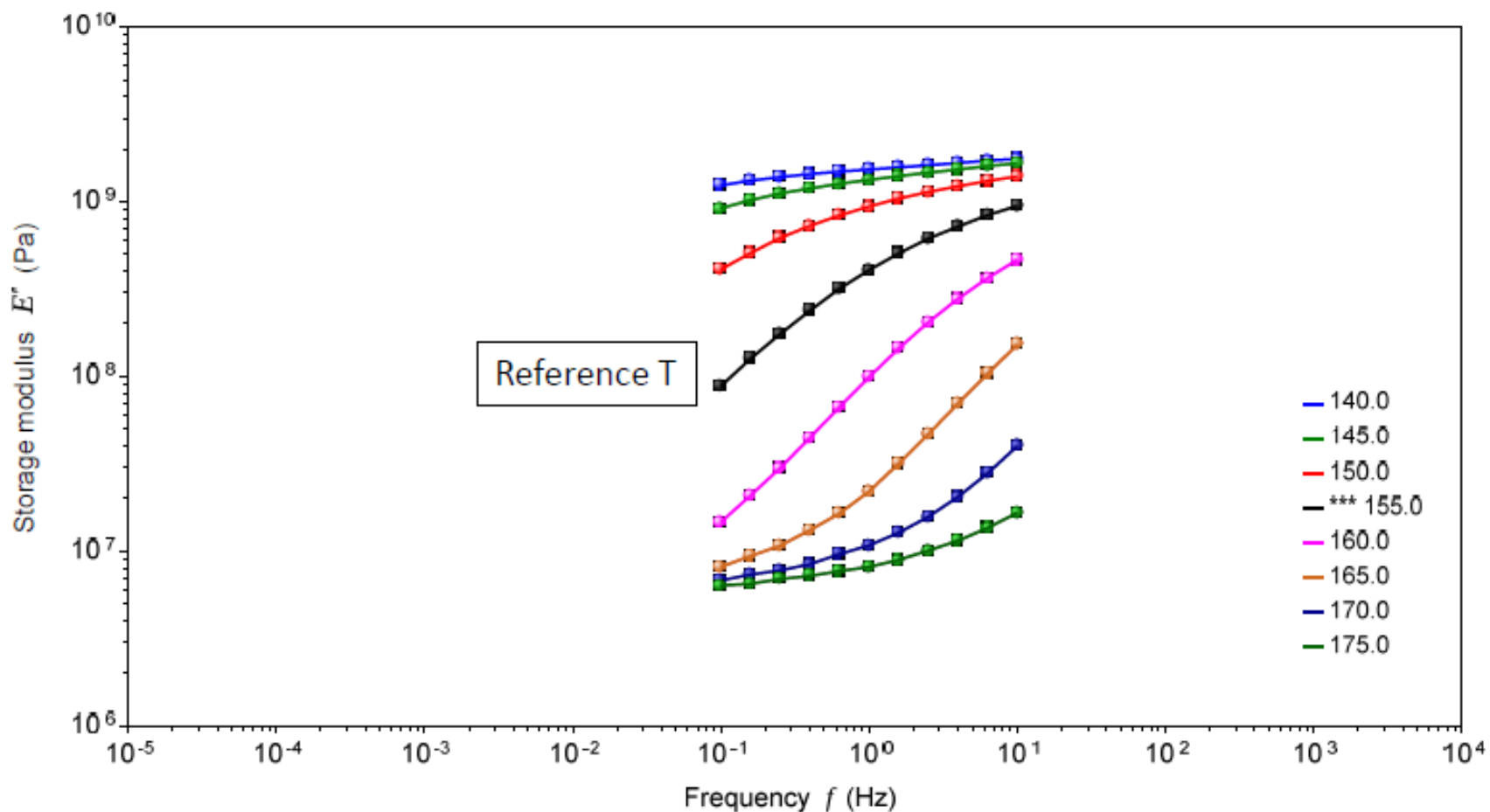
Et l'équivalence temps-température ?

a_T = facteur de glissement

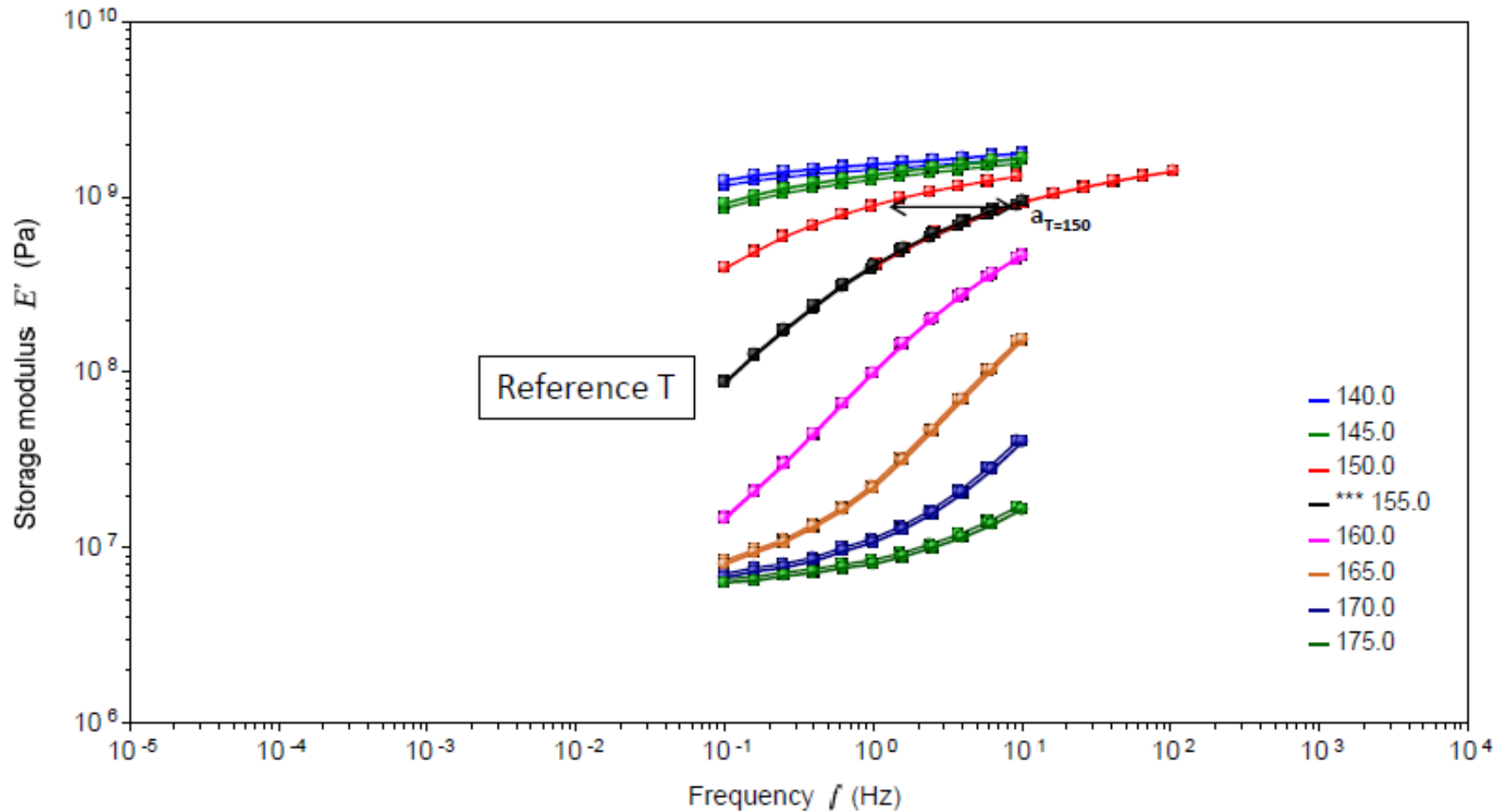


Dealy, J., Plazek, D., *Time-Temperature Superposition – A Users Guide*, Rheology Bulletin, **78**(20) 16 (2009)

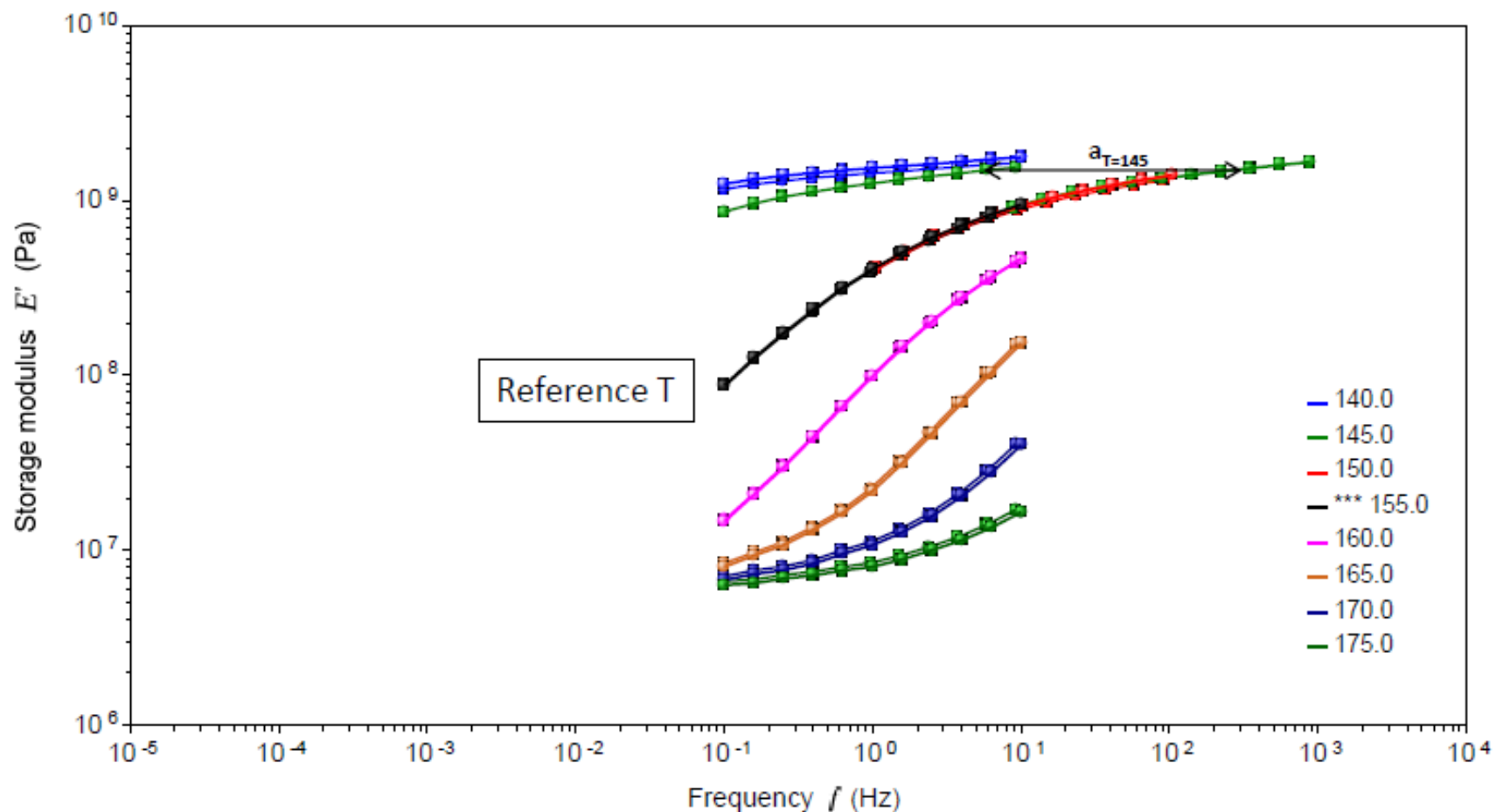
Et l'équivalence temps-température ?



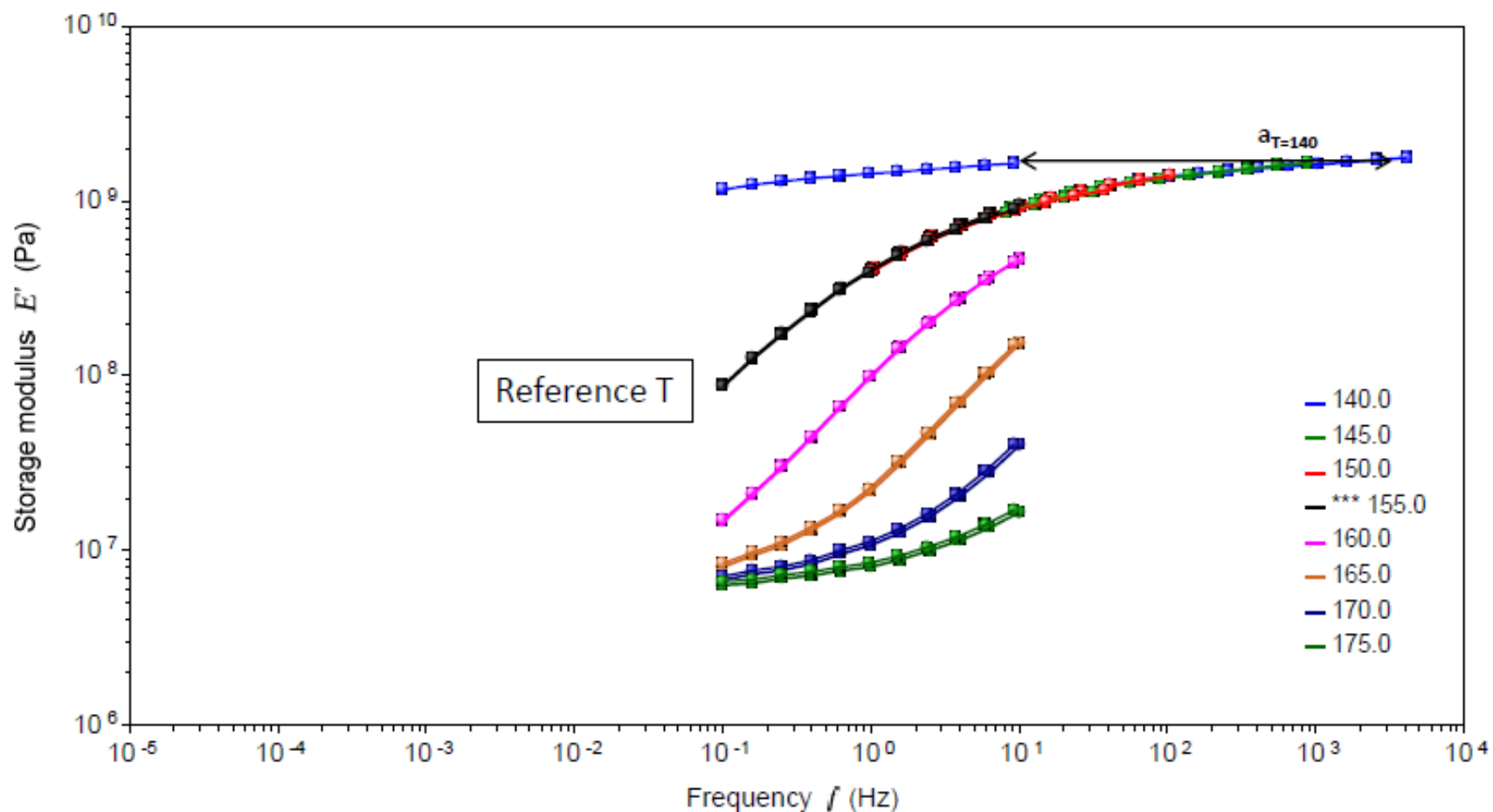
Et l'équivalence temps-température ?



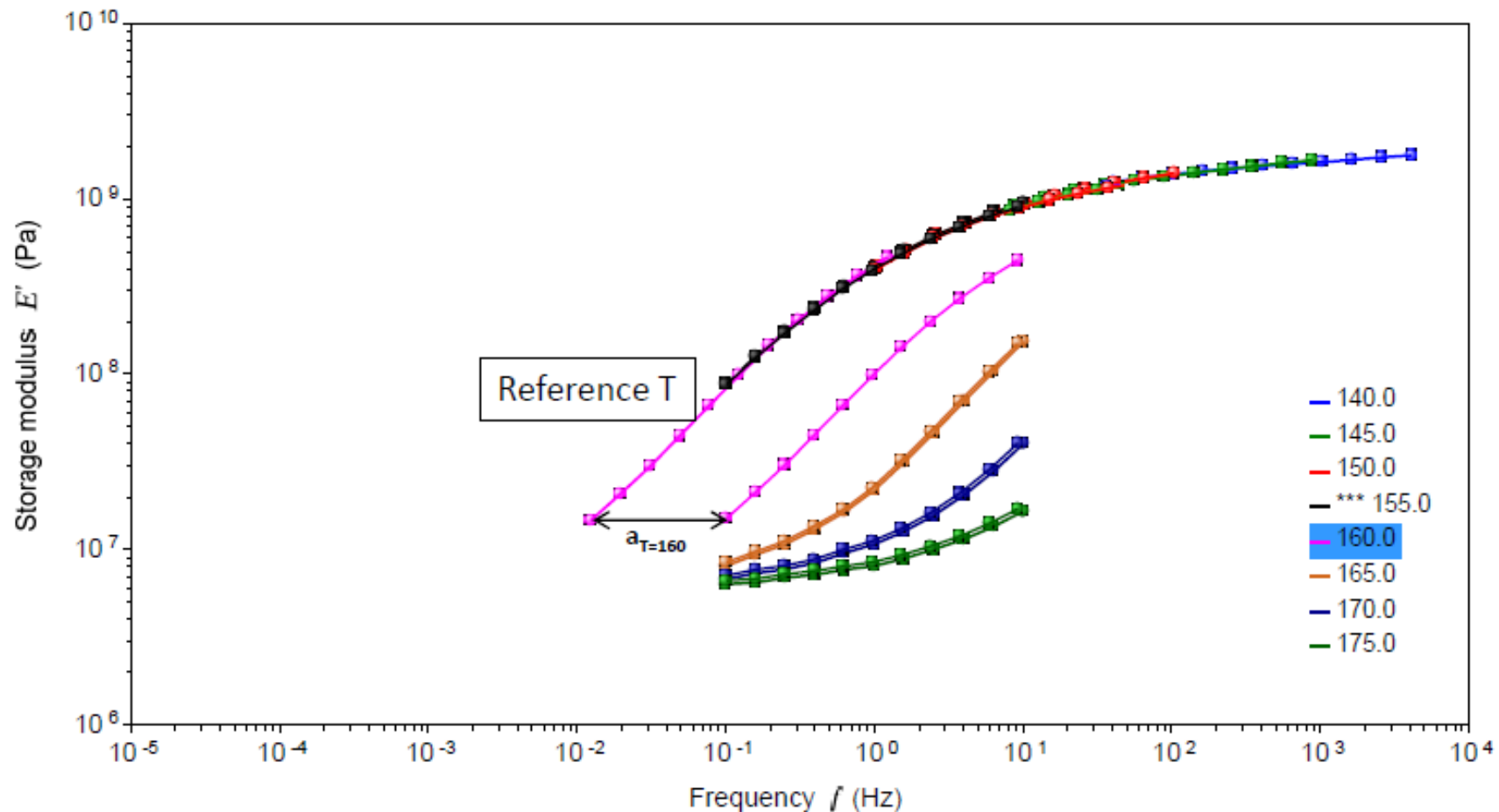
Et l'équivalence temps-température ?



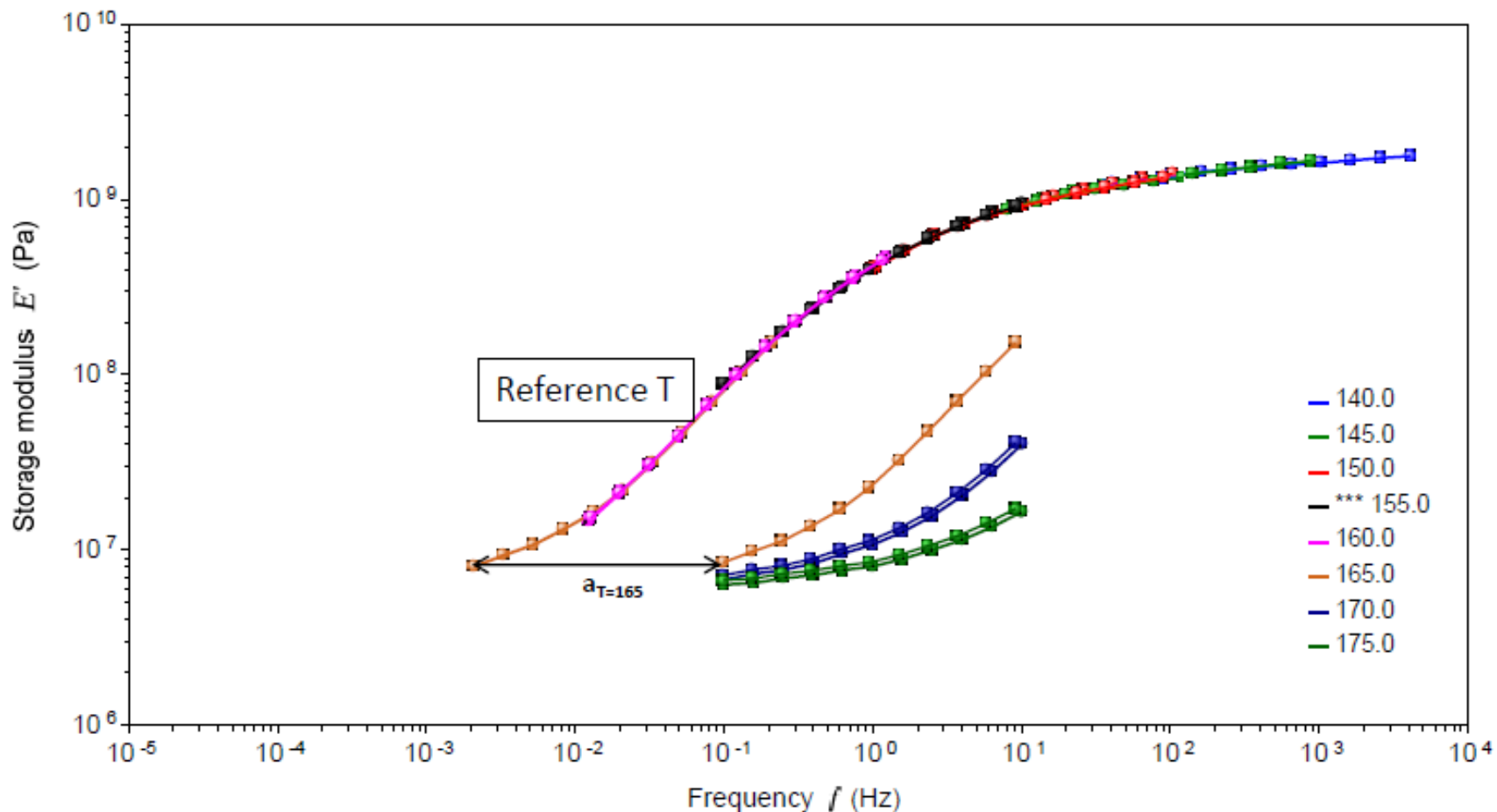
Et l'équivalence temps-température ?



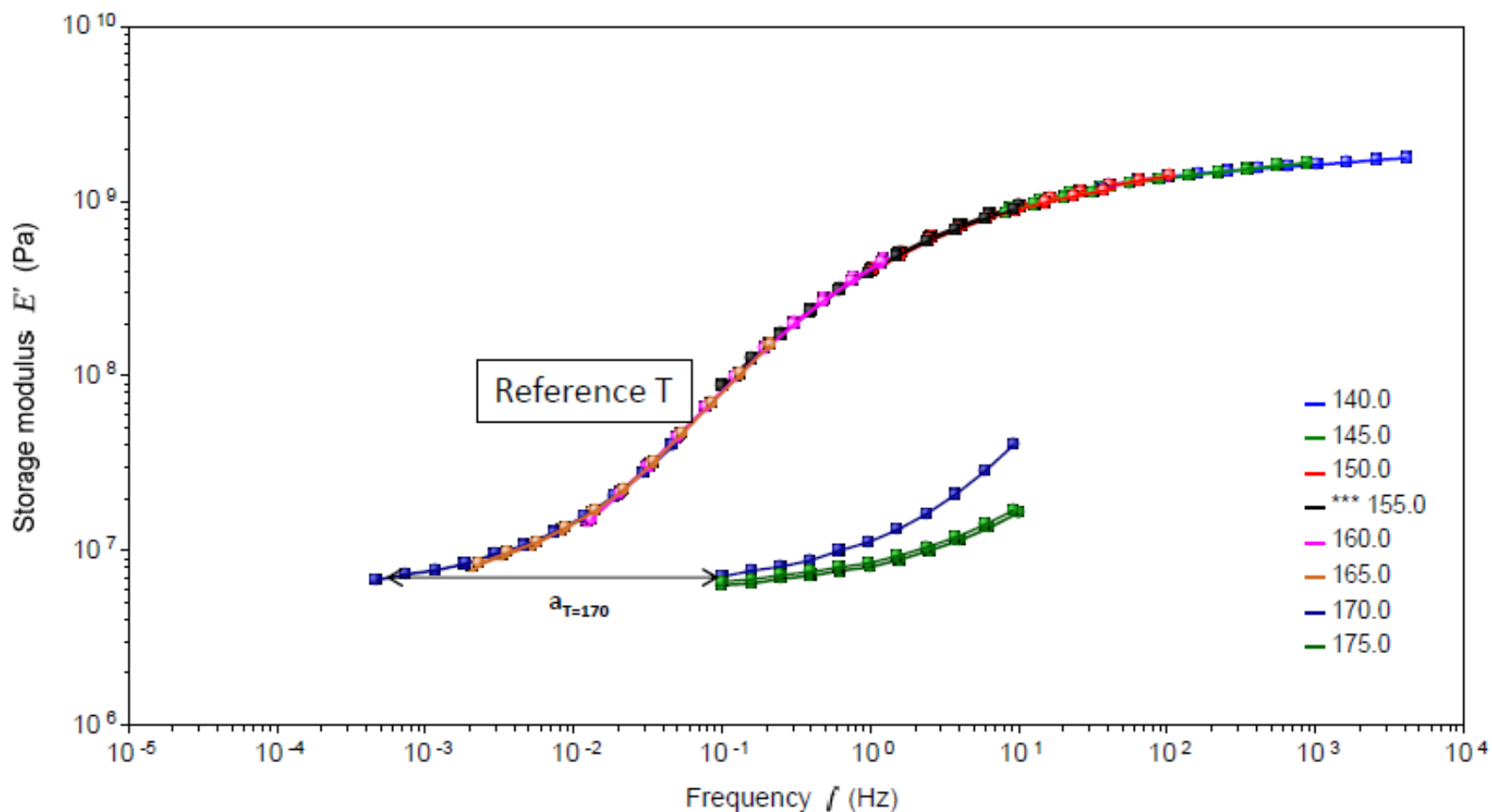
Et l'équivalence temps-température ?



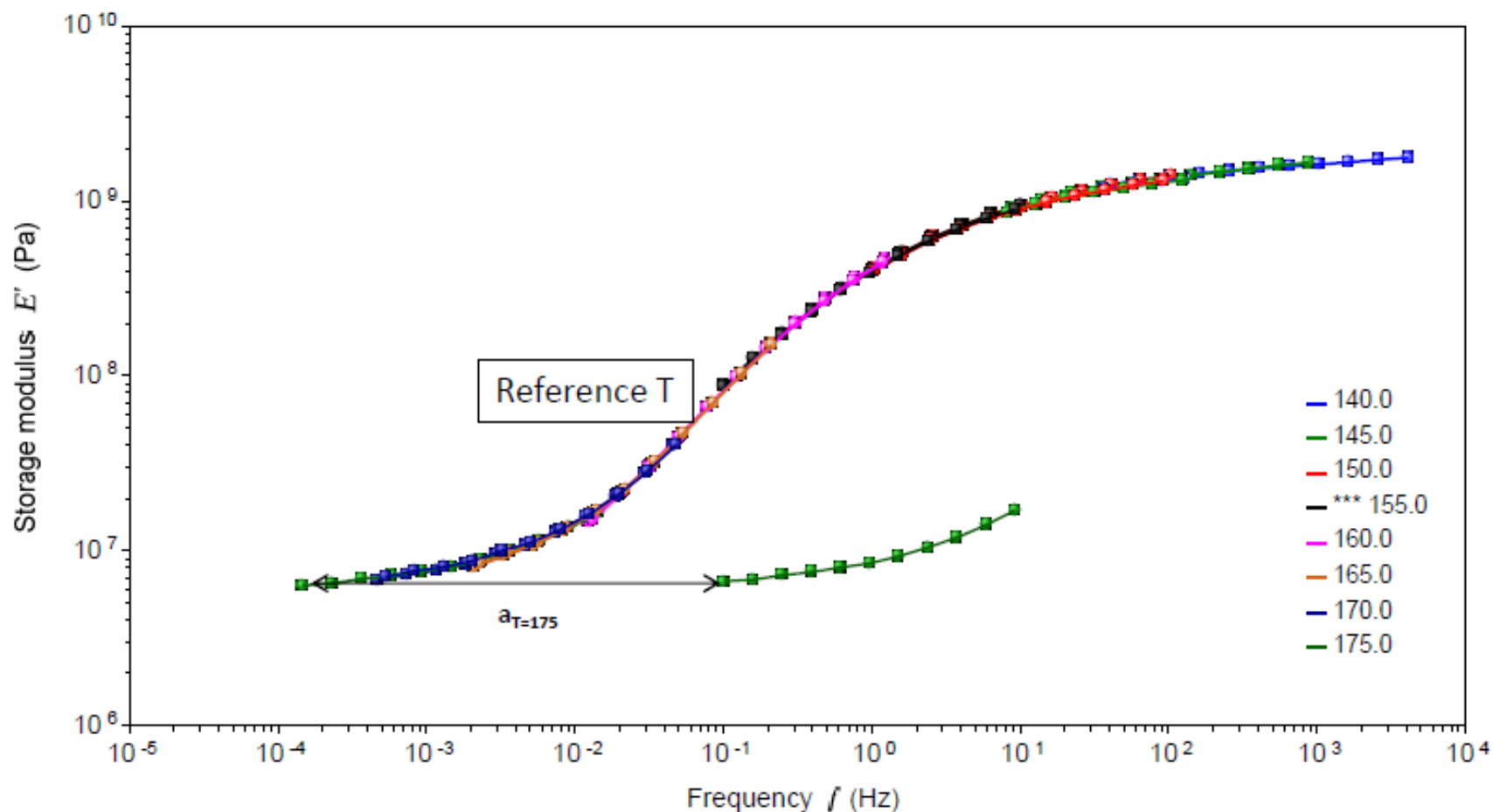
Et l'équivalence temps-température ?



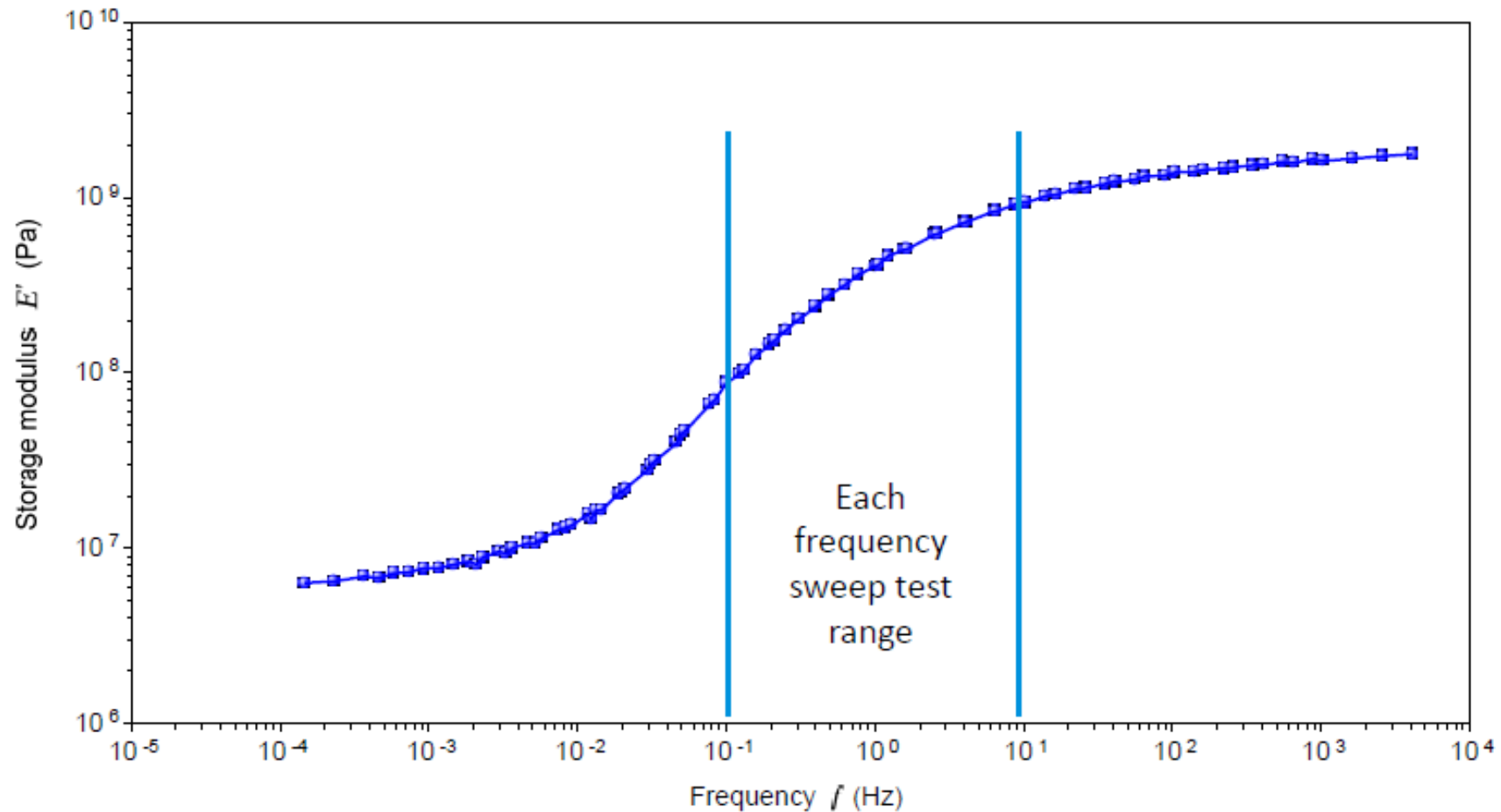
Et l'équivalence temps-température ?



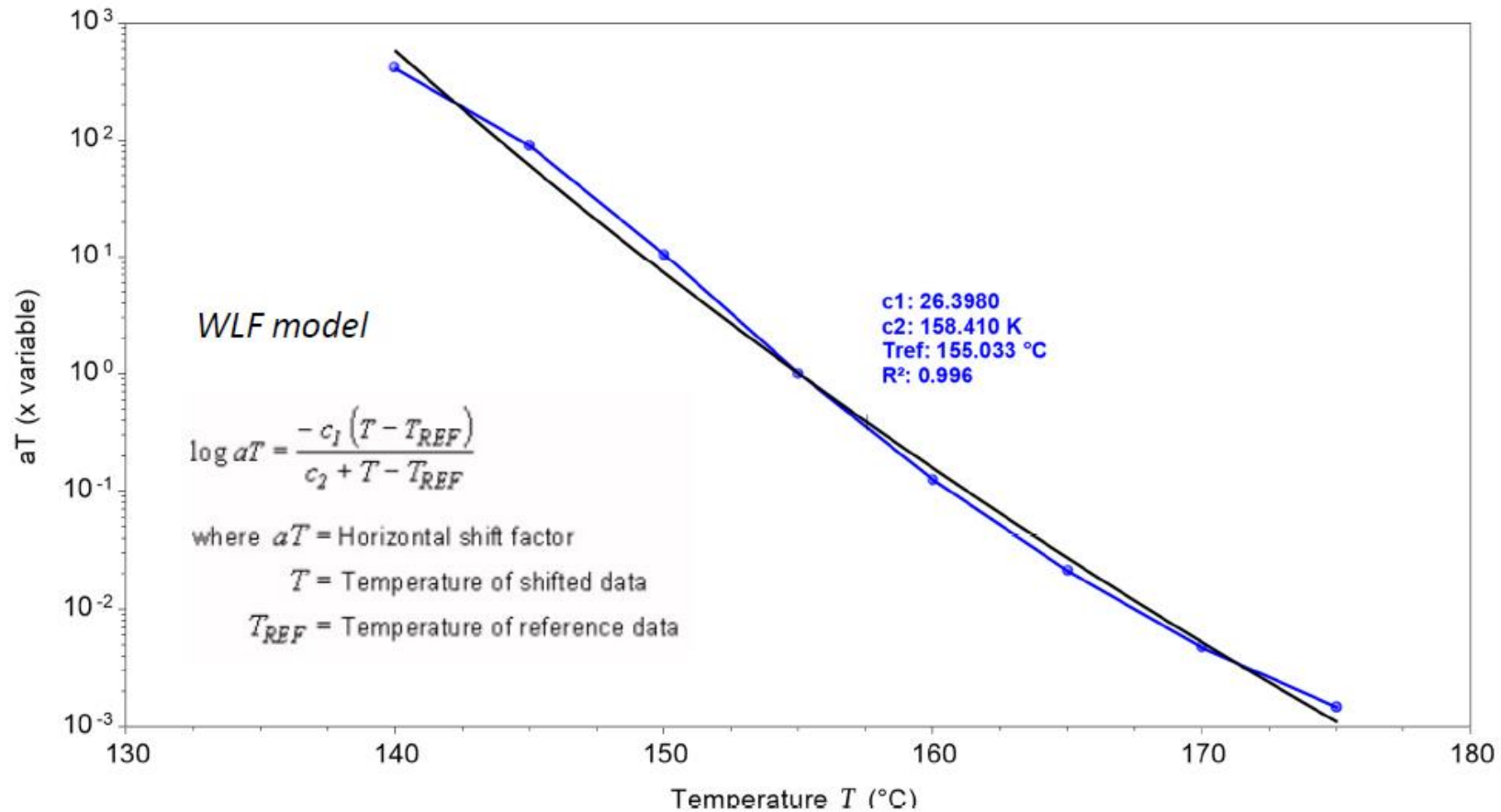
Et l'équivalence temps-température ?



Et l'équivalence temps-température ?

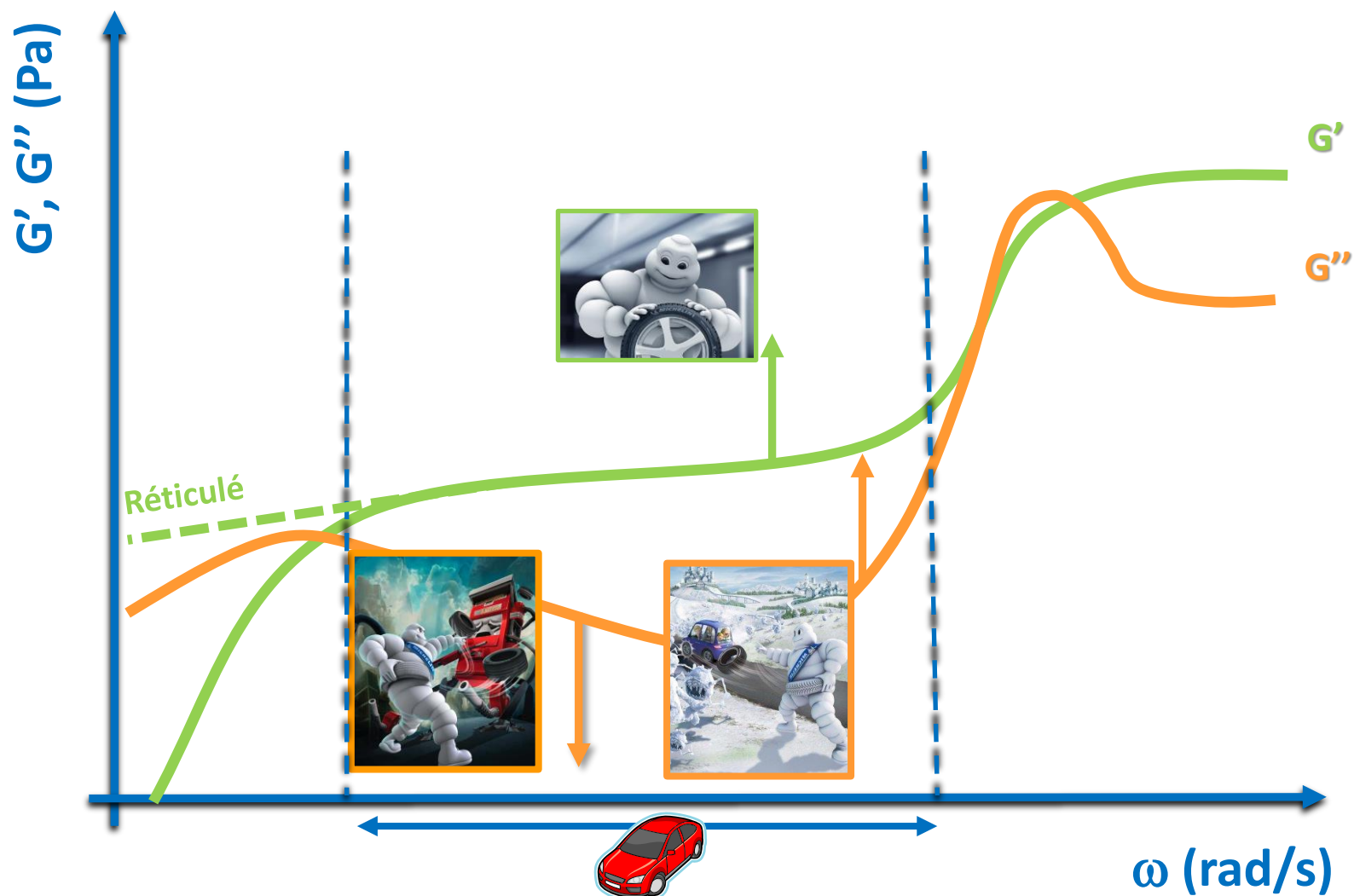


Et l'équivalence temps-température ?

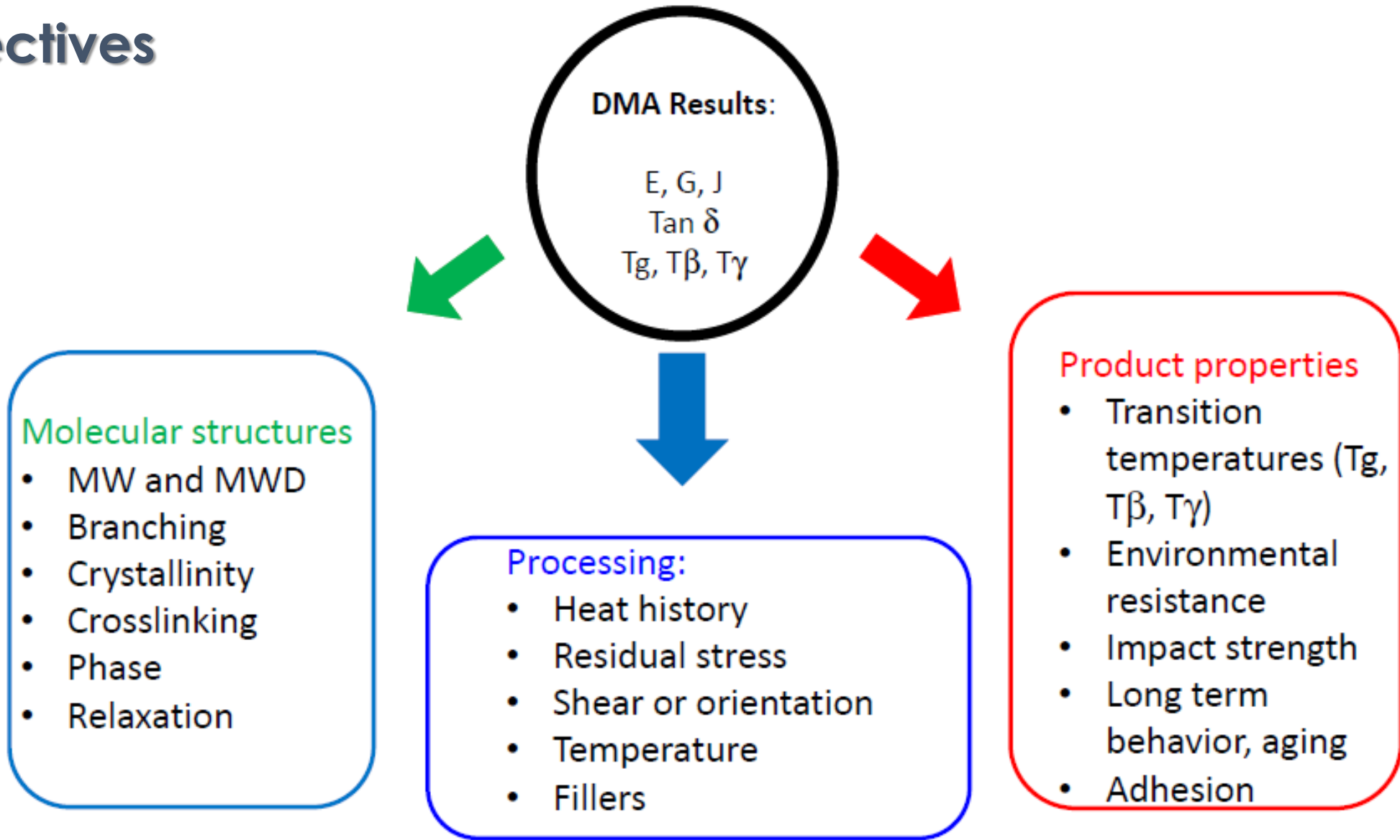


L'évolution du facteur de glissement a_T avec la température permet de remonter à des **énergies d'activations** des phénomènes de relaxation via des lois de type Arrhenius ou WLF (Williams Landel Ferry).

Et l'équivalence temps-température ?



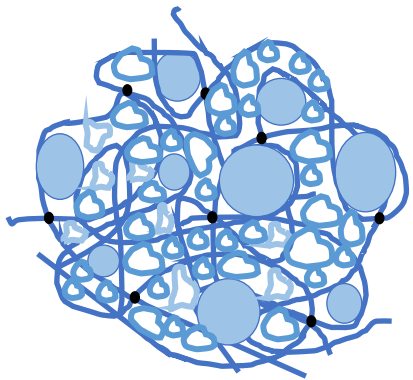
Perspectives



Perspectives

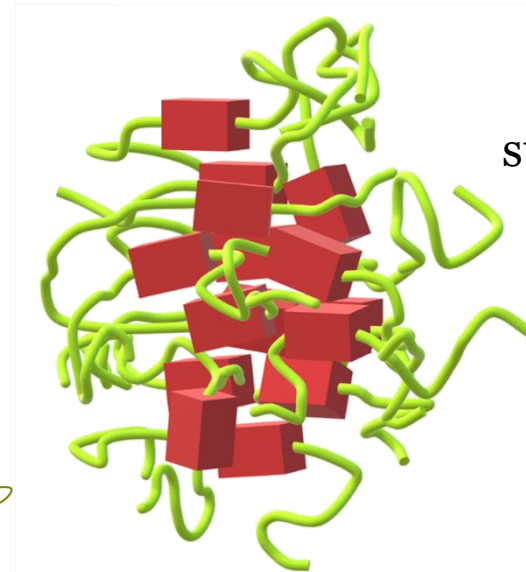
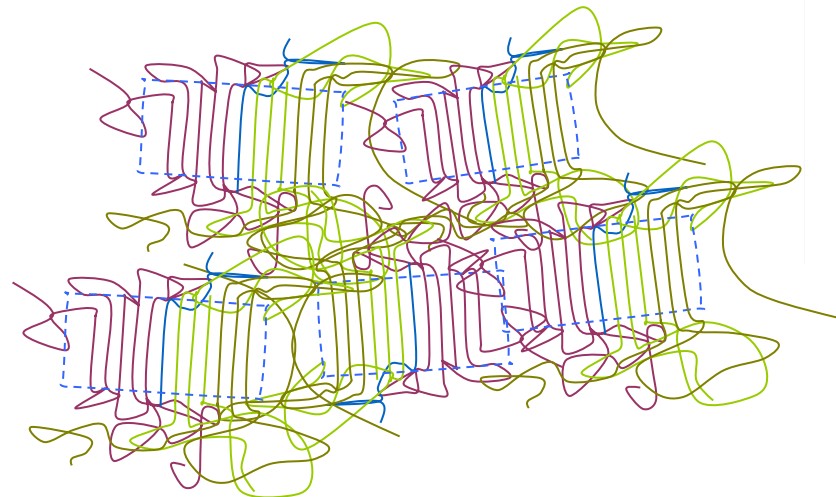
Problématique des polymères hétérogènes : présence d'interfaces et/ou d'interphases + l'équivalence temps-température ne s'applique pas.

- ➔ Localement, modification de la mobilité moléculaire et des propriétés viscoélastiques
- ➔ Présence d'hétérogénéités de mobilité.



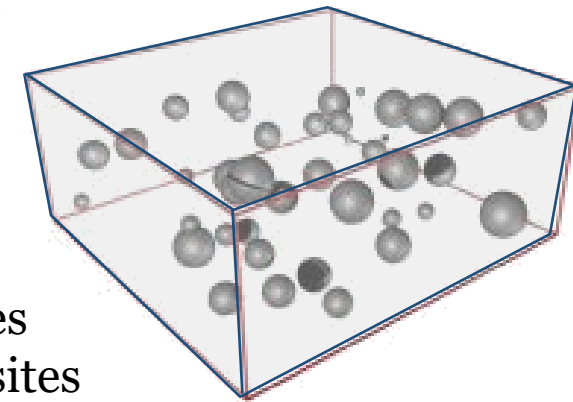
Polymères
semi-cristallins

*Les mélanges
physiques*



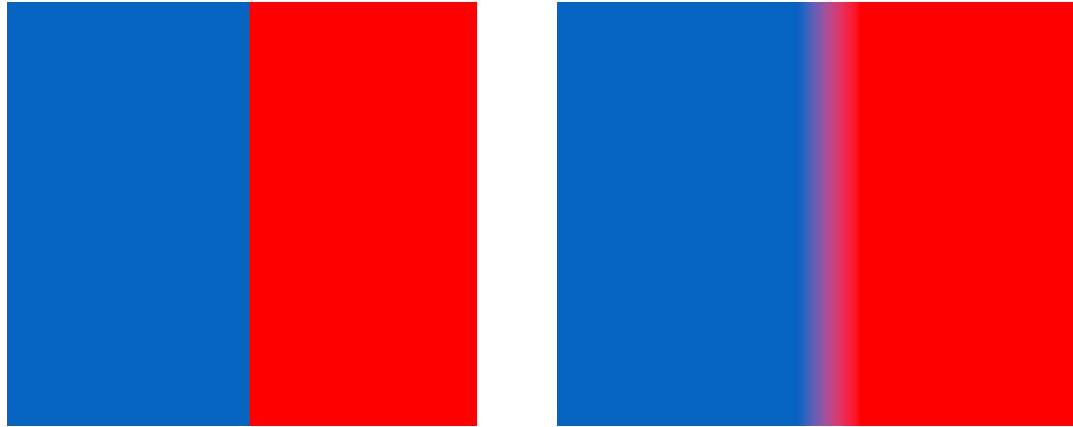
Assemblages
supramoléculaires
Copolymères
TPE

Composites
Nanocomposites



Perspectives

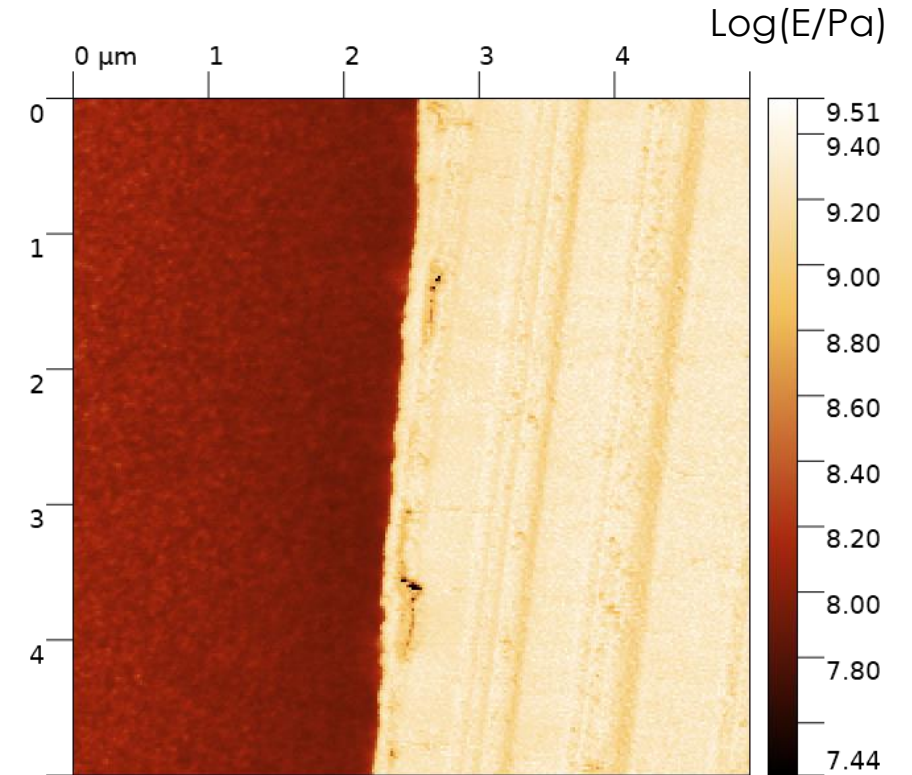
Exemple : Interface dans un mélange de polymère



Interface



Interphase



- Compréhension des propriétés viscoélastiques locales pour mieux appréhender le comportement macroscopique du matériau
- Paramètres d'entrée pour des approches de modélisation

Vers la nano-DMA !

Plan

La viscoélasticité, qu'est-ce que c'est ?

La spectroscopie mécanique ou DMA

L'équivalence temps-température

Limites et perspectives : vers les mesures locales

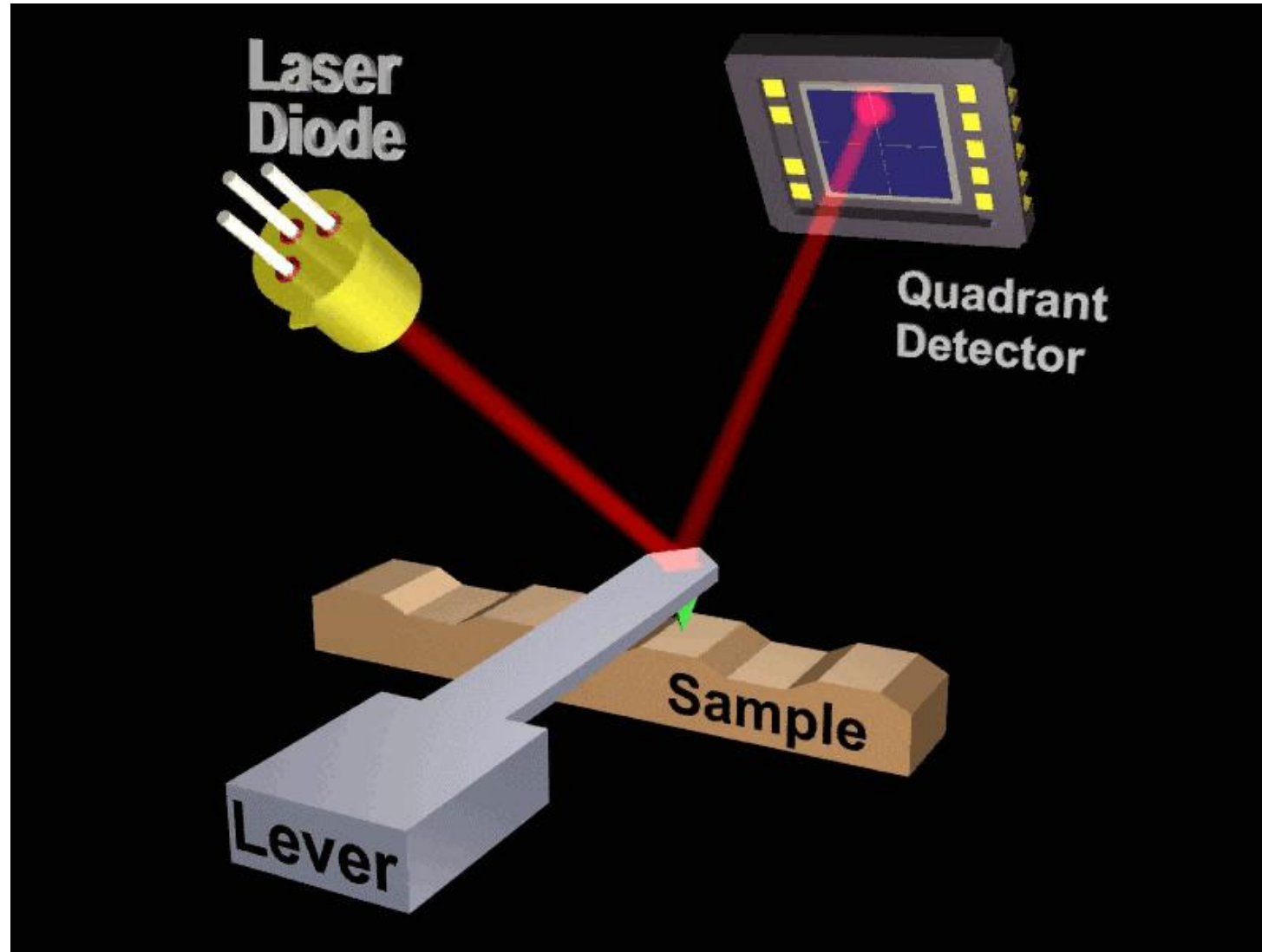
Extension à la nanoéchelle

- nanoDMA
- CR-AFM
- Intermodulation AFM
- ...

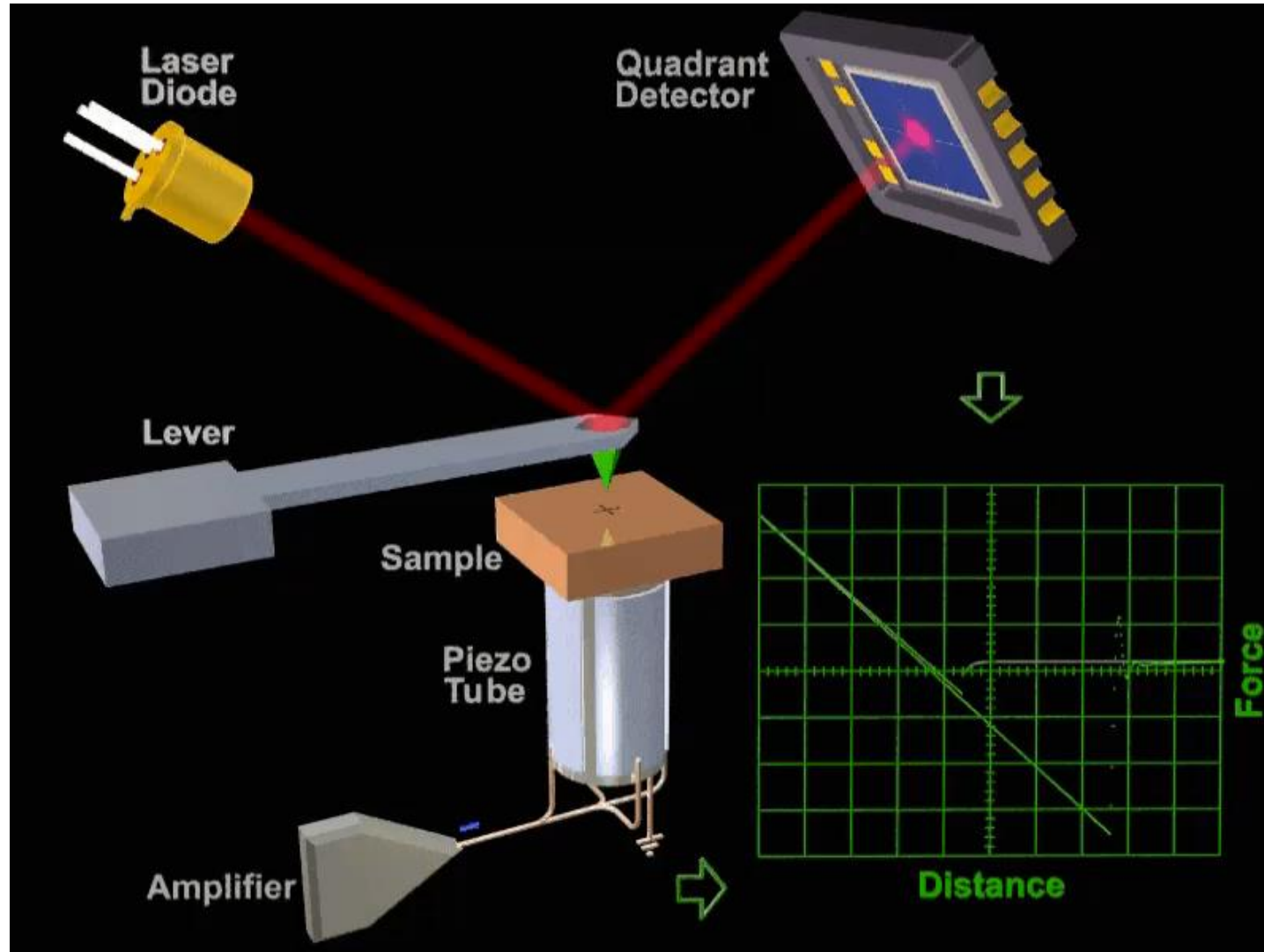
Conclusions et Perspectives



Scanning Probe Microscopy



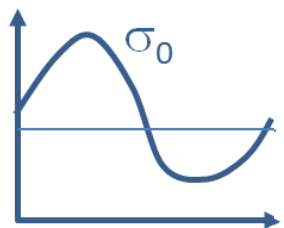
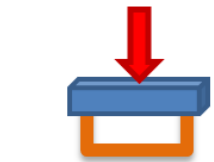
Scanning Probe Spectroscopy



Viscoelastic properties at the nanoscale

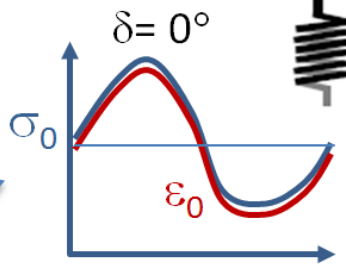
Dynamic Mechanical Analysis (DMA)

$$\sigma(t) = \sigma_0 \sin(\omega t)$$

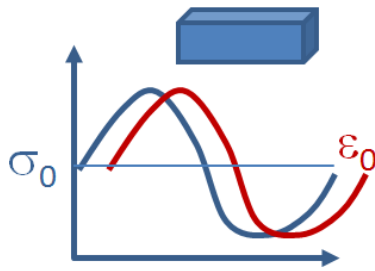
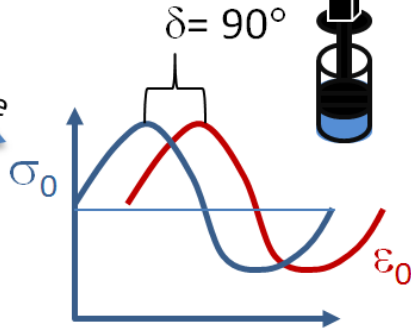


In-Phase

$$\varepsilon(t) = \varepsilon_0 \sin(\omega t)$$

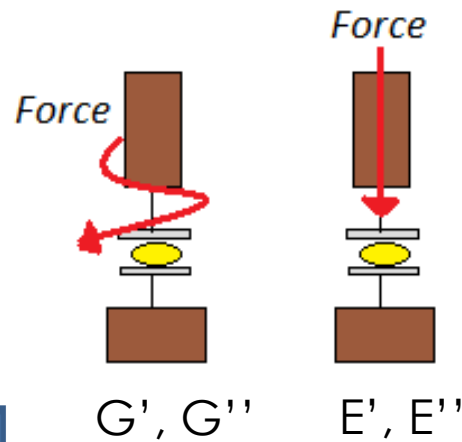


Out-of-Phase

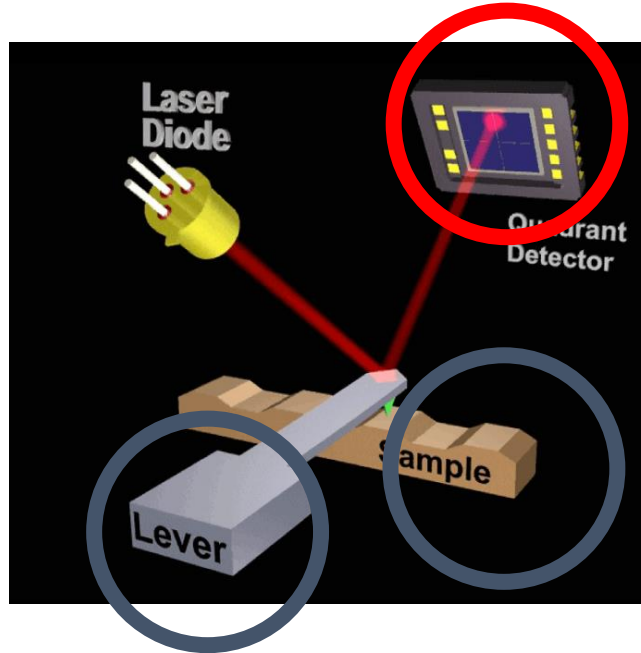
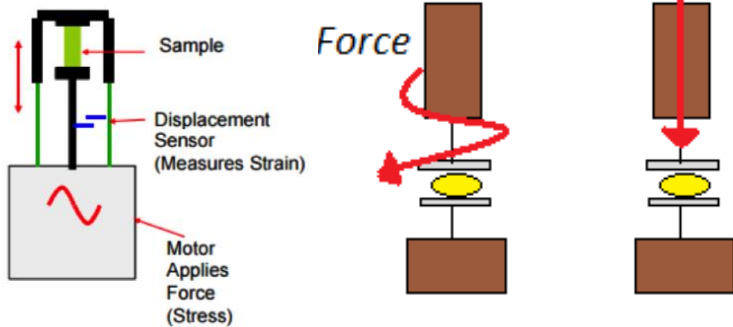


$$\varepsilon(t) = \varepsilon_0 \sin(\omega t + \delta)$$

$$d\sigma/dt = \varepsilon \sigma_0 \cos(\omega t) = \varepsilon_0 \sin(\omega t + \pi/2)$$

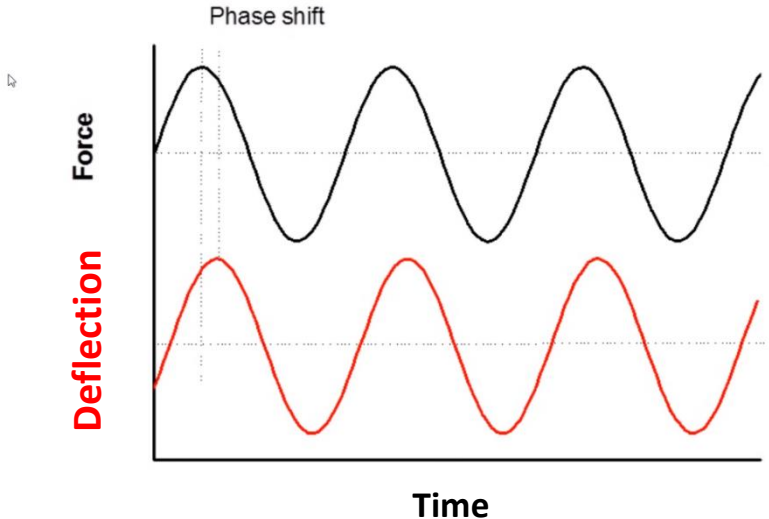
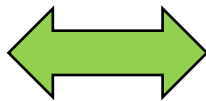
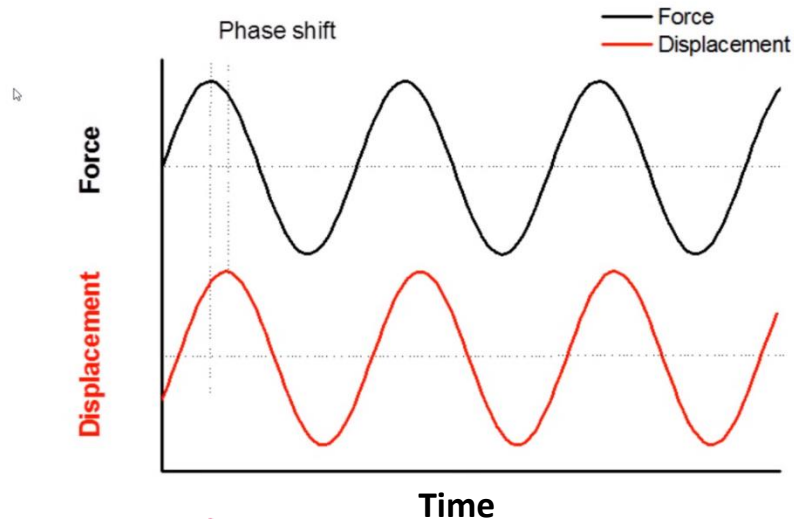


From DMA to nDMA ...

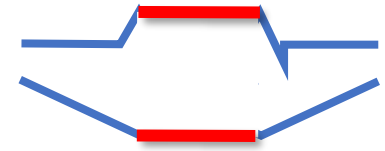


$$D(t) = D_1 e^{i(\omega t + \phi)} + D_0$$

$$Z(t) = Z_1 e^{i(\omega t + \psi)} + Z_0$$



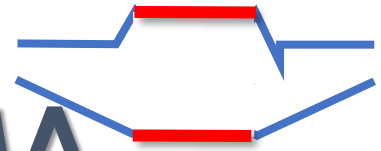
Notation



- One common notation set will describe all three regimes introduced above, with Z modulation via either Z scanner or sample actuator
- Harmonic signals:
- $z(t) = Z_1 \sin(\omega t + \psi) + Z_0$
 - Z displacement, probe or sample actuator
 - Z_1, ψ - amplitude and phase at frequency $\omega = 2\pi f$
(from Z-sensor lock-in or deflection-calibrated on a hard reference sample)
 - Measured or calibrated on a hard reference sample
- $d(t) = D_1 \sin(\omega t + \varphi) + D_0$
 - Measured deflection (Vertical deflection signal)
 - D_1, φ - amplitude and phase (from lock-in) at frequency $\omega = 2\pi f$
- K_c - cantilever spring constant (calibrated, known)

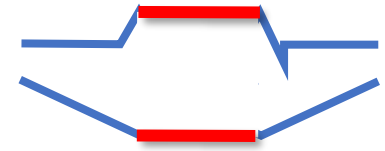


Derivation of General Equations for nDMA



- Key idea: Using definition of Dynamic Stiffness
- Stiffness $[N/m] = \text{Force } [nN] / \text{Deformation } [nm] : S^* = F^* / L^*$
 - Force: measured by deflection (channel 1)
 - Deformation: in the displacement measurement (channel 2)
 - Deformation $L = Z$ displacement minus Deflection
- For dynamic stiffness and harmonic excitation: complex values
- Equations for complex values:
 - $F^* = K_c D_1 e^{i(\omega t + \varphi)}$
 - $L^* = Z_1 e^{i(\omega t + \psi)} - D_1 e^{i(\omega t + \varphi)}$
 - $S^* = S' + iS'' = K_c D_1 e^{i(\omega t + \varphi)} / [Z_1 e^{i(\omega t + \psi)} - D_1 e^{i(\omega t + \varphi)}]$

General formulation



Equations relating tan-Delta and Storage/Loss Modulus or Stiffness to measured lock-in amplitudes and phases

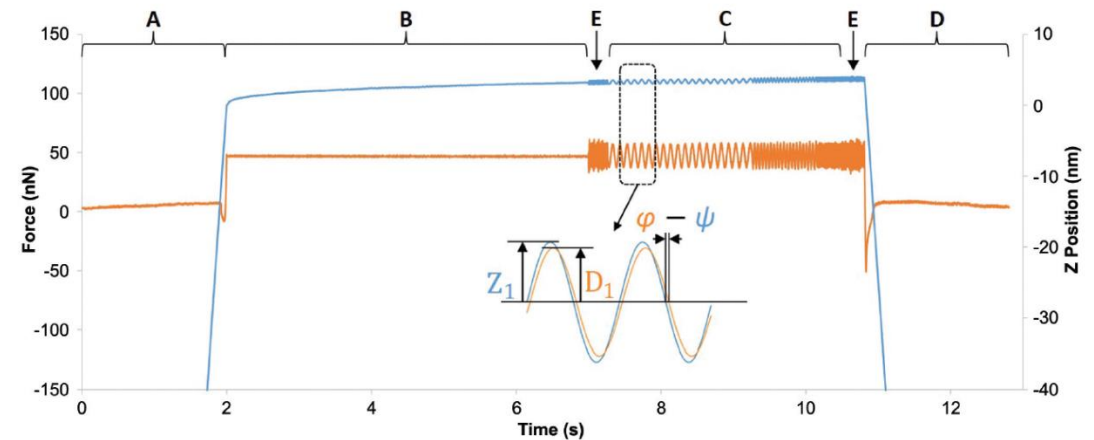
- $S^* = S' + iS'' = K_c D_1 e^{i\varphi} / [Z_1 e^{i\psi} - D_1 e^{i\varphi}]$

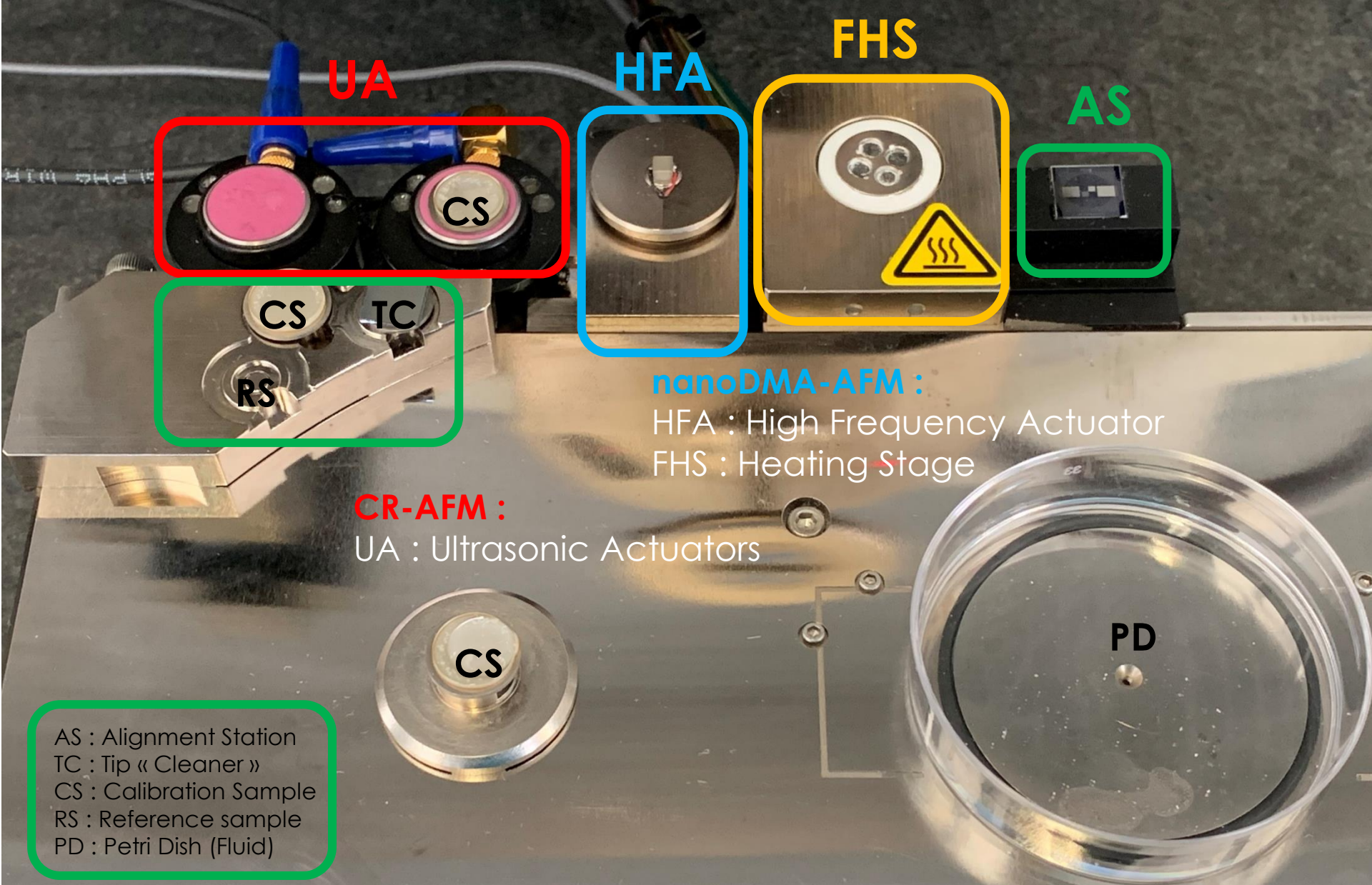
- $S' = \frac{K_c D_1}{Z_1} \frac{\cos(\varphi - \psi) - (D_1/Z_1)}{(D_1/Z_1)^2 - 2(D_1/Z_1) \cos(\varphi - \psi) + 1}$

- $S'' = \frac{K_c D_1}{Z_1} \frac{\sin(\varphi - \psi)}{(D_1/Z_1)^2 - 2(D_1/Z_1) \cos(\varphi - \psi) + 1}$

- $\tan \delta = S''/S' = \frac{\sin(\varphi - \psi)}{\cos(\varphi - \psi) - (D_1/Z_1)}$

- $E' = \frac{S'}{2a_c}$; $E'' = \frac{S''}{2a_c}$; where a_c is contact radius (e.g., from JKR)





UA

HFA

FHS

AS

CS

CS

TC

RS

nanoDMA-AFM :

HFA : High Frequency Actuator

FHS : Heating Stage

CR-AFM :

UA : Ultrasonic Actuators

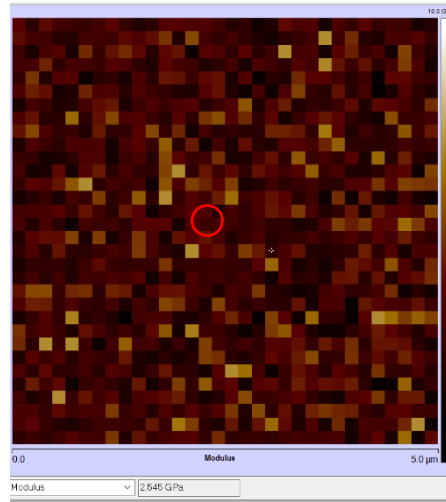
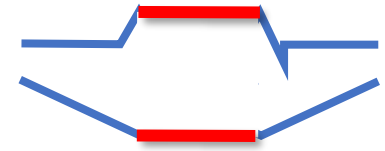
CS

PD

AS : Alignment Station
TC : Tip « Cleaner »
CS : Calibration Sample
RS : Reference sample
PD : Petri Dish (Fluid)

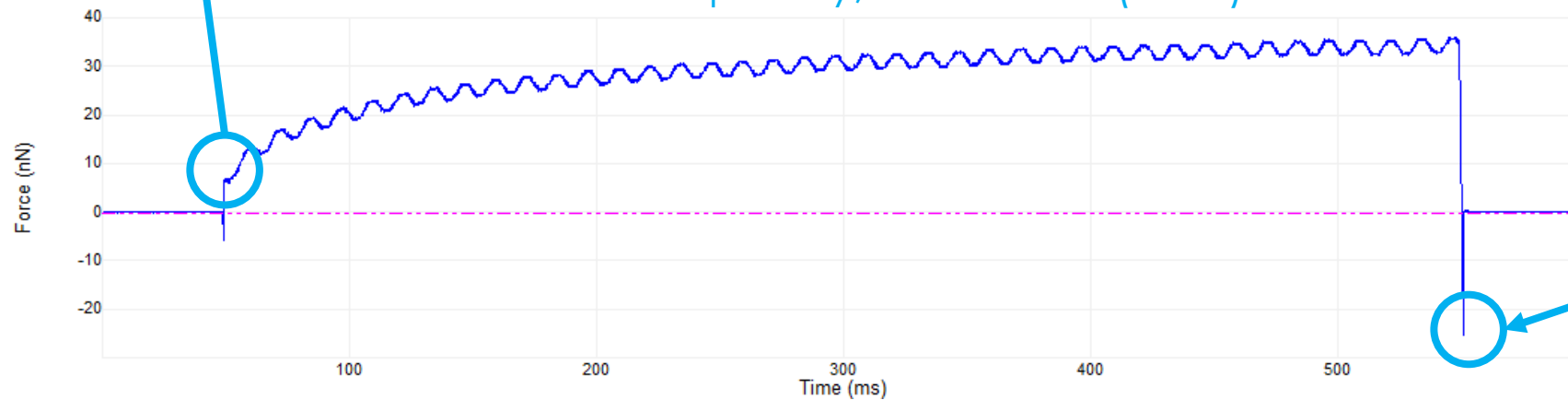


nanoDynamic Mechanical Analysis



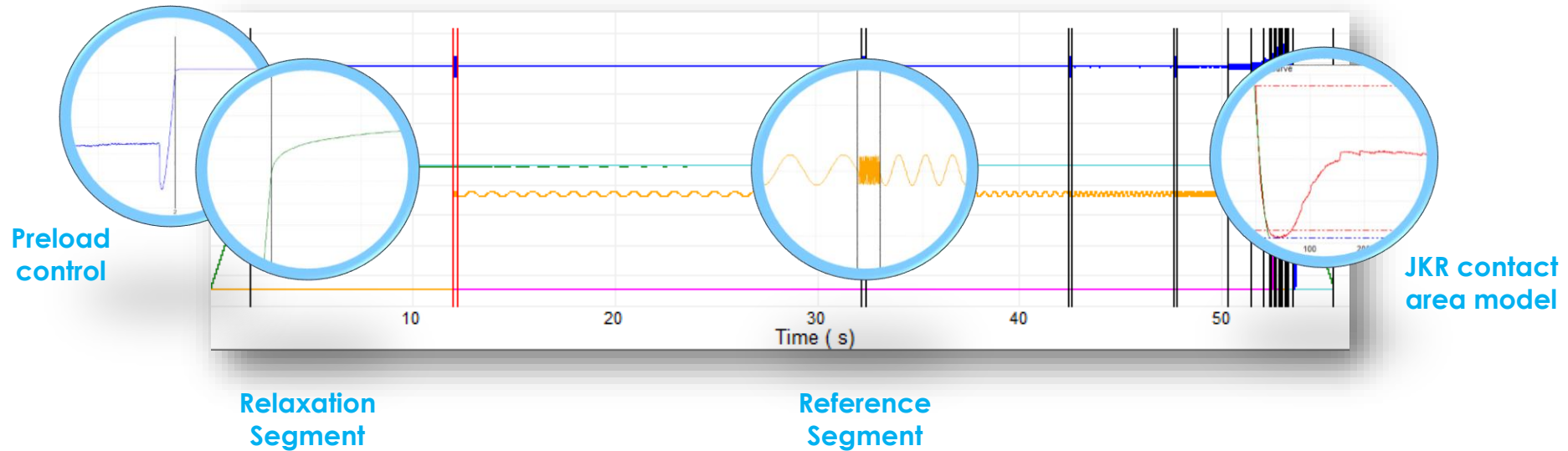
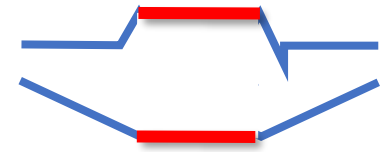
Preload control

Low-frequency, in-contact (80Hz)



Adhesion quantified

nanoDynamic Mechanical Analysis



Modulated force, embedded in force curve

Sub-nm amplitudes, stays in linear regime.

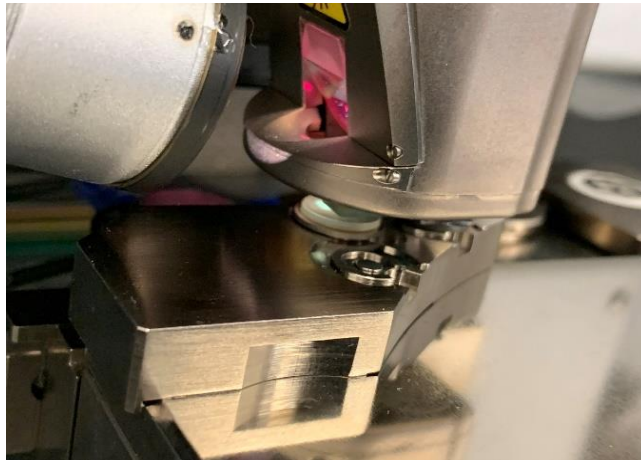
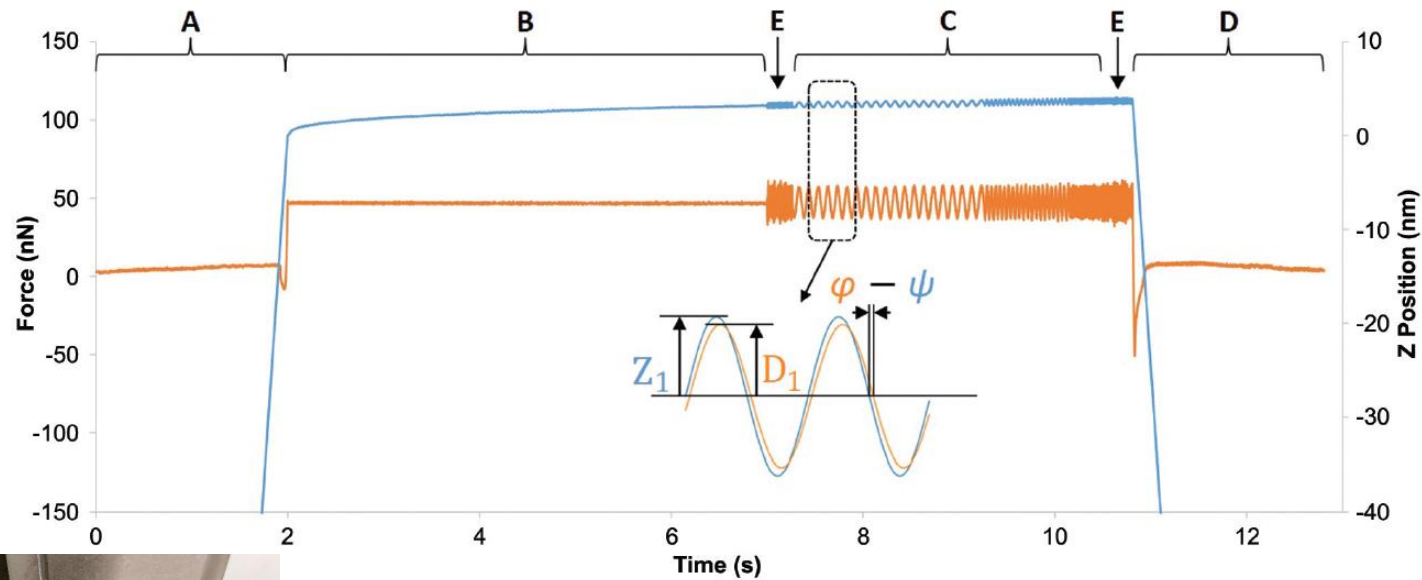
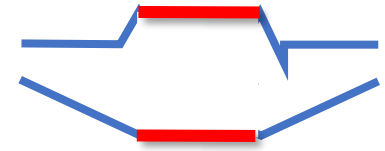
No ripping out of contact during modulation

Imaging and point spectroscopy

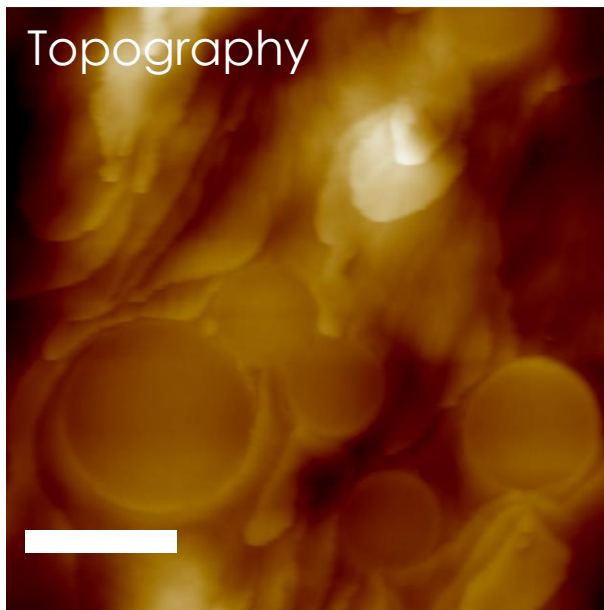
Quantitative data in both

Adapted from Bruker

nanoDynamic Mechanical Analysis



Calibration on Sapphire in the exact same conditions !



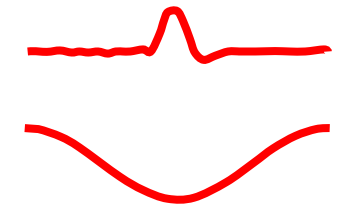
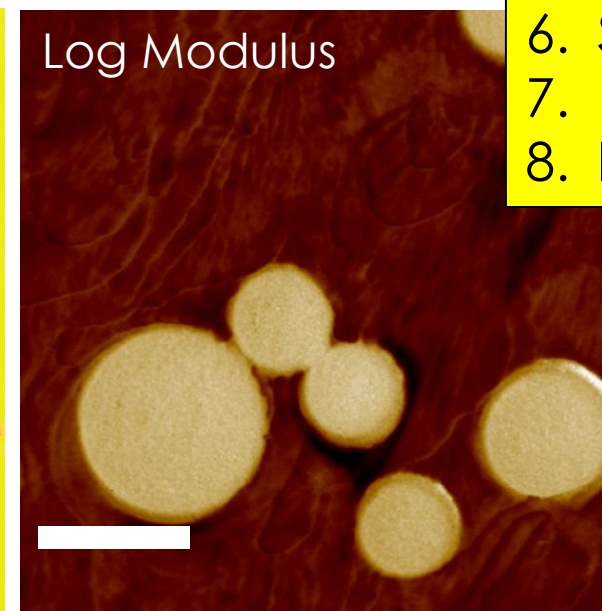
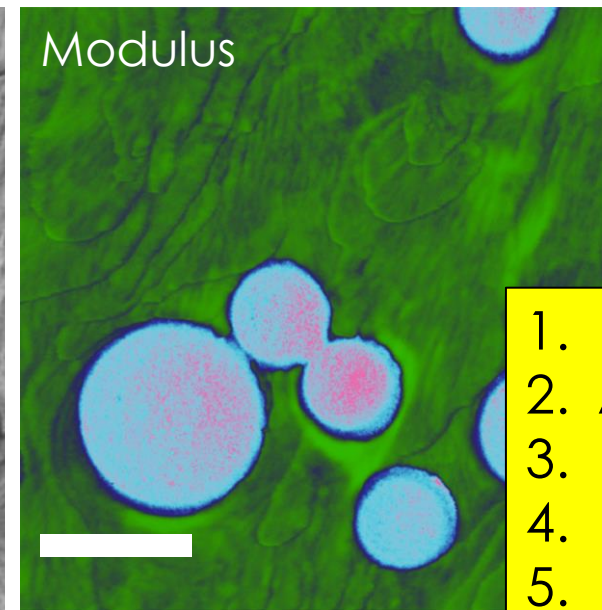
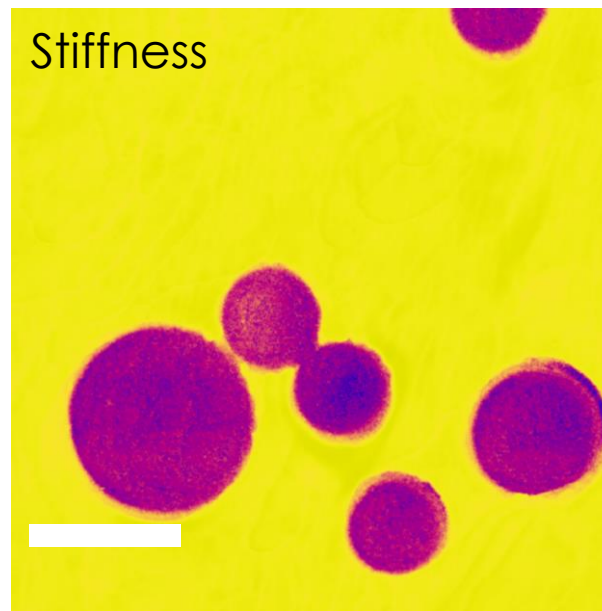
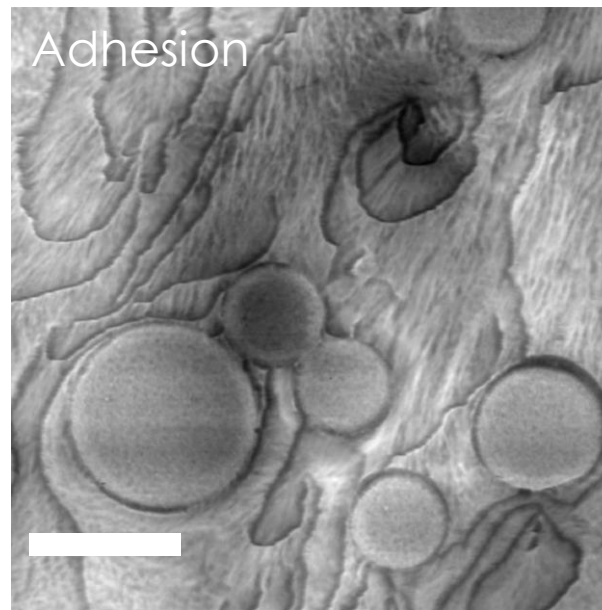
PS – PCL (25:75)

Tip : RTESPA 300 – 125

$k = 29.22 \text{ N/m}$
Radius = 125 nm

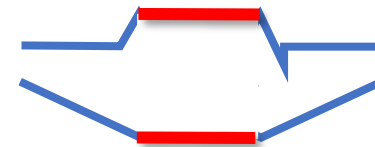
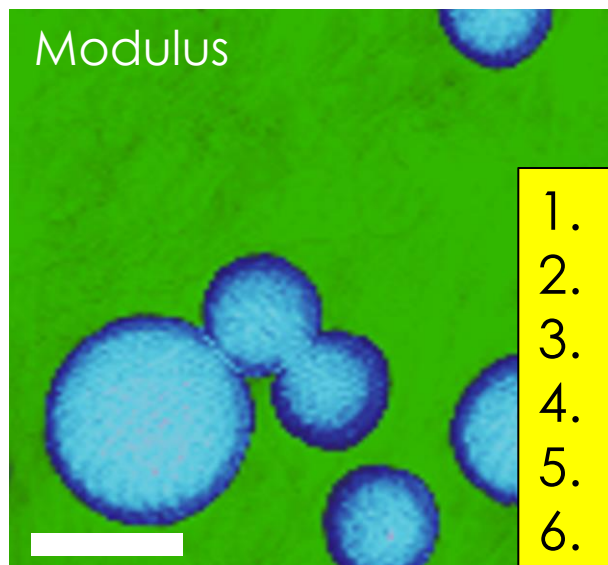
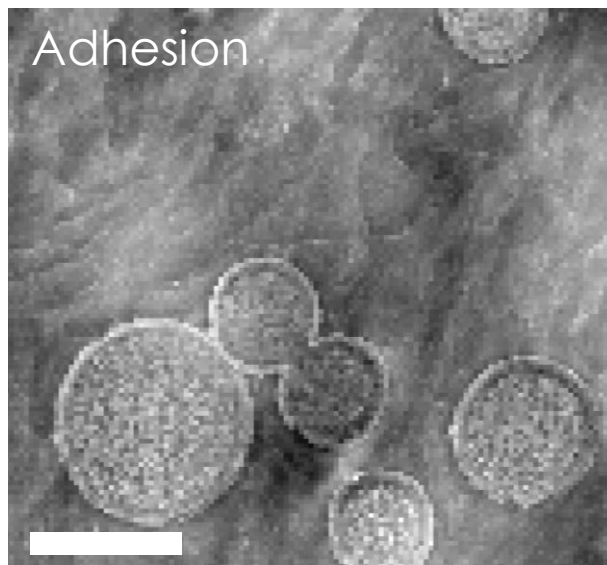
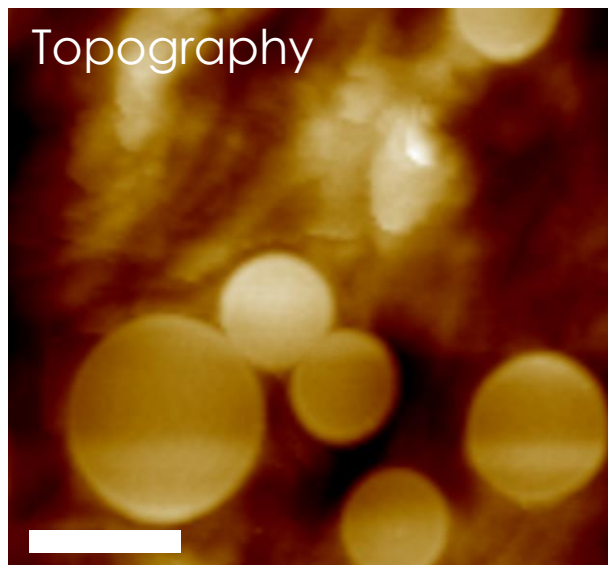
PeakForce Tapping QNM

512 x 512 pixels, PFSP = 15 nN
Scale bar = 2.0 μm



1. Height
2. Adhesion
3. DMT Modulus
4. Log DMT Modulus
5. Dissipation
6. Stiffness
7. Deformation
8. Indentation





PS – PCL (25:75)

Tip : RTESPA 300 – 125

$k = 29.22 \text{ N/m}$

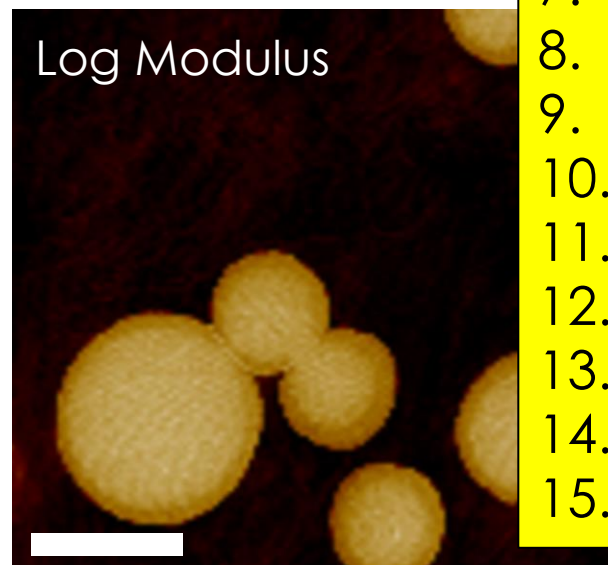
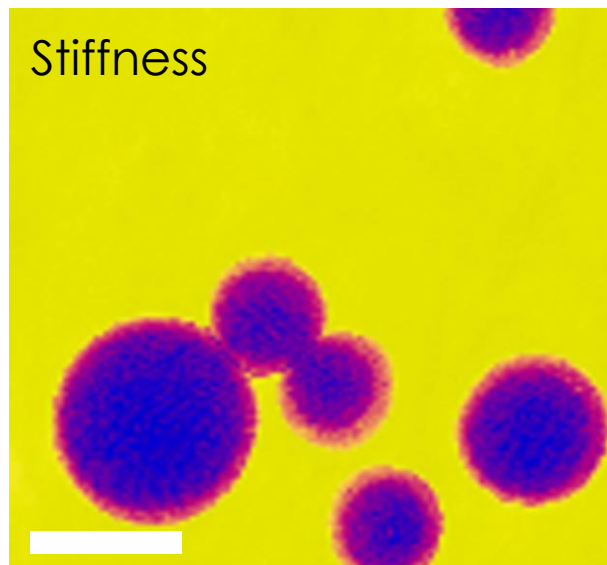
Radius = 125 nm

PeakForce Tapping nanoDMA

128 x 128 pixels,

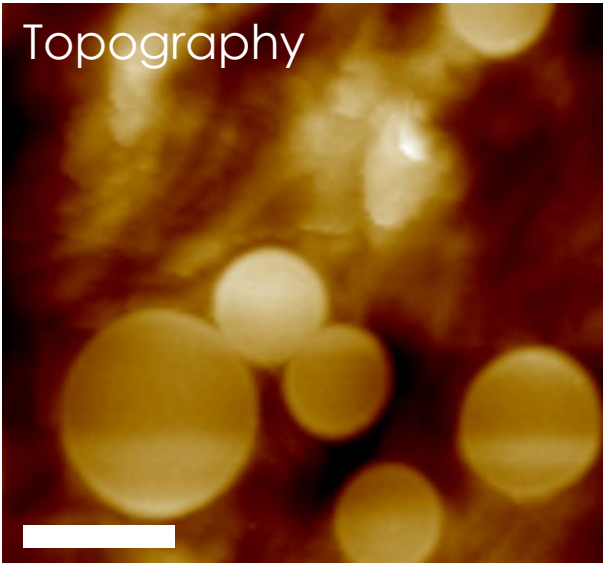
Frequency = 80 Hz

Scale bar = 2.0 μm

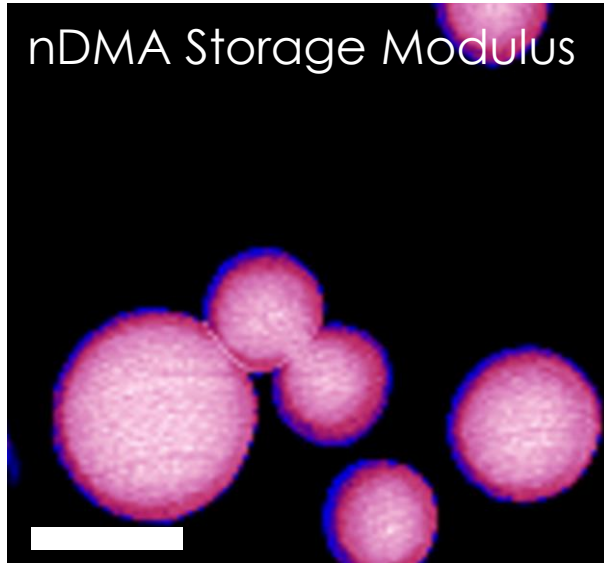


1. Height
2. Adhesion
3. Modulus
4. Log Modulus
5. Stiffness
6. Force Section
7. Storage Modulus
8. Loss Modulus
9. Tan Delta
10. Tan Phase lag
11. Phase lag
12. Storage Stiffness
13. Loss Stiffness
14. Stiffness Mag
15. Contact Radius

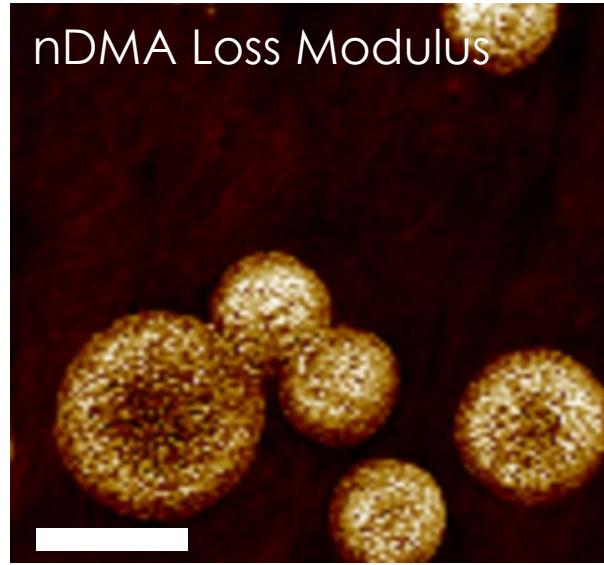
Topography



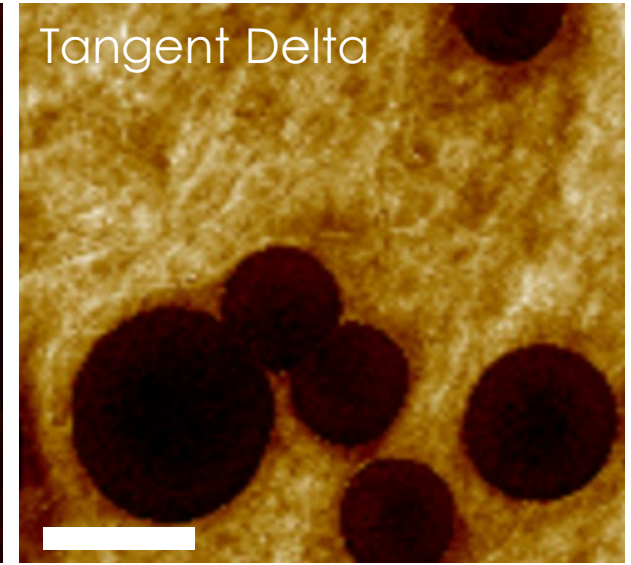
nDMA Storage Modulus



nDMA Loss Modulus



Tangent Delta



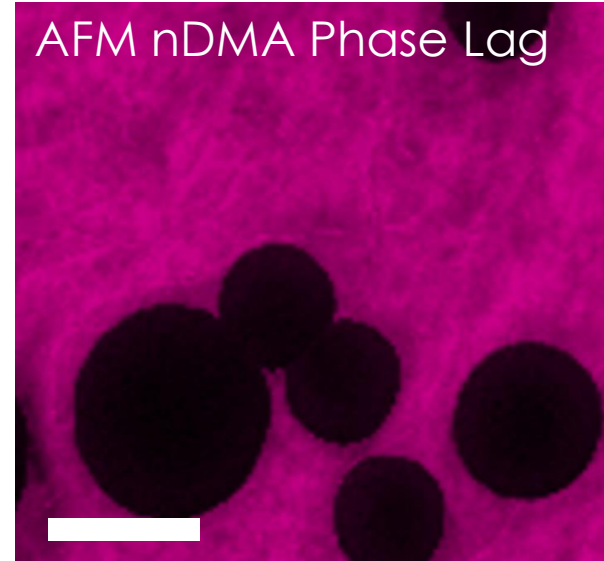
+ Temperature
+ Ramp scripting



AFM nDMA Storage Stiffness



AFM nDMA Phase Lag



AFM nDMA Contact Radius



pyCAROS

Python Code for Approach and Retract force curve analysis of Organic and hybrid Soft materials

Automatised multidimensional analysis based on Machine Learning algorithms for :

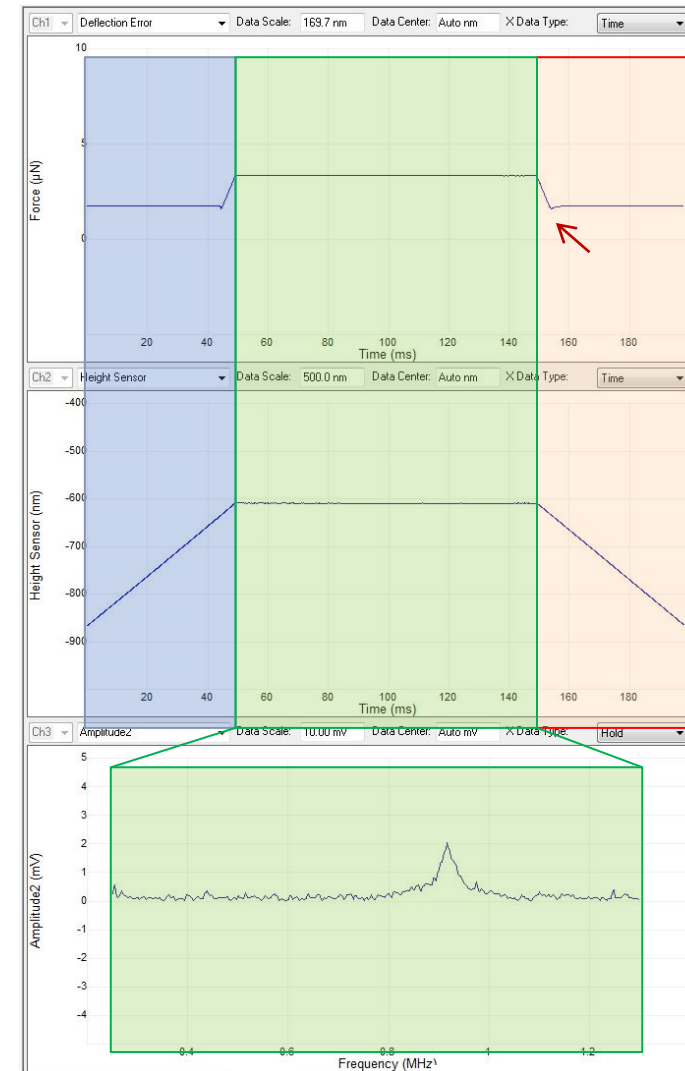
- Clustering of the data (PCA, Kmeans, GMM, ...)
- Force curve analysis + «smart» mapping (Tabor, Contact mechanics models, R^2)
- Force curve quality analysis



Contact Resonance

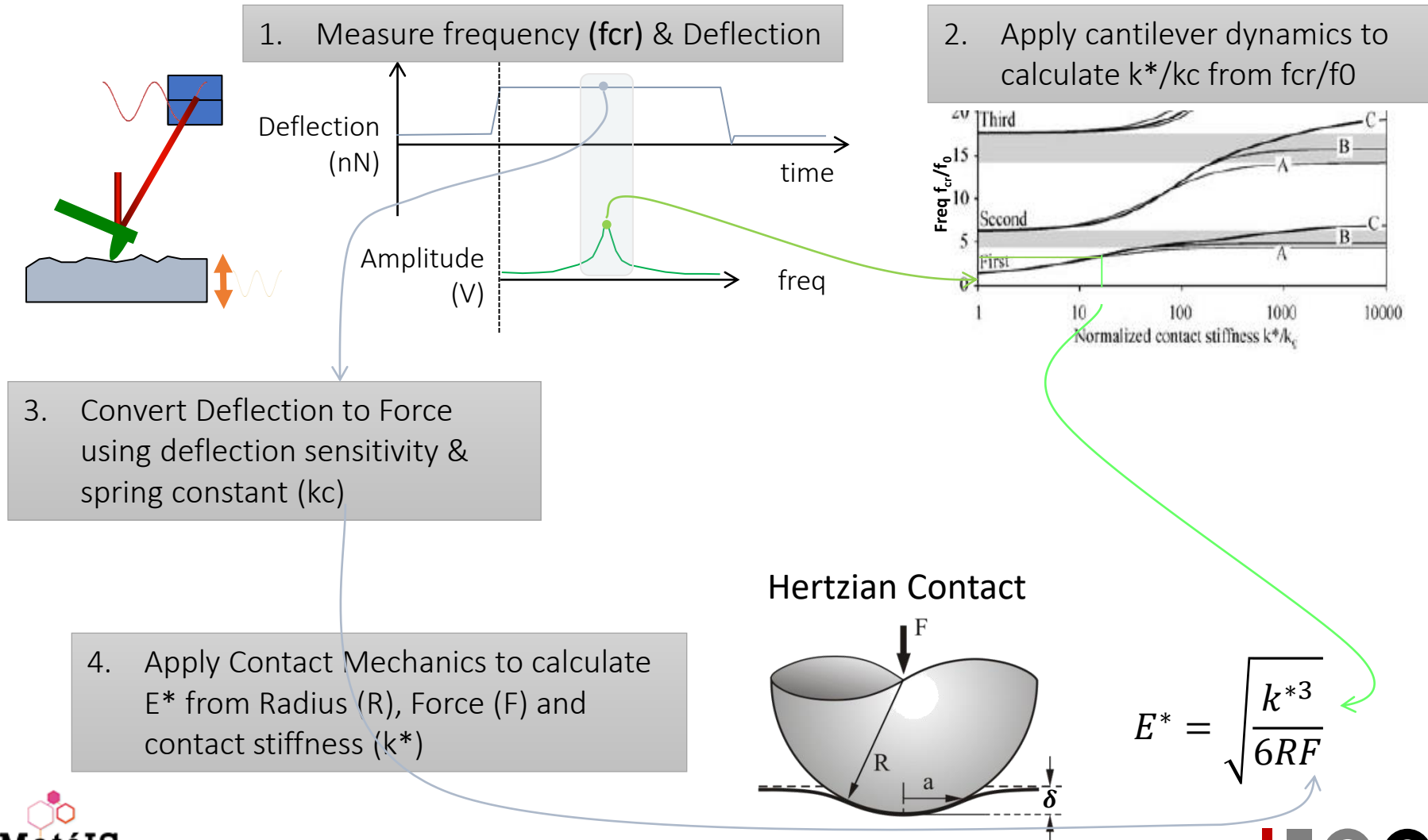
From Frequency and Deflection to modulus

- CR is based on FASTForce Volume
 - Provides standard force curve for comparison for each pixel in map
 - Approach
 - Hold Force and sweep frequency
 - Retract
 - **More repeatable:** lateral force on tip is minimized, reducing tip wear
 - **More information:** allows measurement of adhesion force for each pixel better contact mechanics modeling
 - **Real-time maps** of both raw data and mechanical props (E' , E'' , loss tan)
 - **Whole sweep is saved**, allowing detection of artifact peaks, etc. (unlike frequency tracking methods like DA(F)RT)

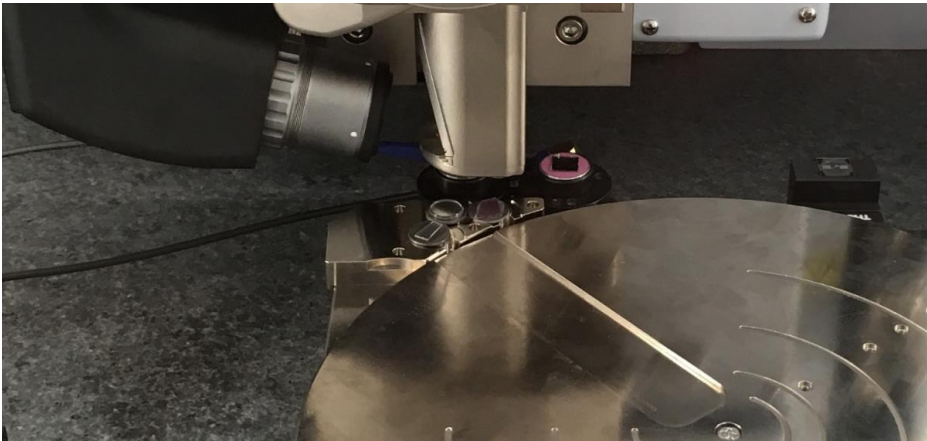


Contact Resonance

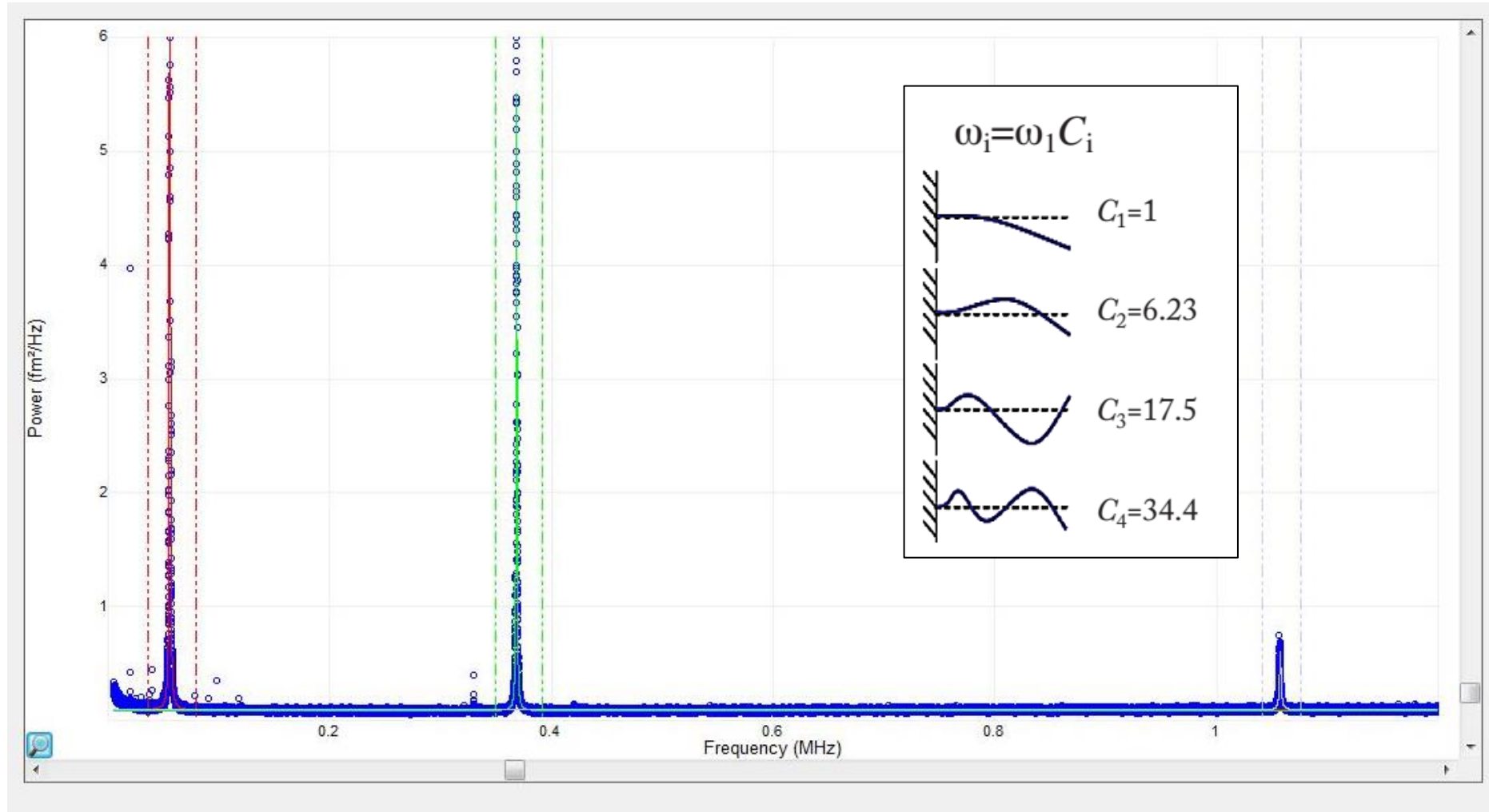
From Frequency and Deflection to modulus



Contact Resonance



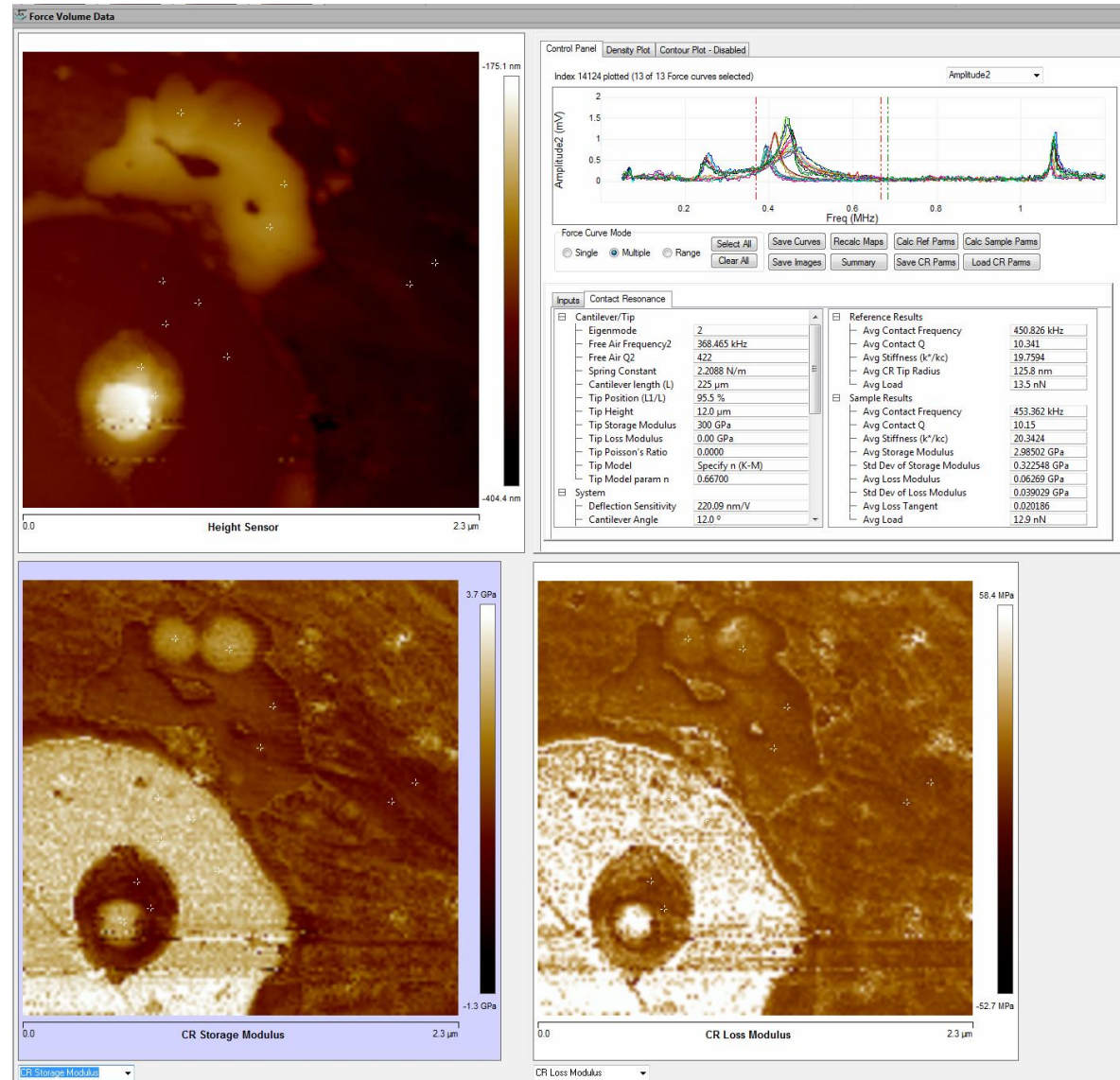
Contact Resonance



Contact Resonance

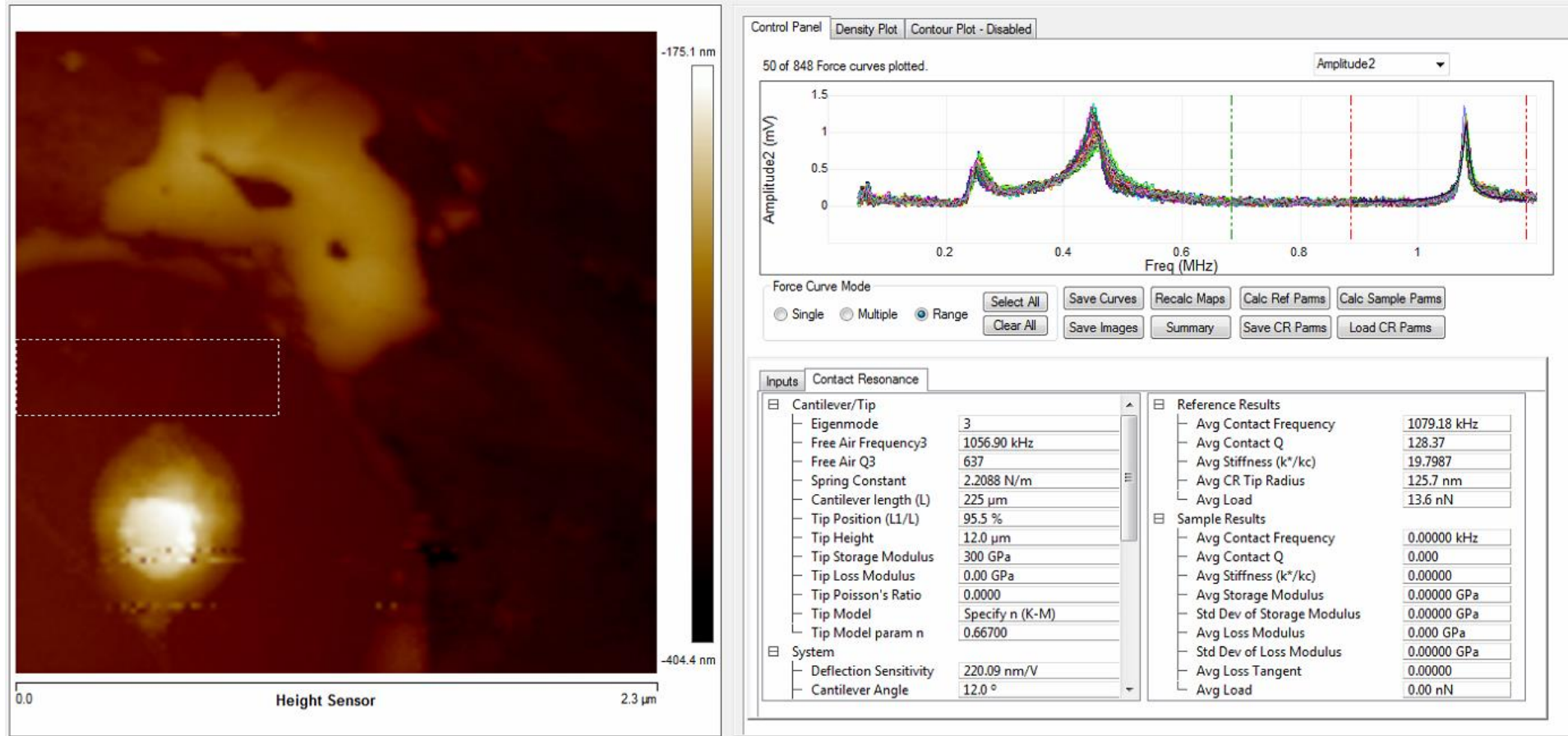
PS-PCL

2nd peak



Contact Resonance

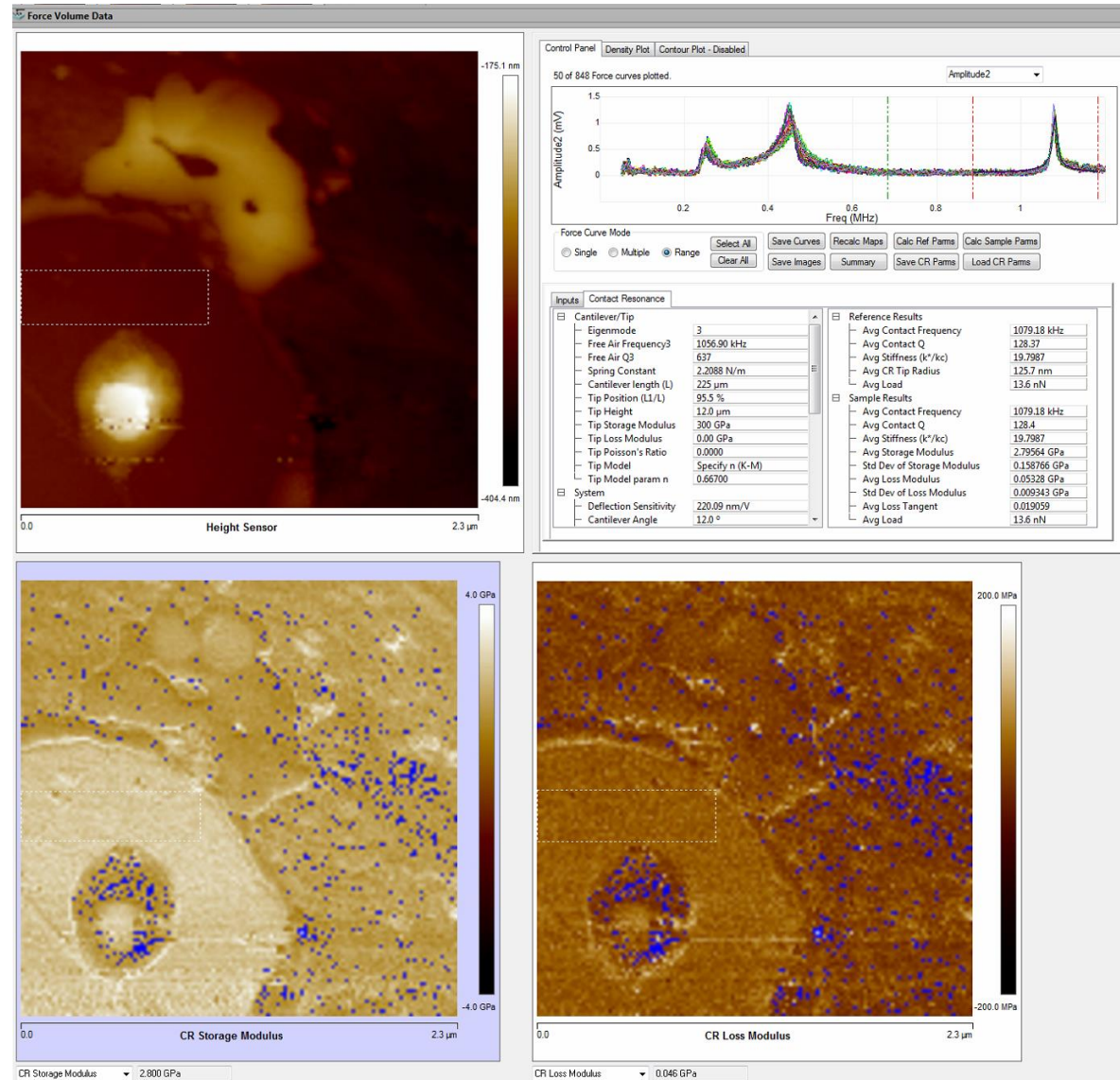
PS-PCL



Contact Resonance

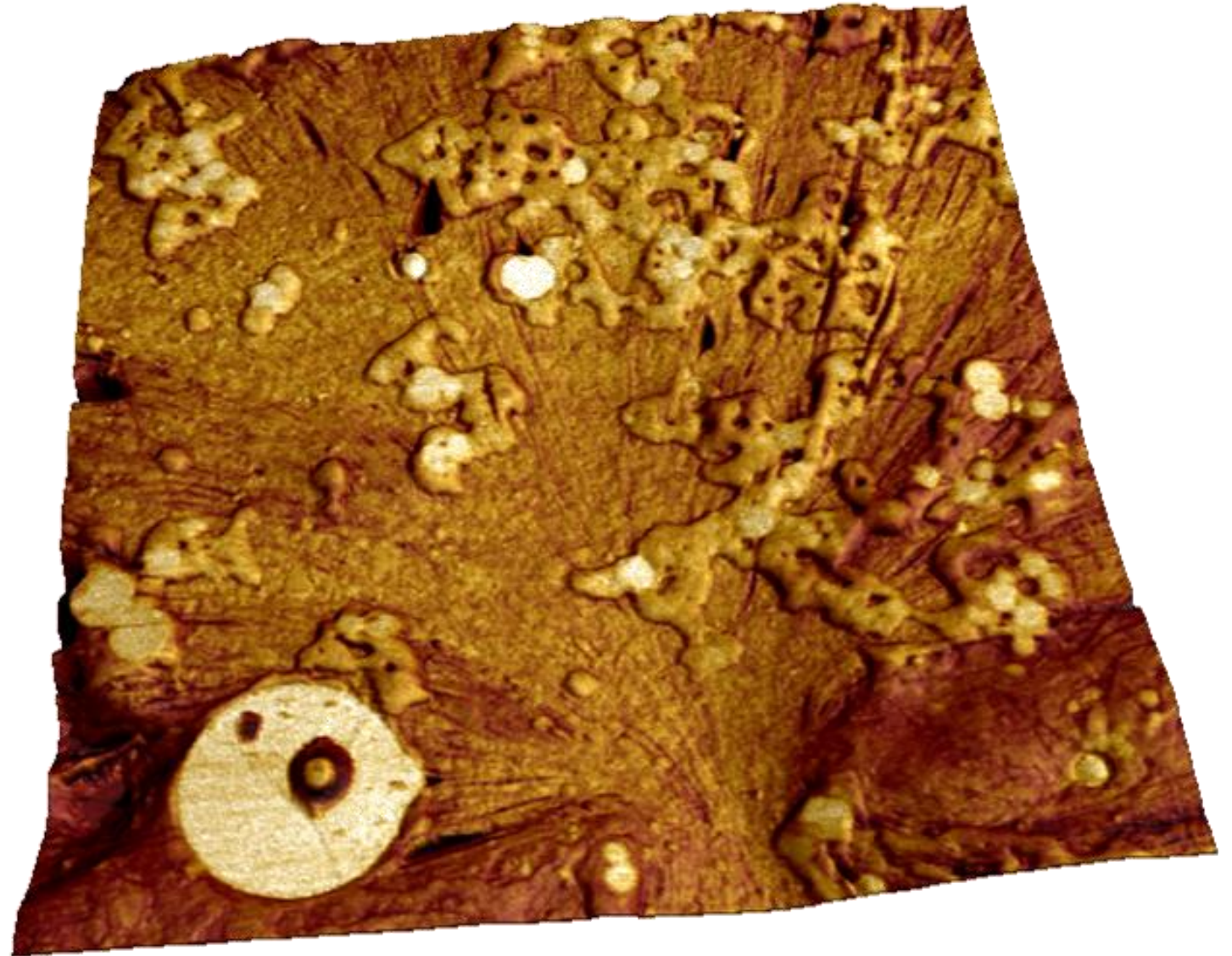
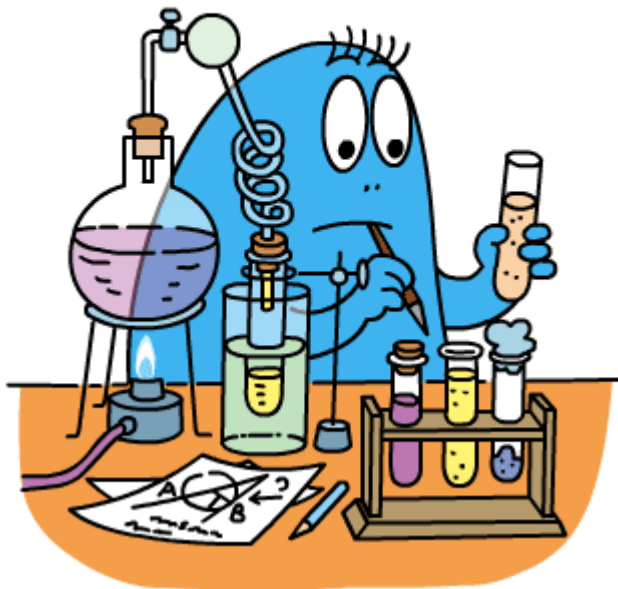
PS-PCL

3rd peak

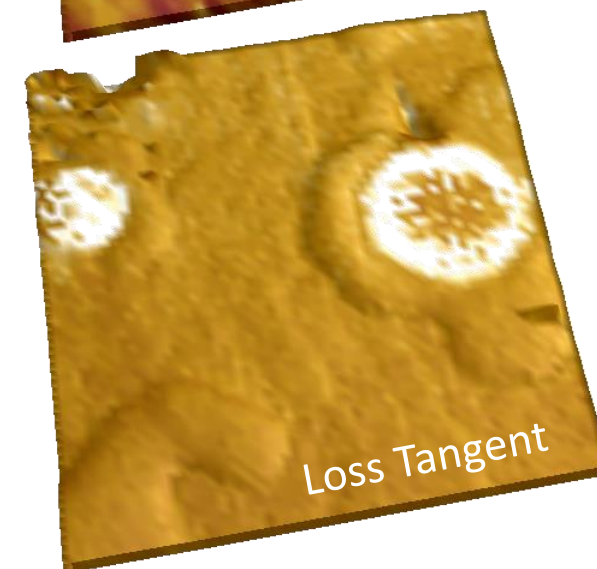
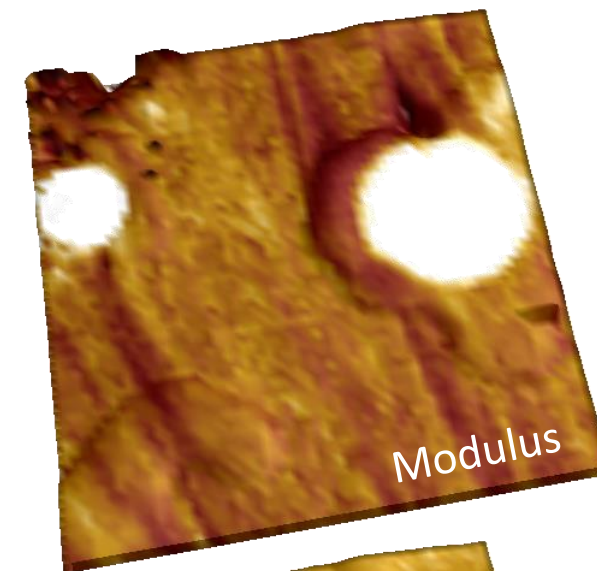
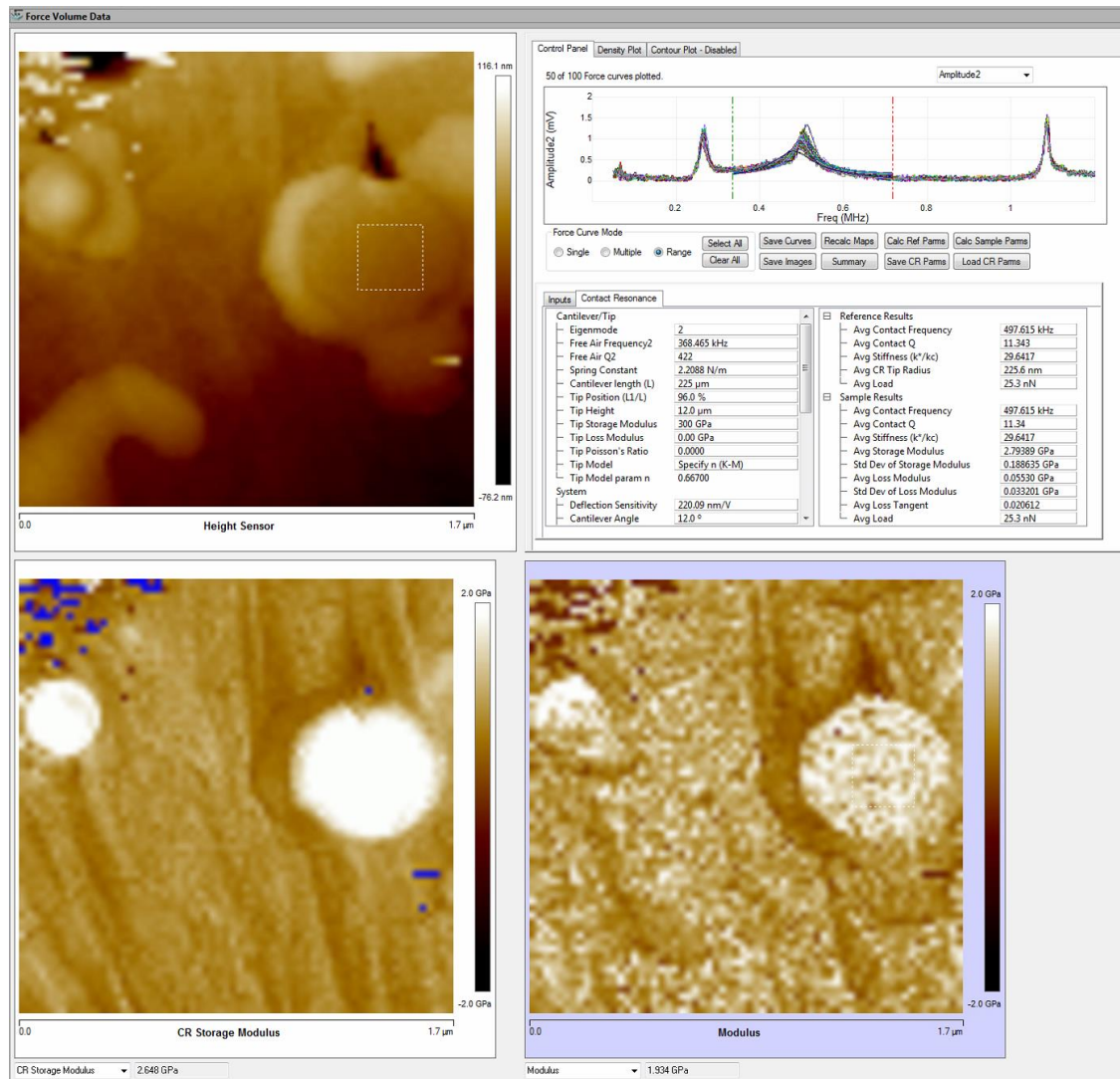


Contact Resonance

PS-PCL

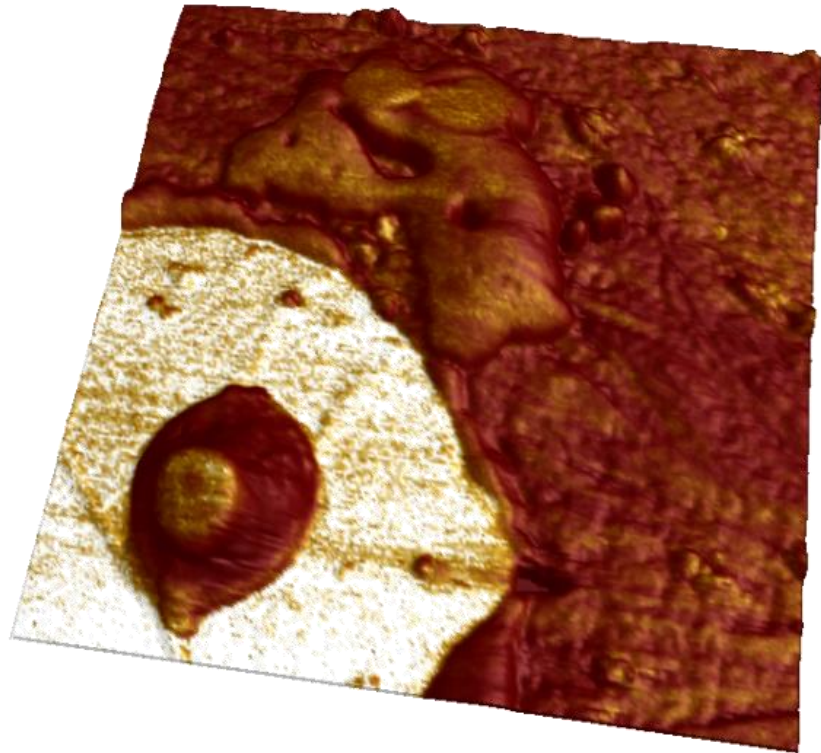


Contact Resonance

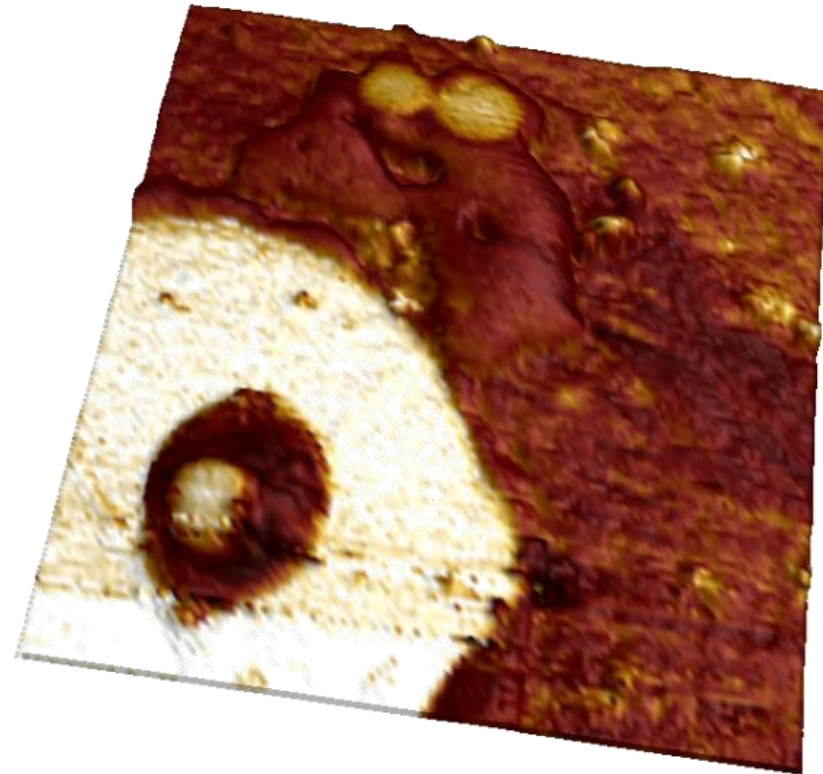


Comparing PFTQNM and CR

- PFTQNM

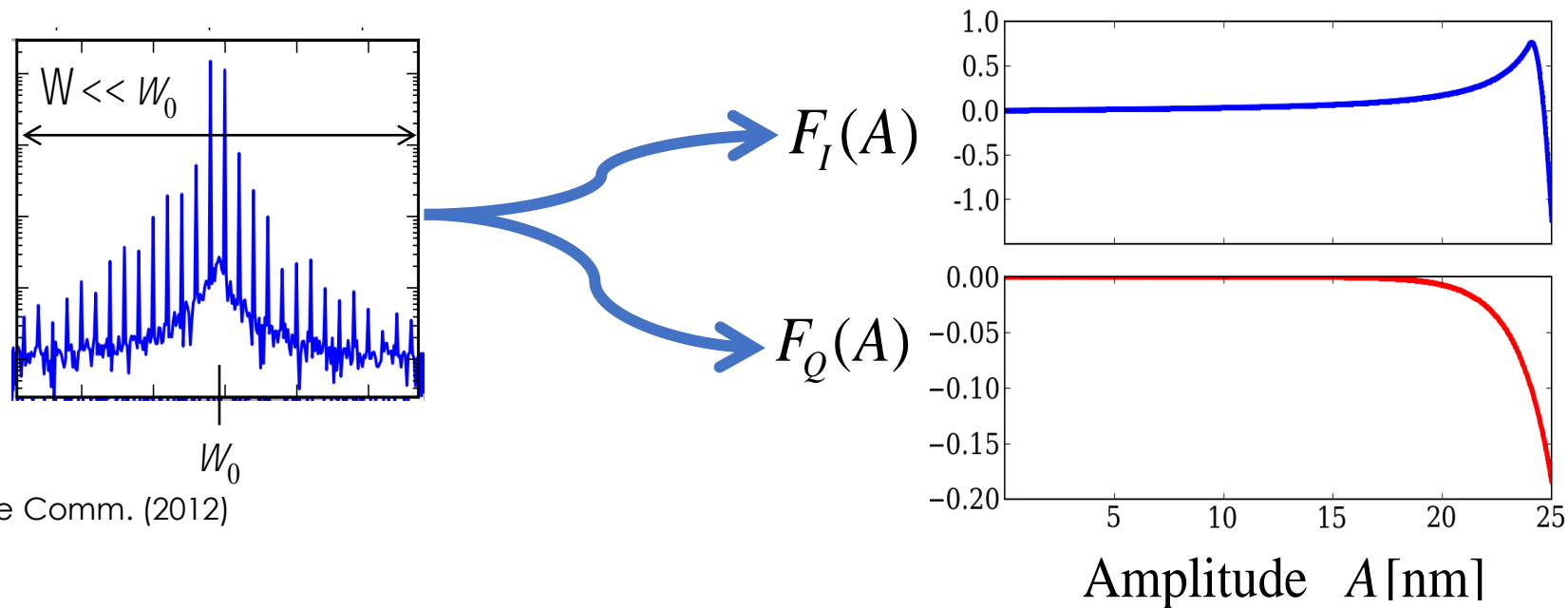


- CR 128x128



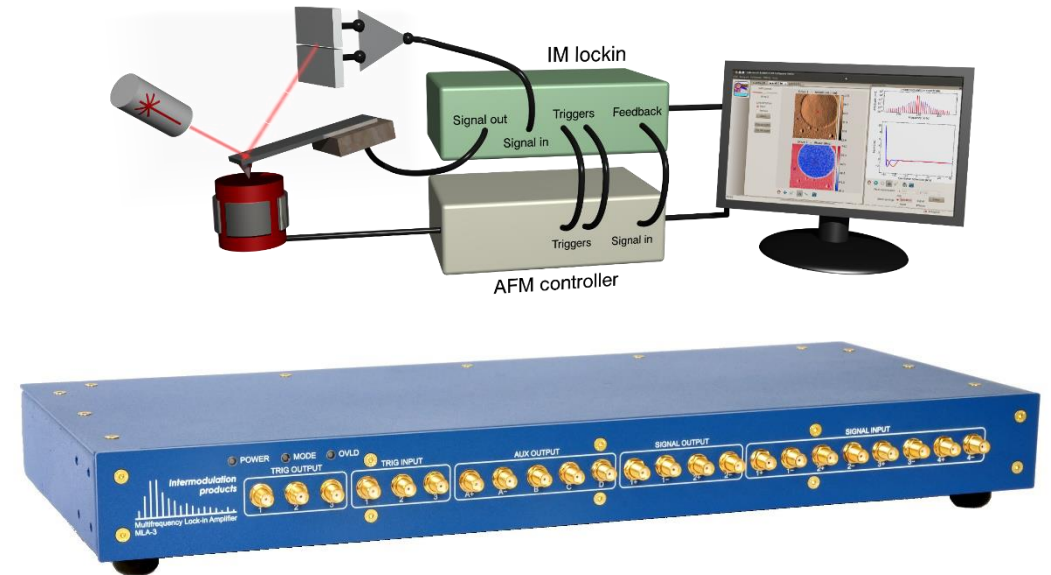
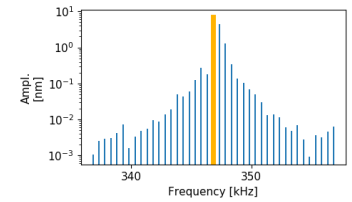
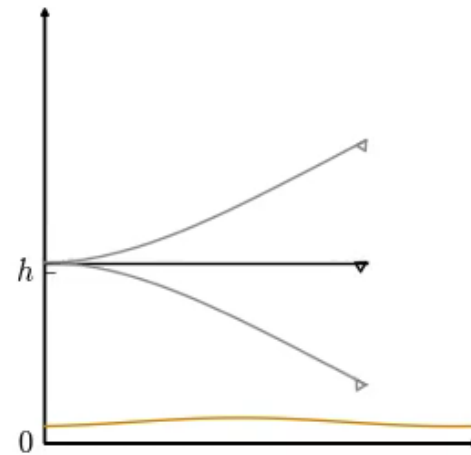
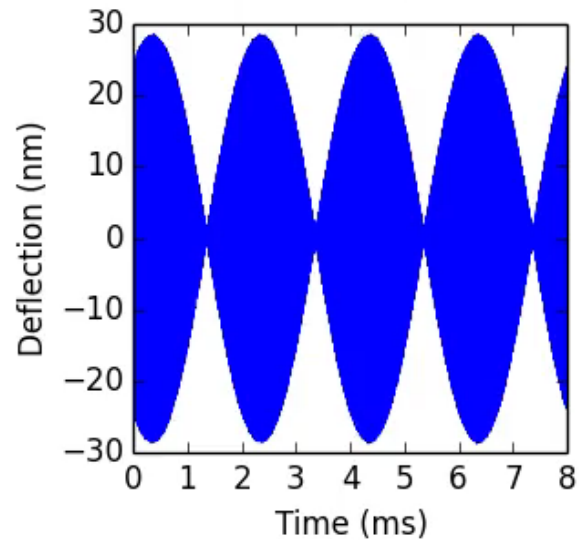
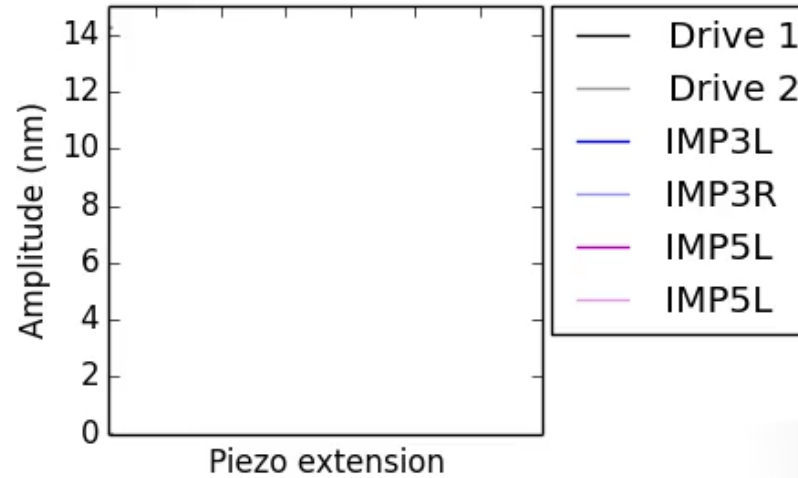
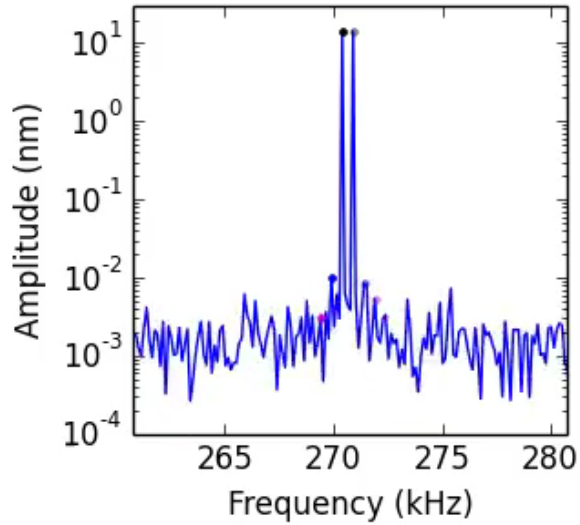
Intermodulation AFM

The methodology

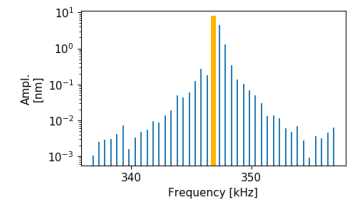
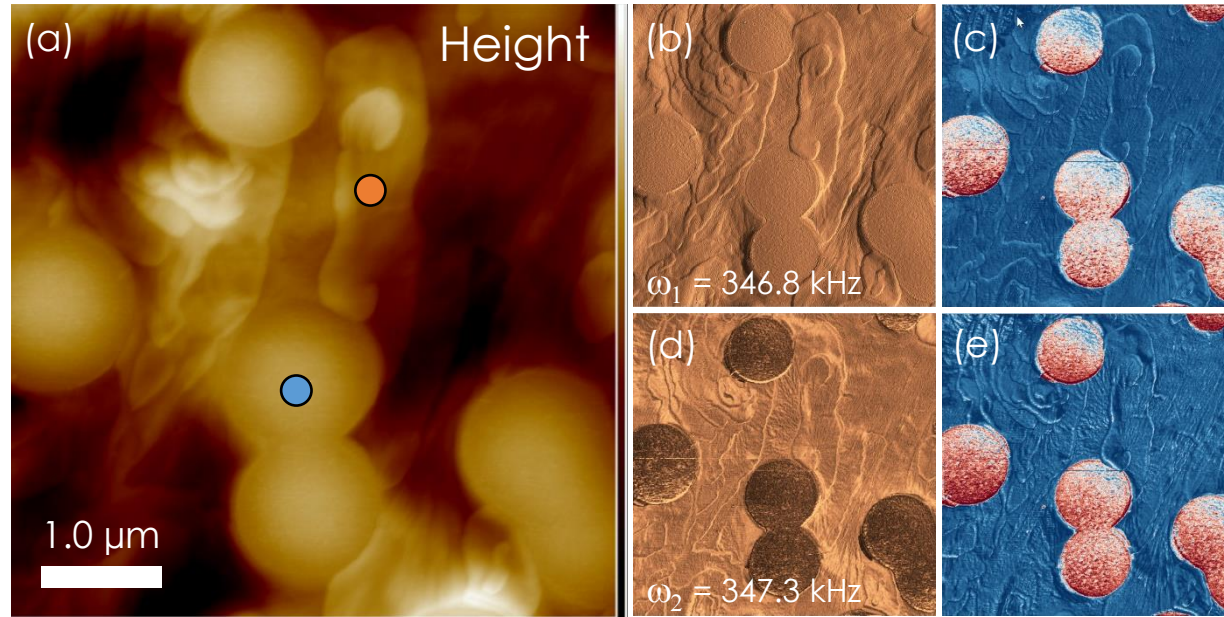


D. Platz et al. Nature Comm. (2012)

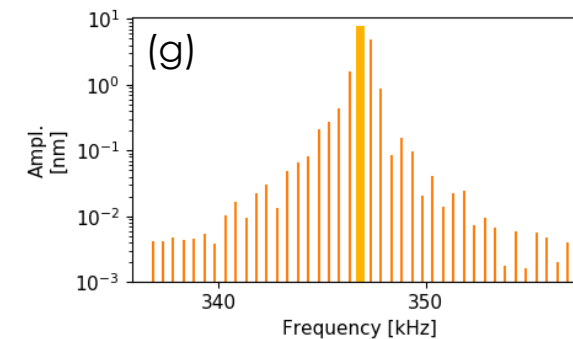
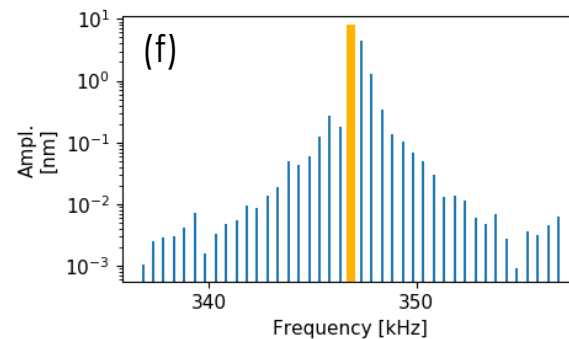
Intermodulation AFM



Blend of PS/PCL



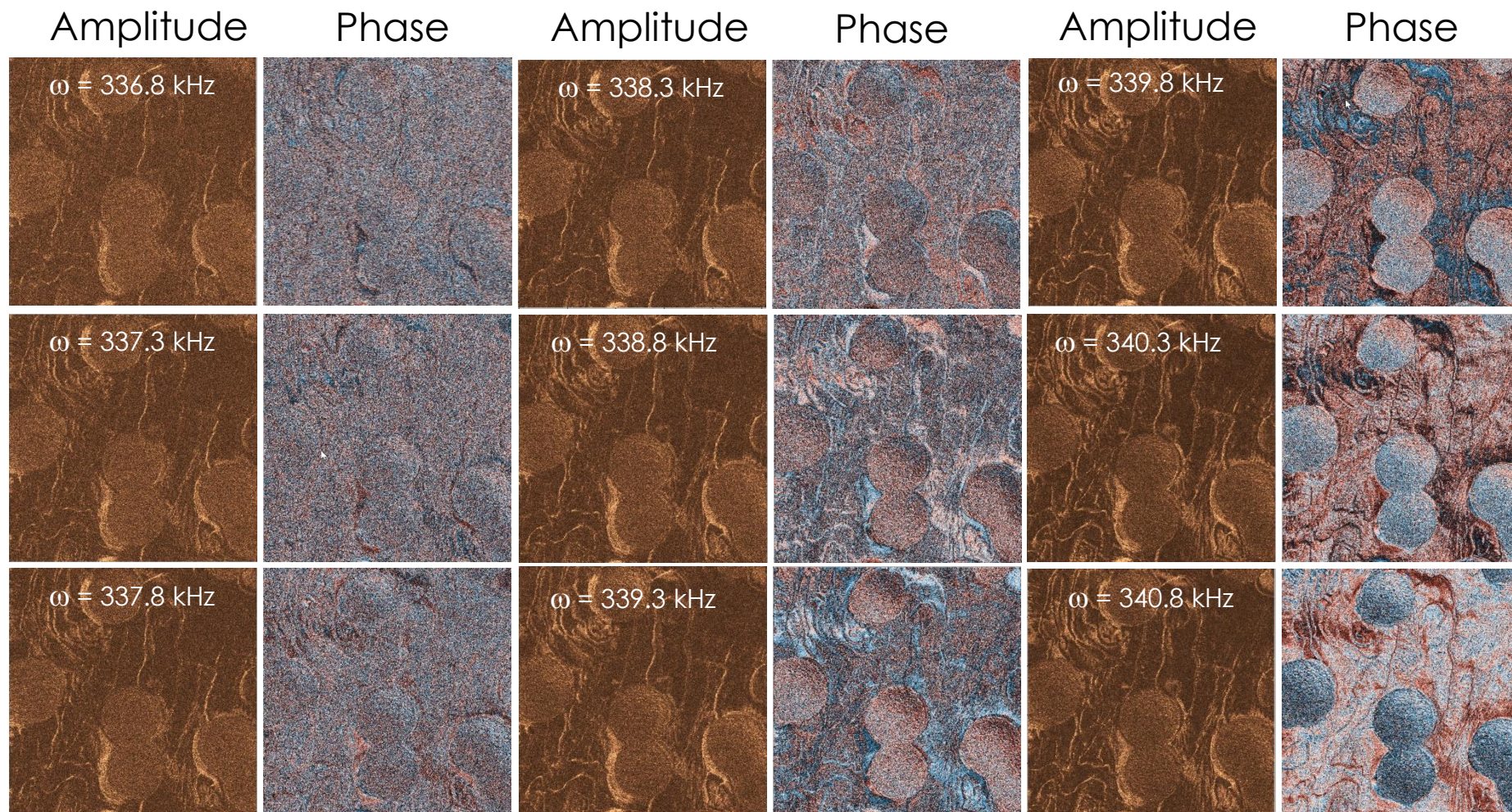
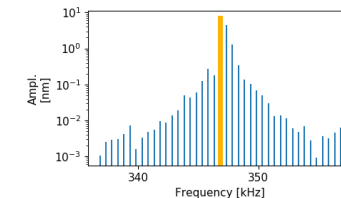
+ 40 others !



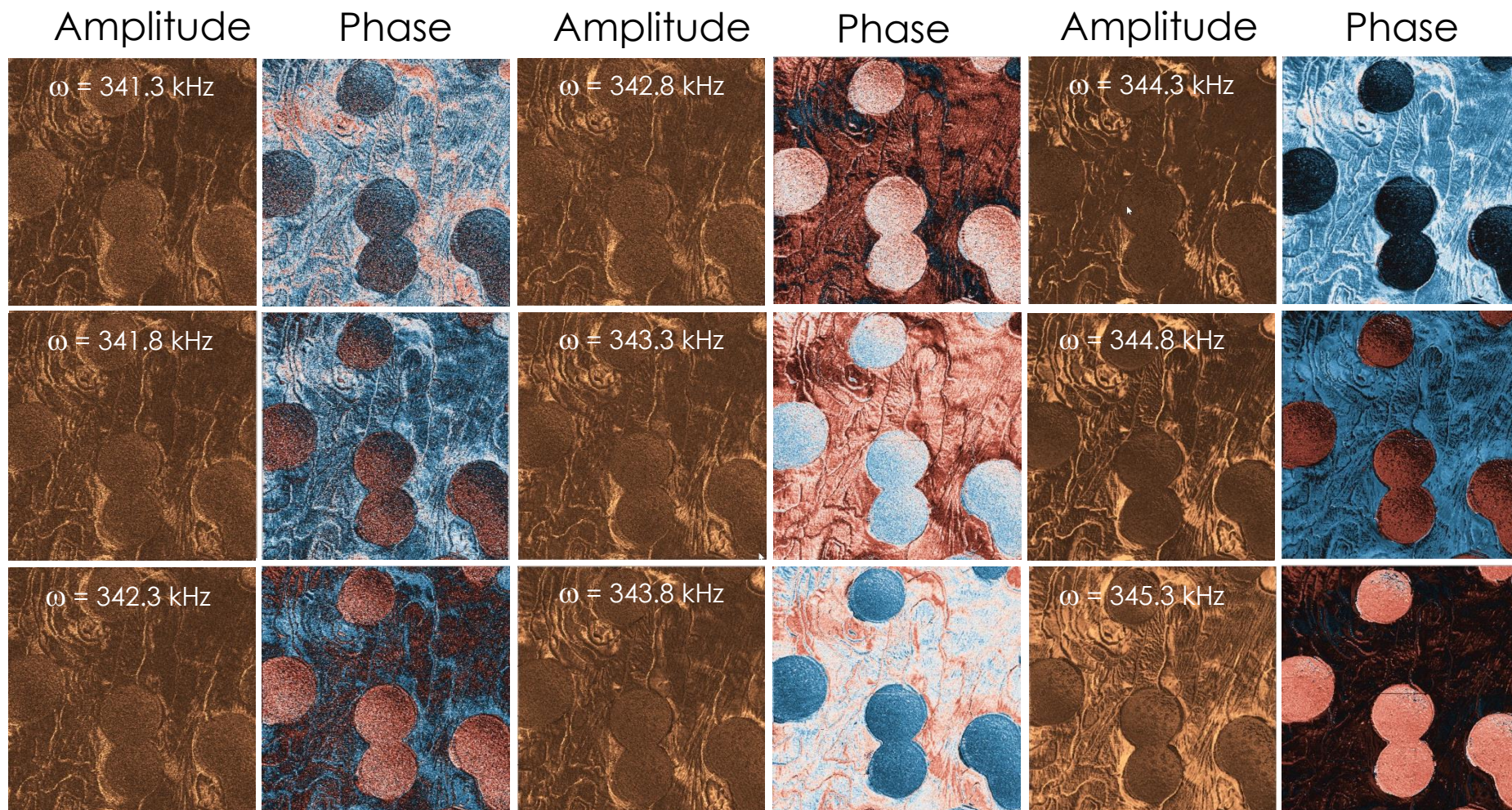
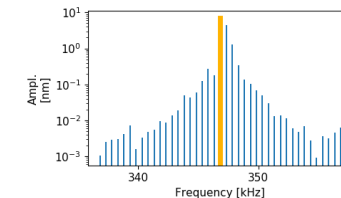
Soft Matter (2016), 12, 619.

PS:PCL 30:70 blend

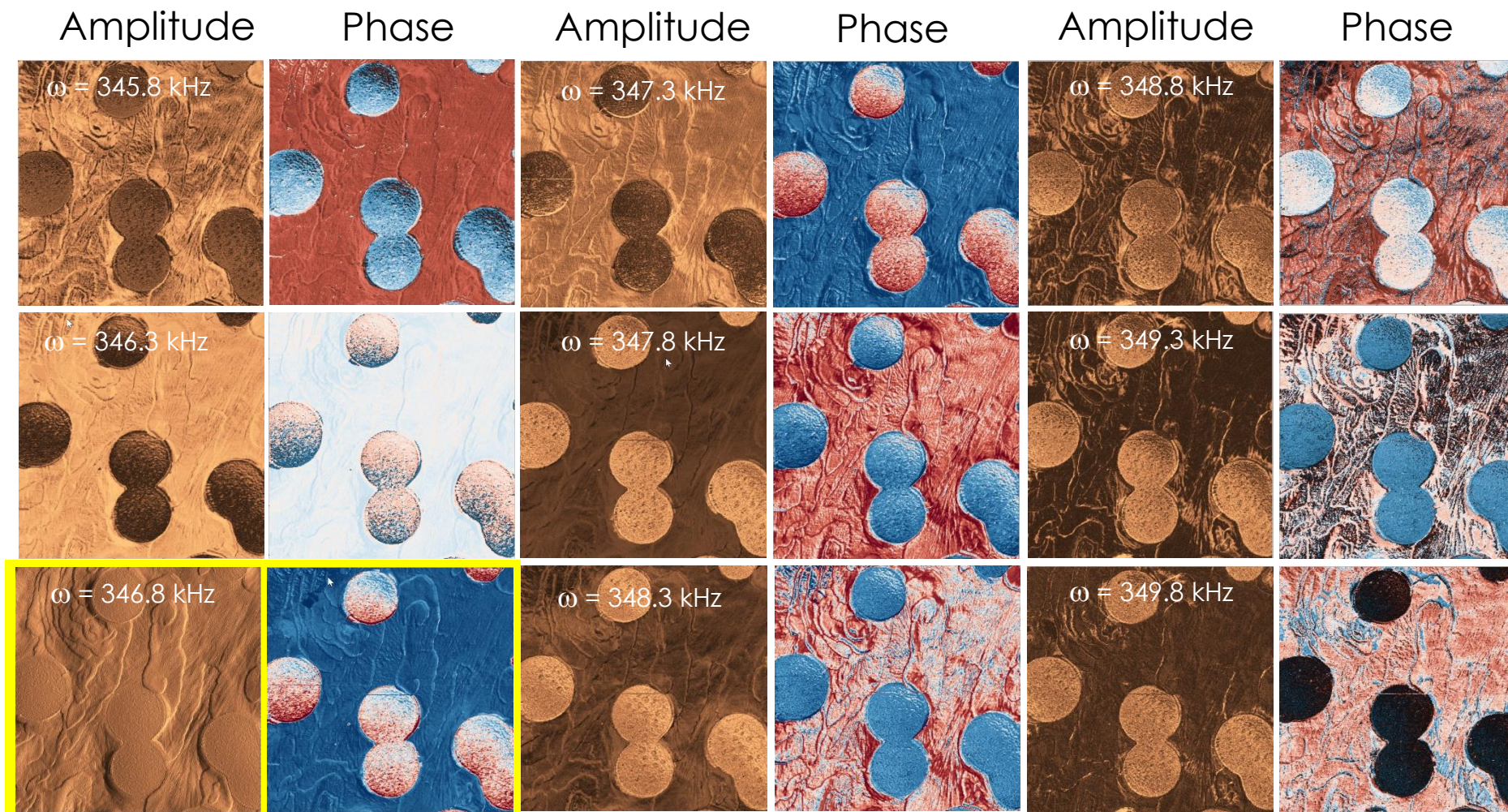
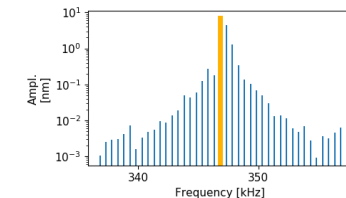
Blend of PS/PCL



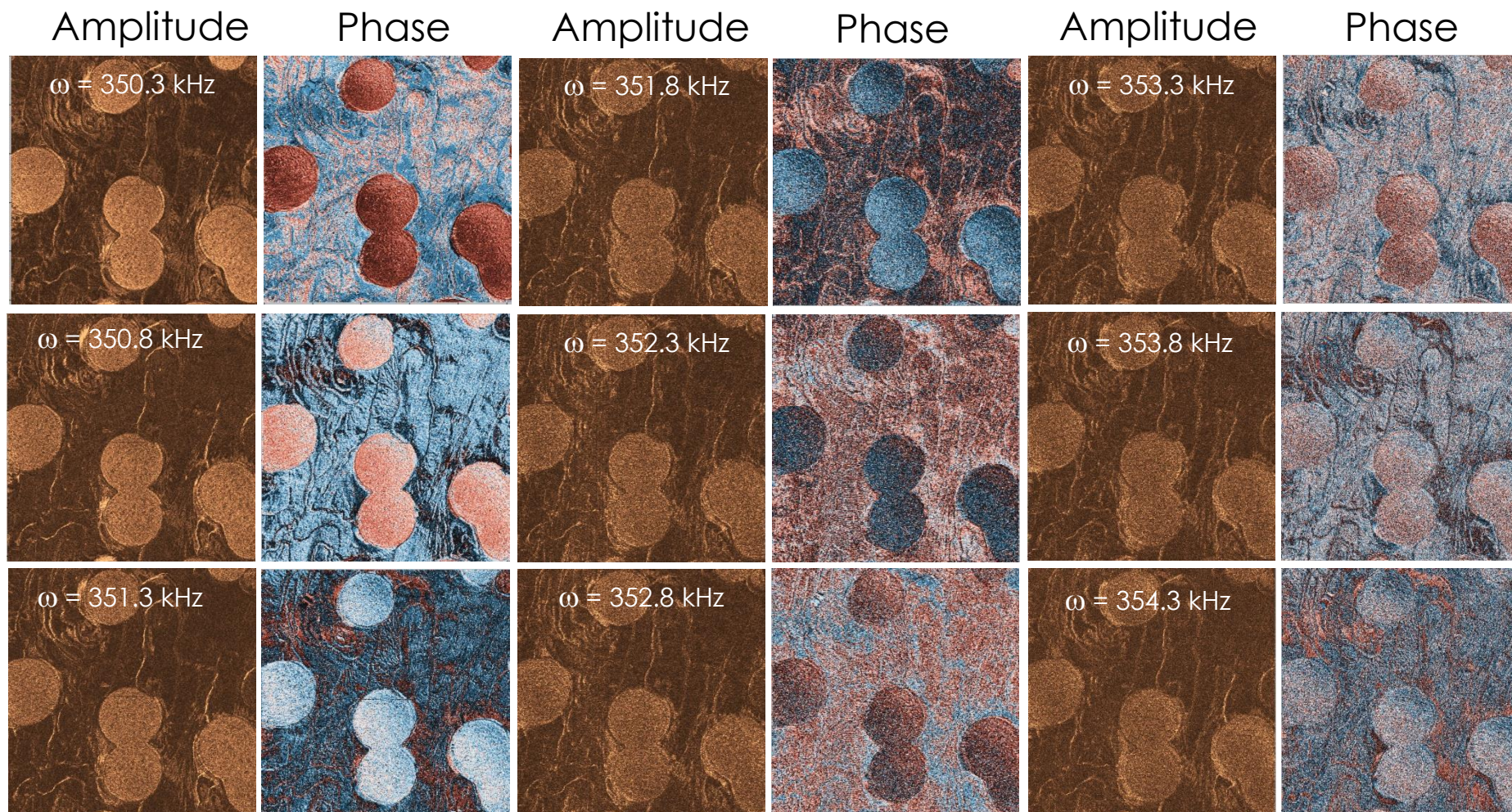
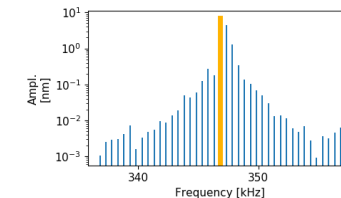
Blend of PS/PCL



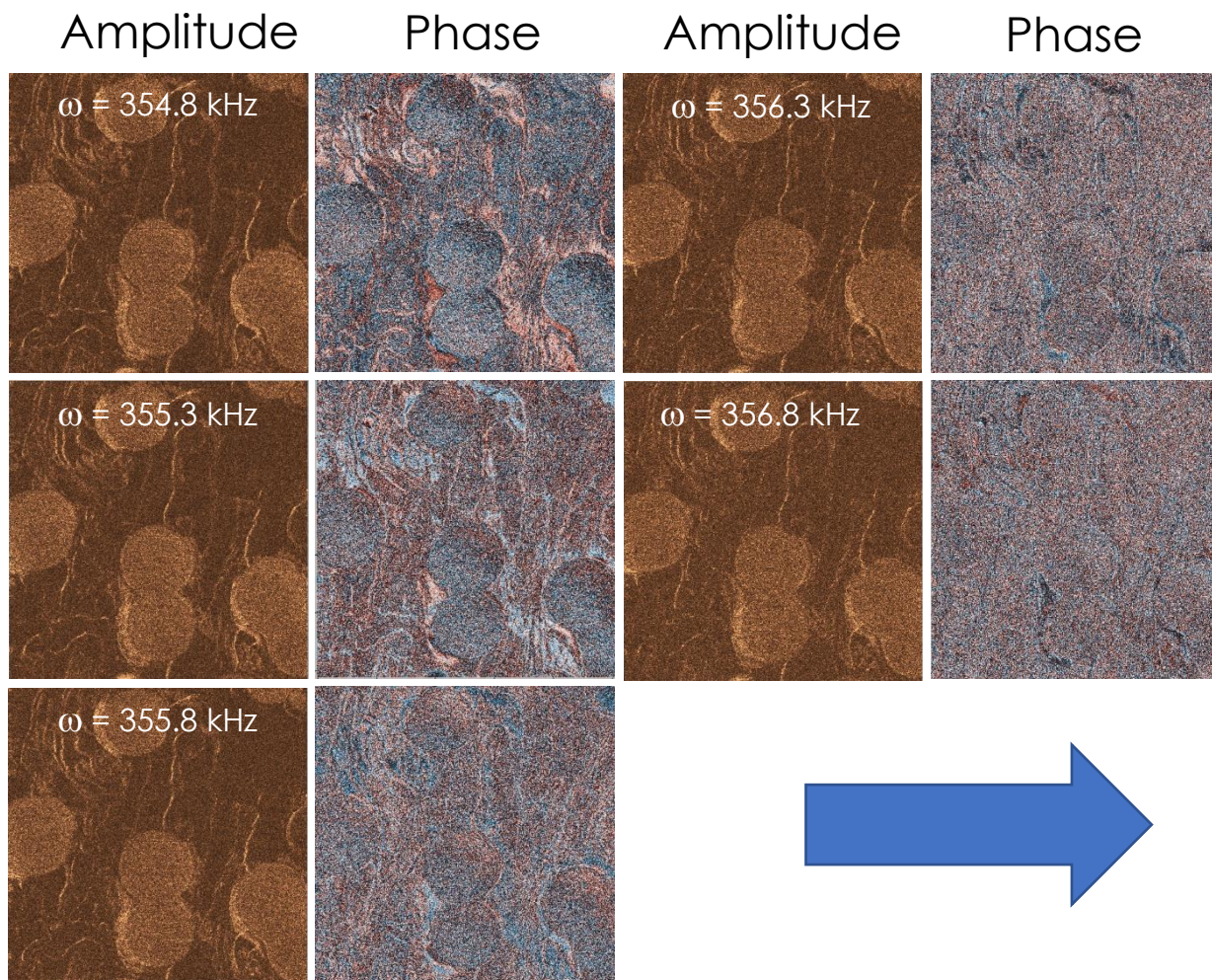
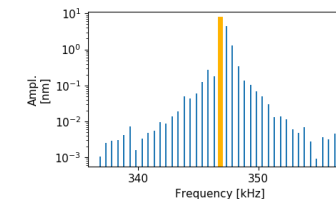
Blend of PS/PCL



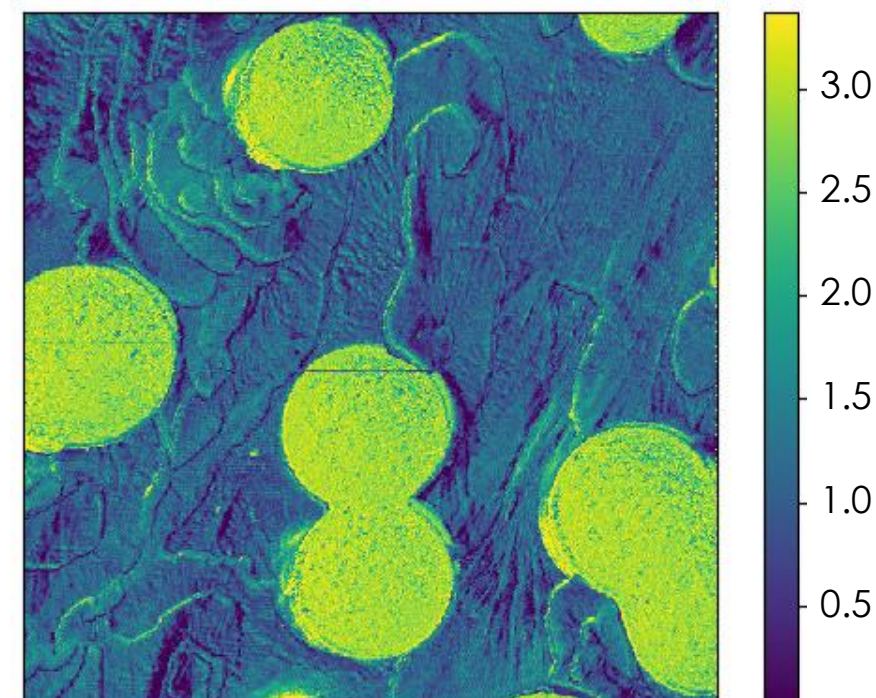
Blend of PS/PCL

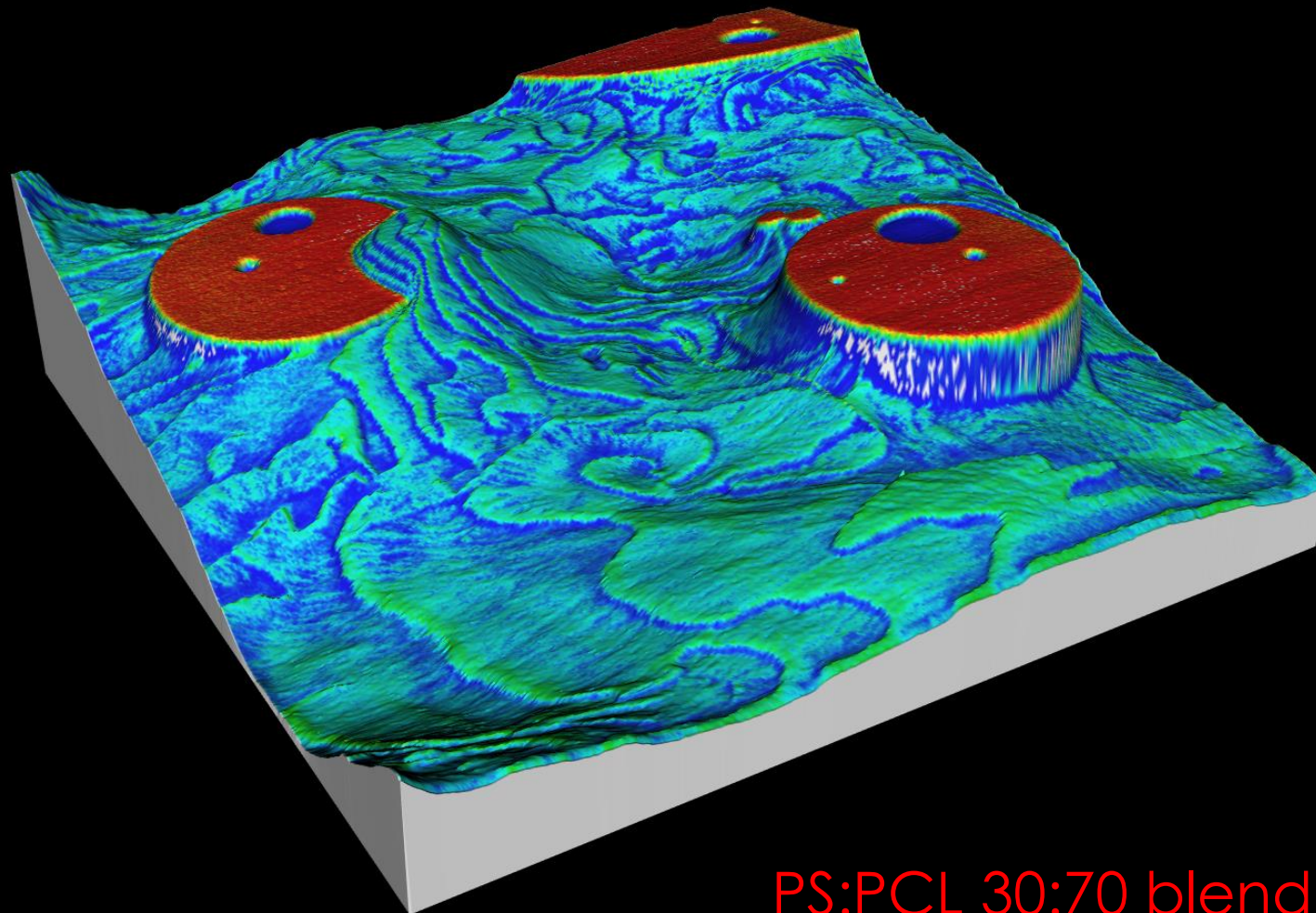


Blend of PS/PCL



Contact Modulus (GPa)





E^* (GPa)

2.70

0.00

PS:PCL 30:70 blend

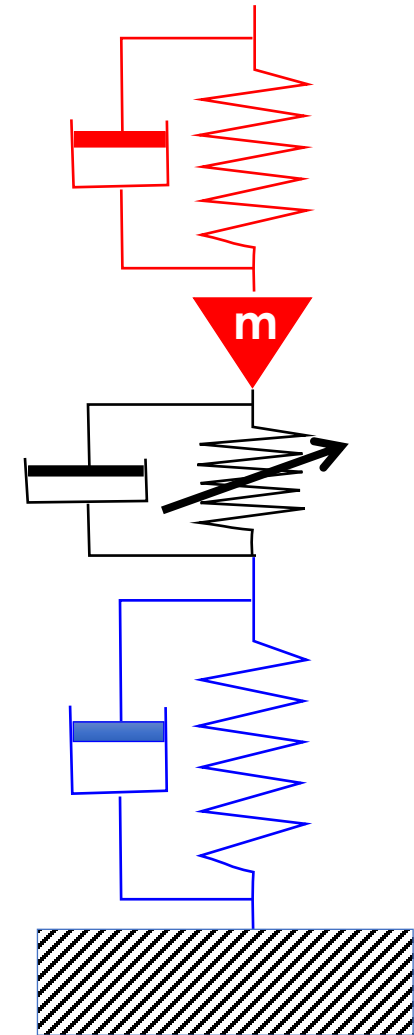
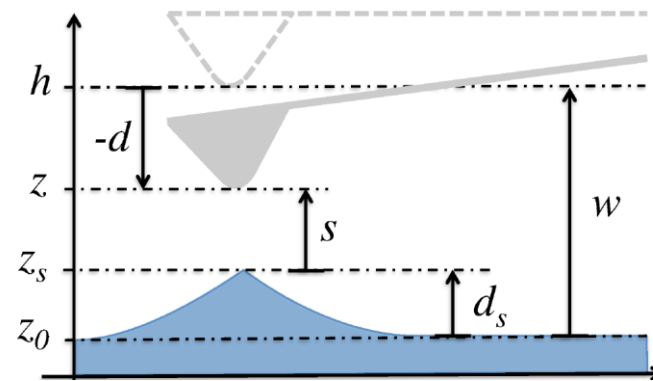
Moving Surface Model

$$m\ddot{d} + \gamma m\dot{d} + kd = F_{TS}(s, \dot{s}) + F_{\text{drive}}$$

$$\eta \dot{d}_s + k_s d_s = -F_{TS}(s, \dot{s})$$

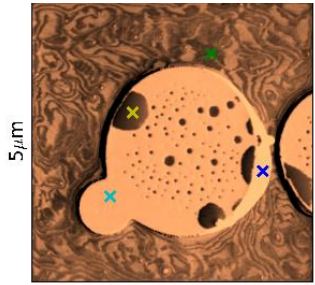
$$\text{separation } s = (d - d_s) + h - z_0$$

$$F_{TS}(s, \dot{s}) = \begin{cases} 0 & \text{if } s > 0 \\ -F_{\text{ad}} - k_v s - \eta_v \dot{s} & \text{if } s \leq 0 \end{cases}$$

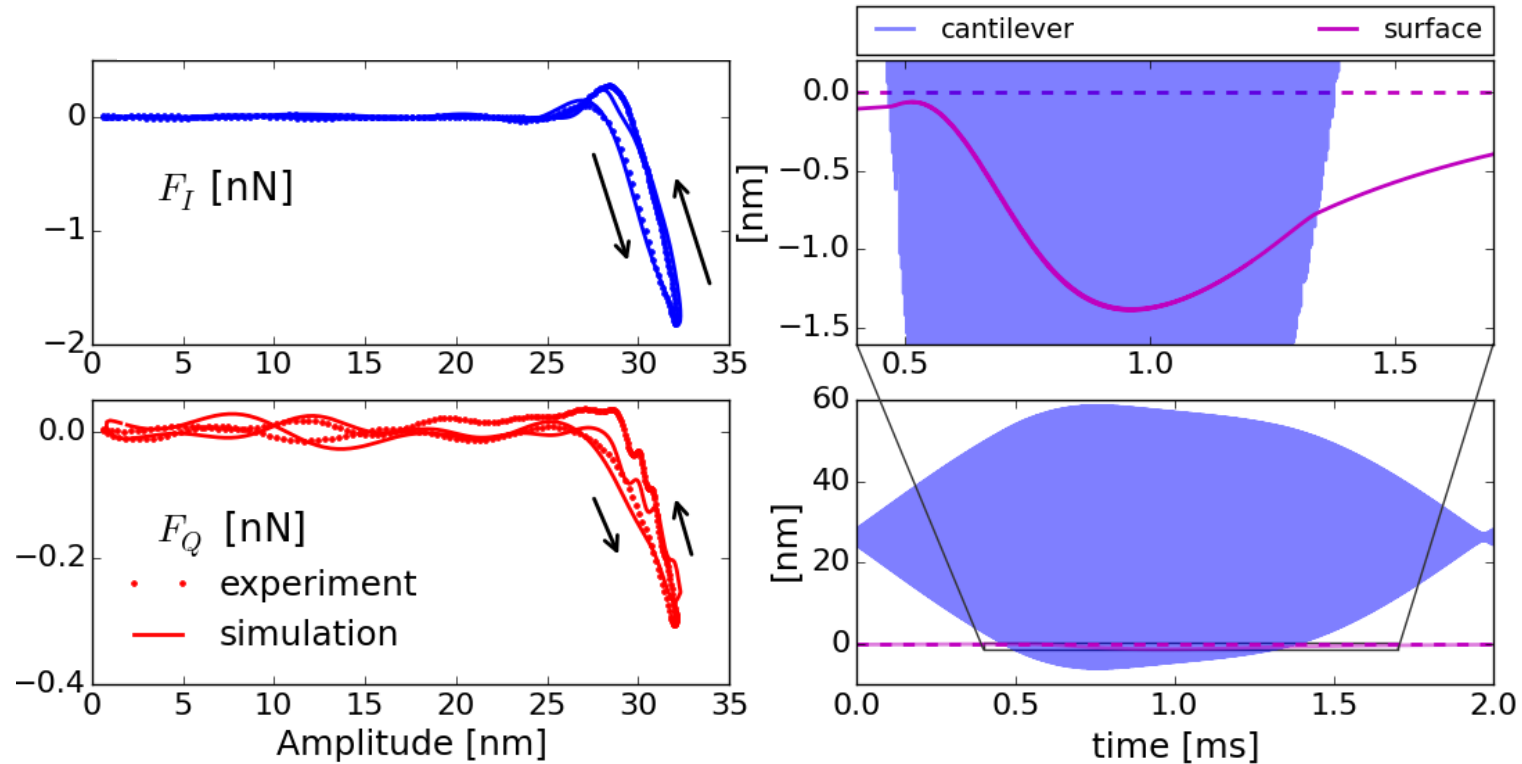


Soft Matter (2016), 12, 619.

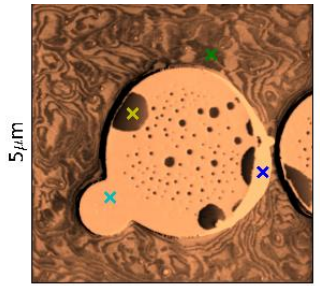
Fit to moving surface model



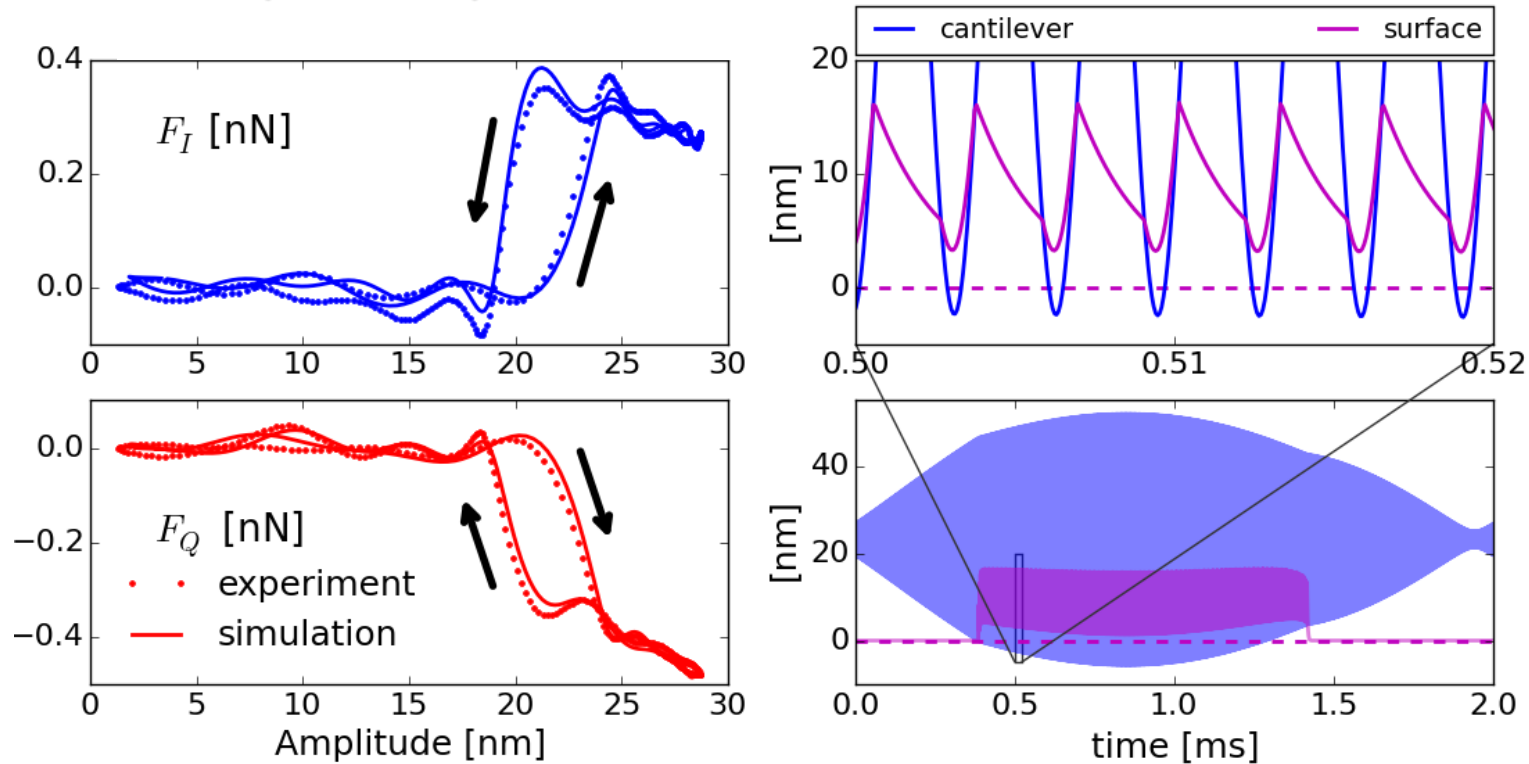
PS phase



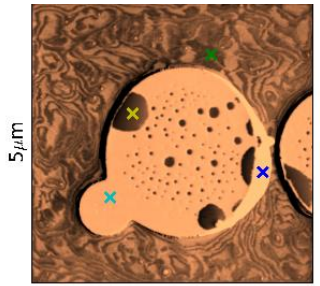
Fit to moving surface model



PCL amorphous phase



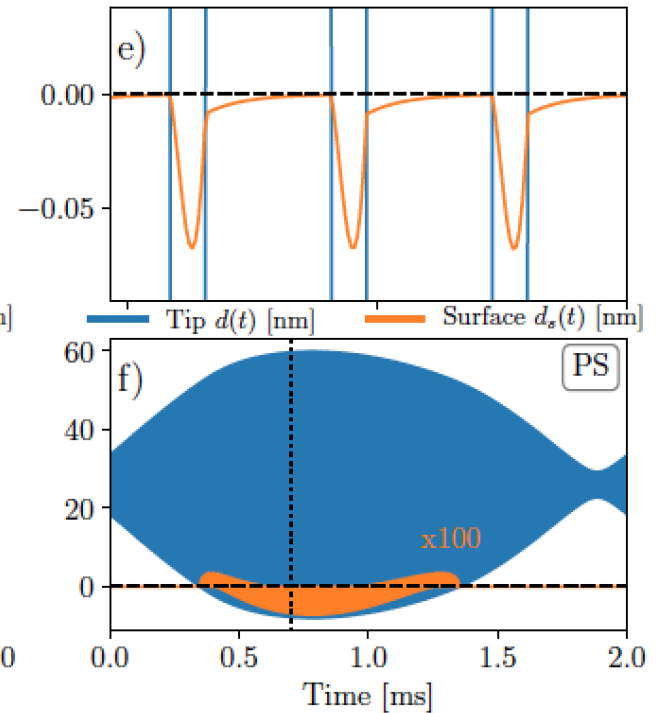
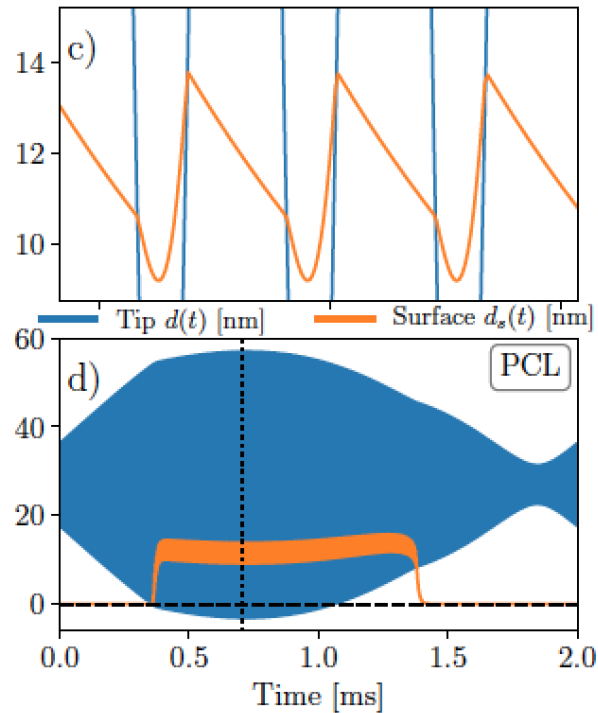
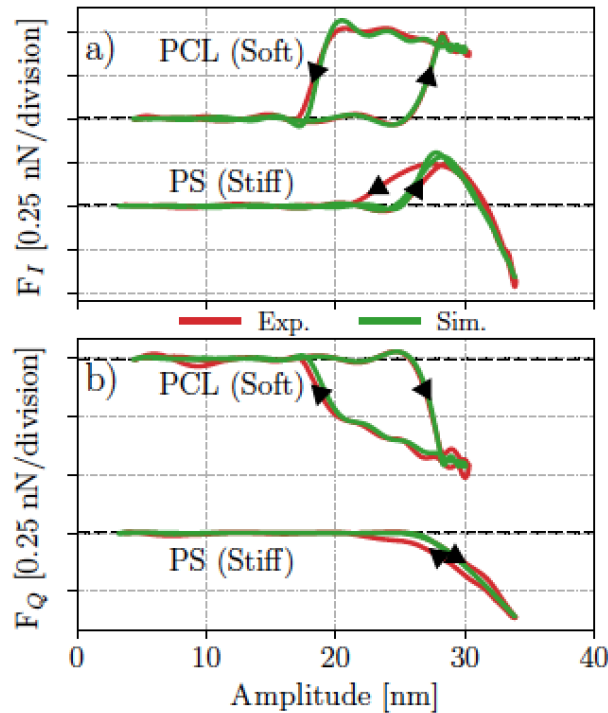
Fit to moving surface model



Soft Matter (2016), 12, 619.

Sample: PS-PCL

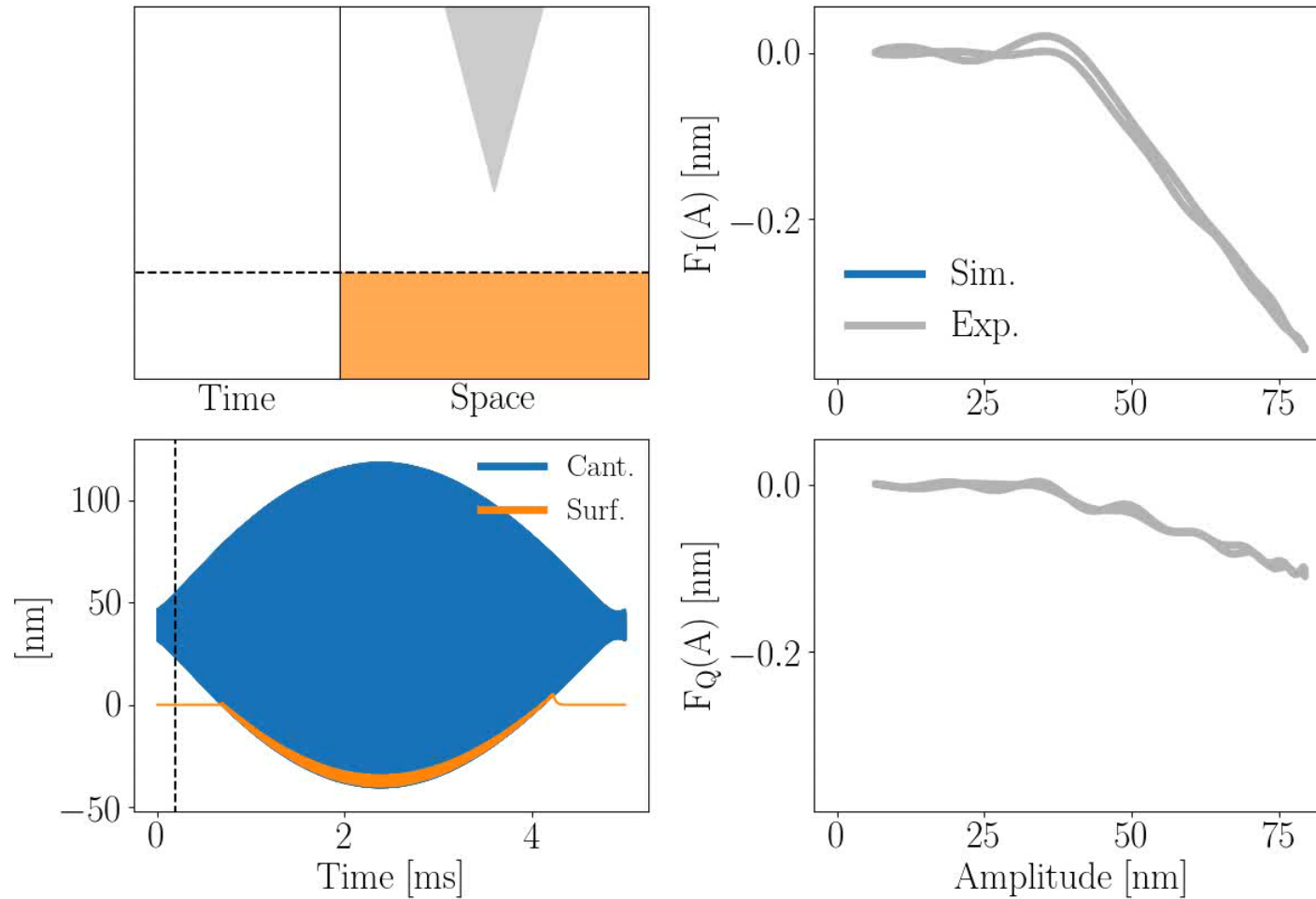
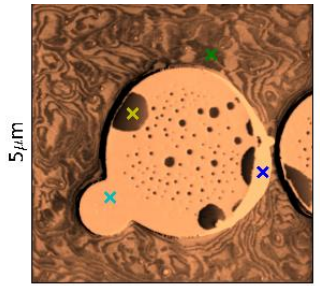
Cantilever: Tap300 ($f_0=310.6$ kHz, $k=26.03$ N/m, $Q=465.1$)



	h [nm]	τ_s [ns]	τ_v [ns]	k_s [N/m]	k_v [N/m]	F_{ad} [nN]	$K=k_s/k_v$	$R=\tau_s/\tau_v$
PCL	26.95	8118	580.3	0.0469	0.1557	2.859	0.3012	13.99
PS	25.98	735.2	152.8	32.53	1.097	3.88	29.65	4.811

Fit to moving surface model

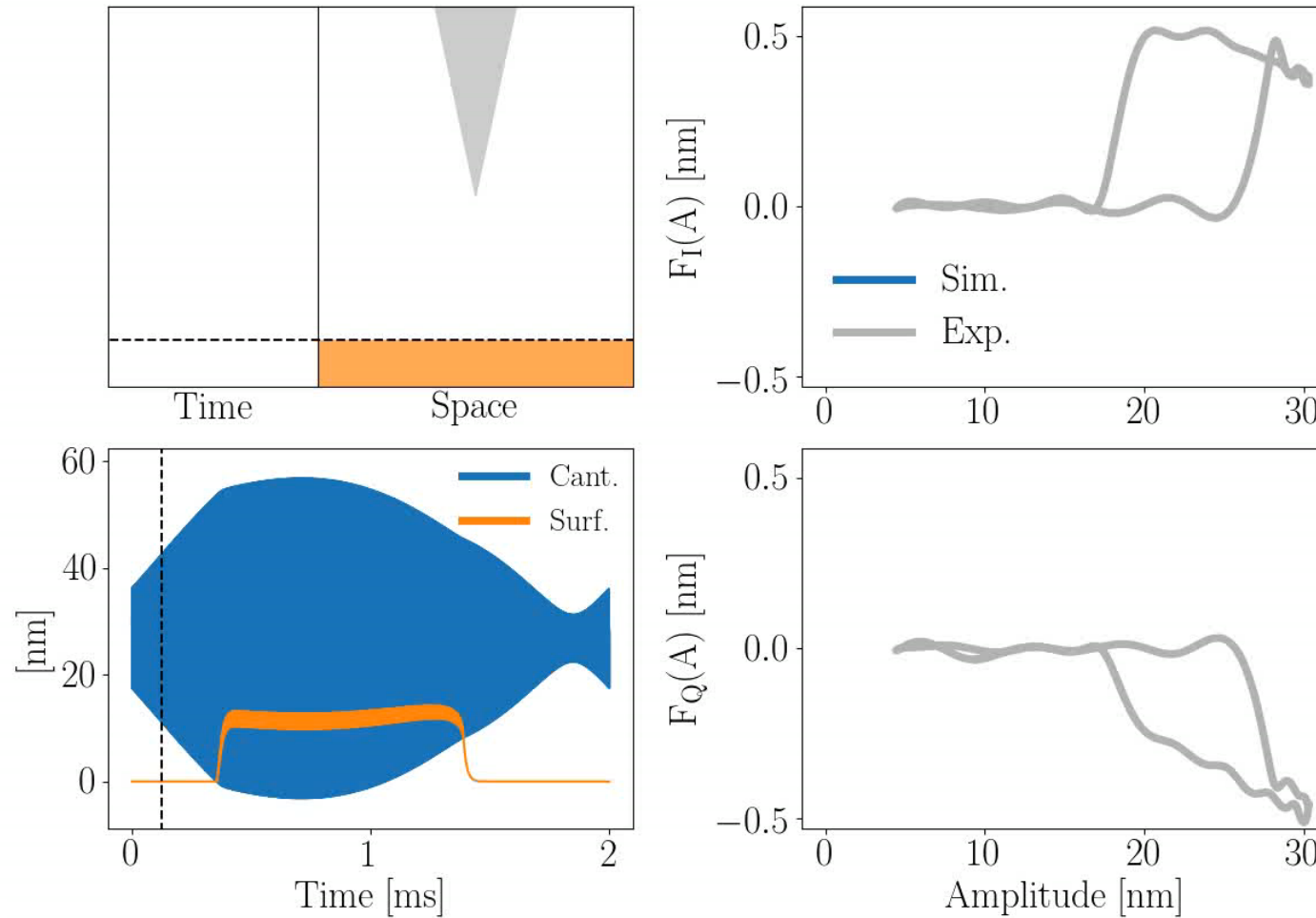
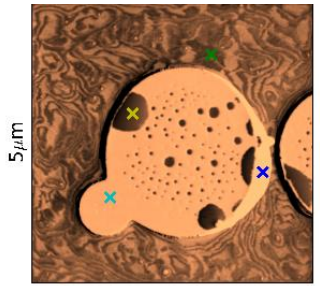
PS phase



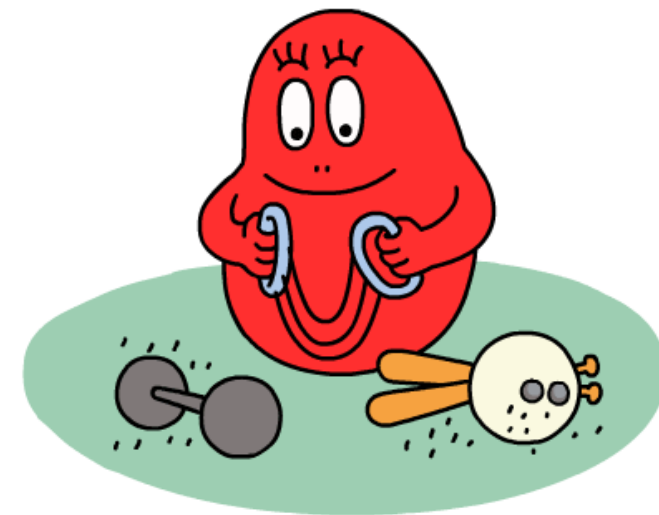
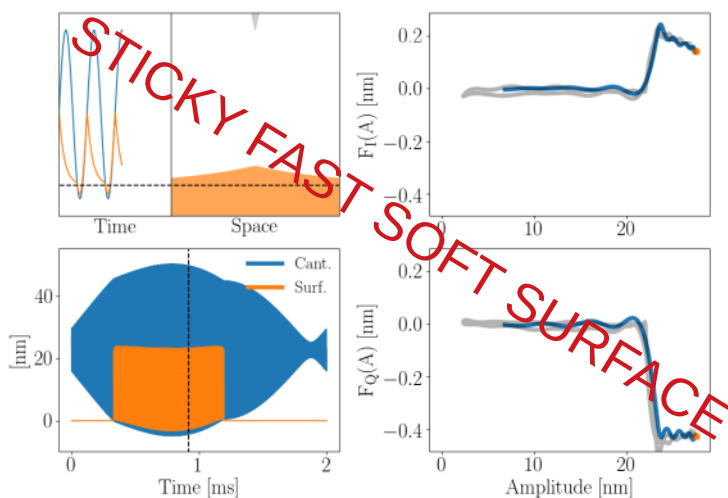
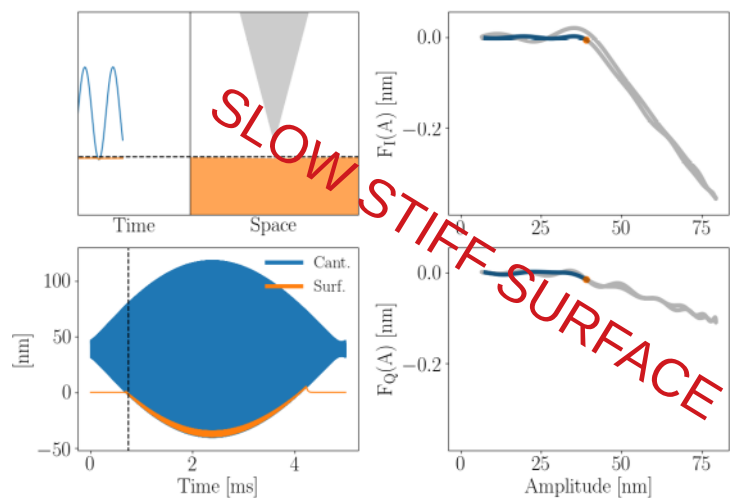
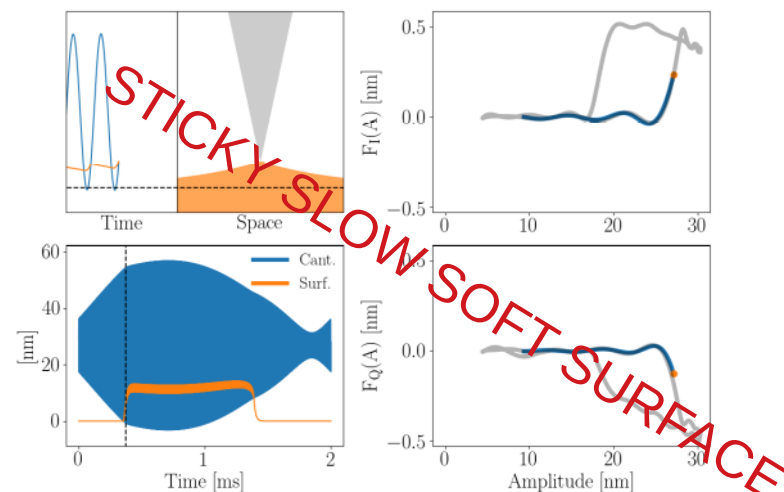
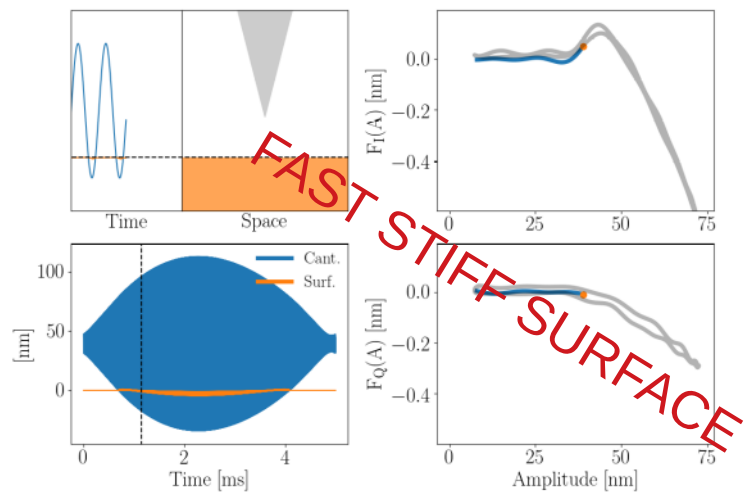
Courtesy of P.A. Thoren

Fit to moving surface model

PCL amorphous phase

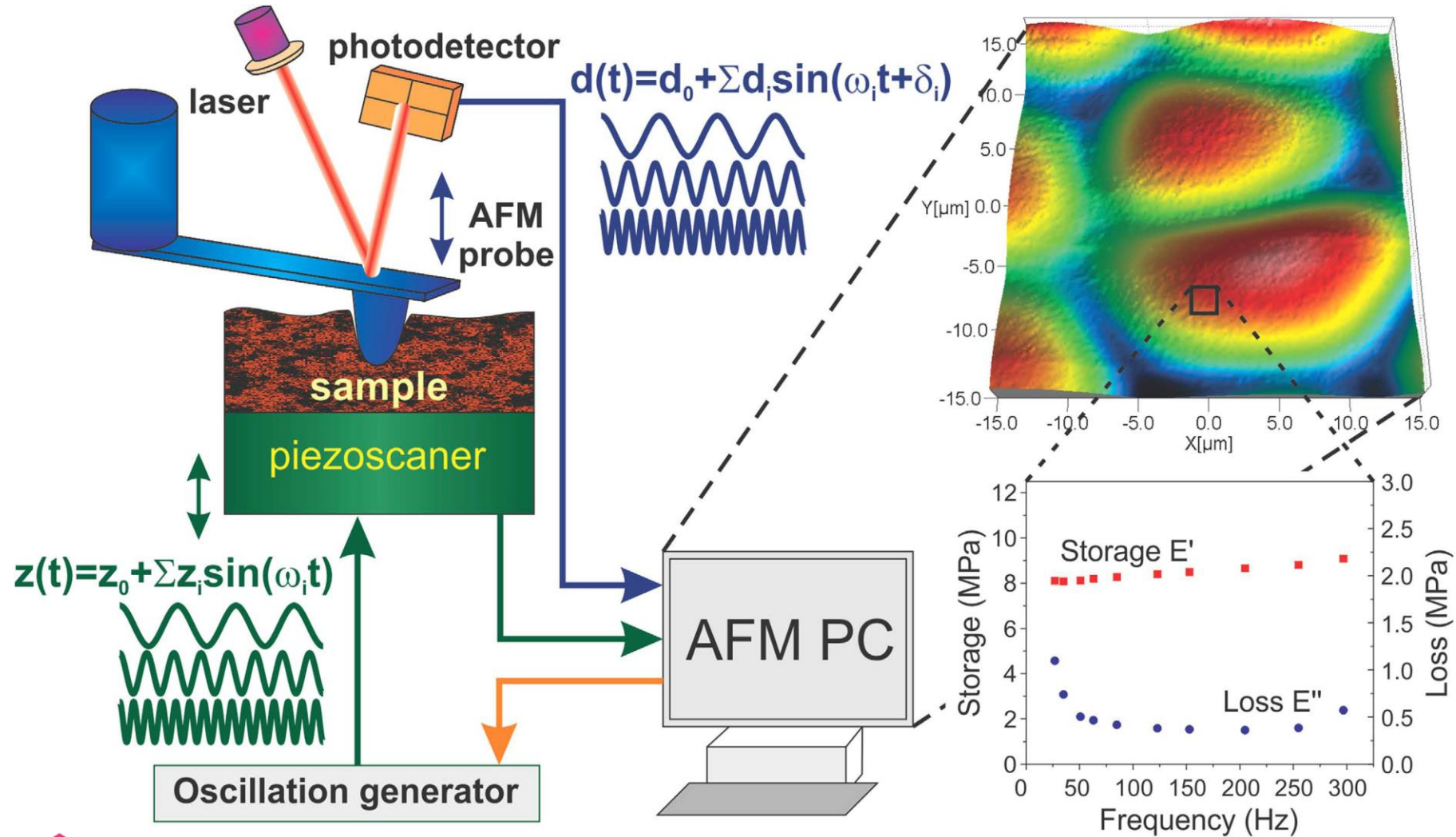


Courtesy of P.A. Thoren



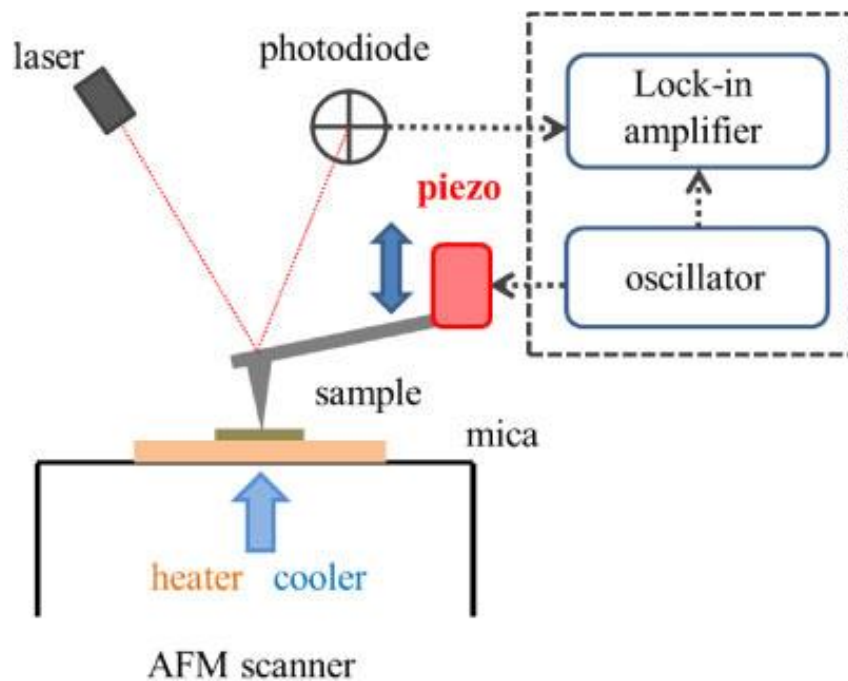
Courtesy of P.A. Thoren

FT-NanoDMA



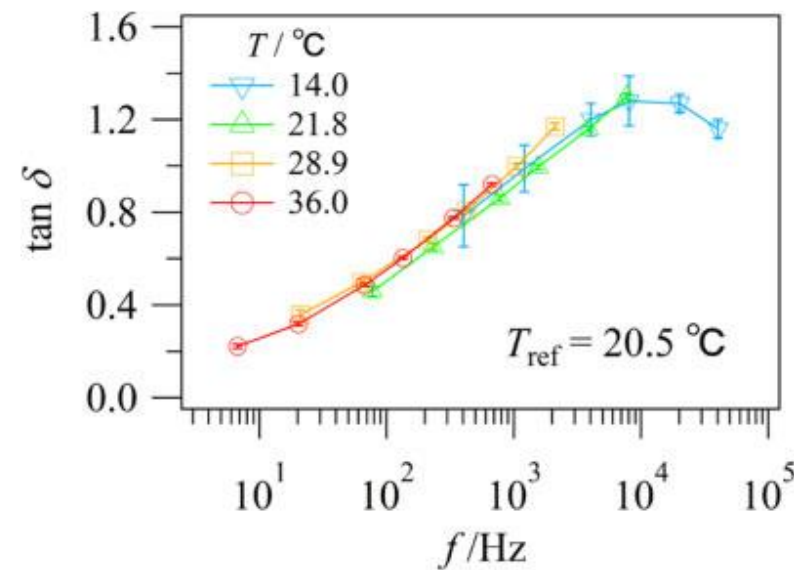
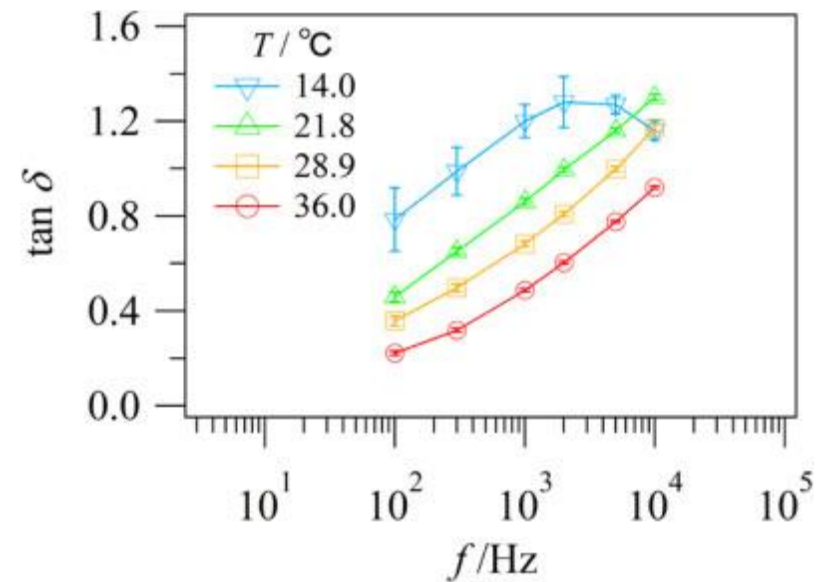
Nanorheological AFM

Vulcanized SBR



Williams–Landel–Ferry (WLF) equation

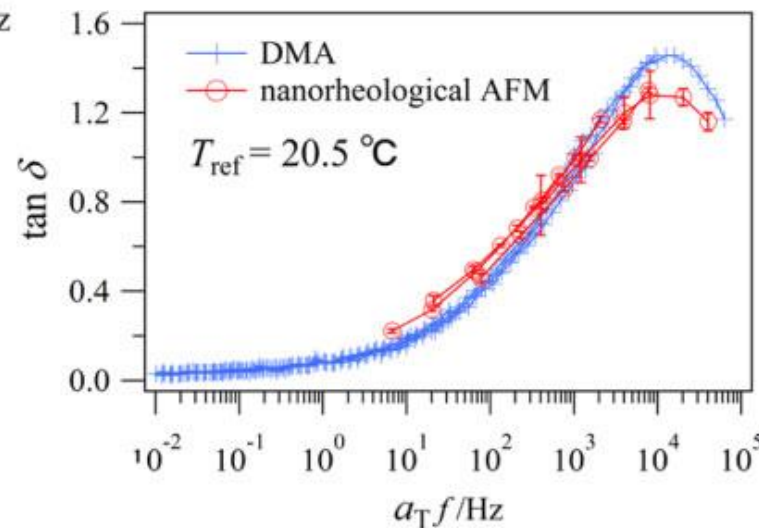
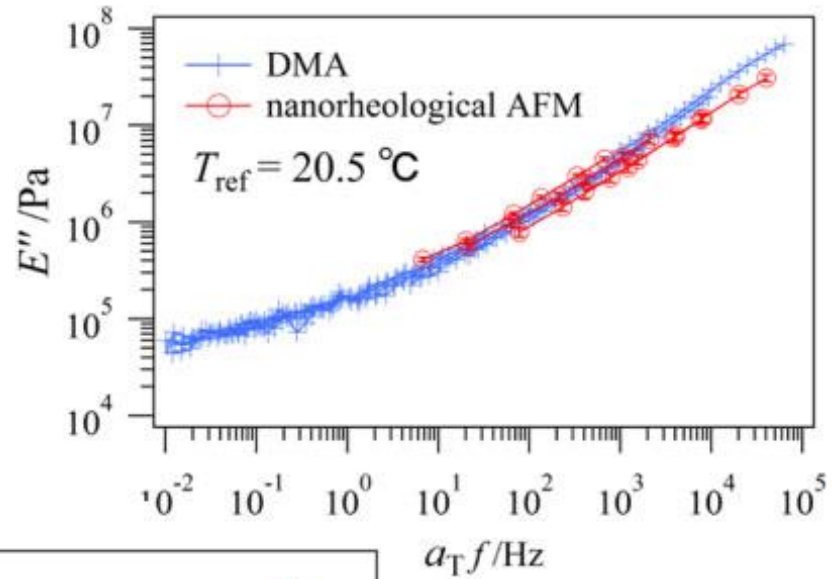
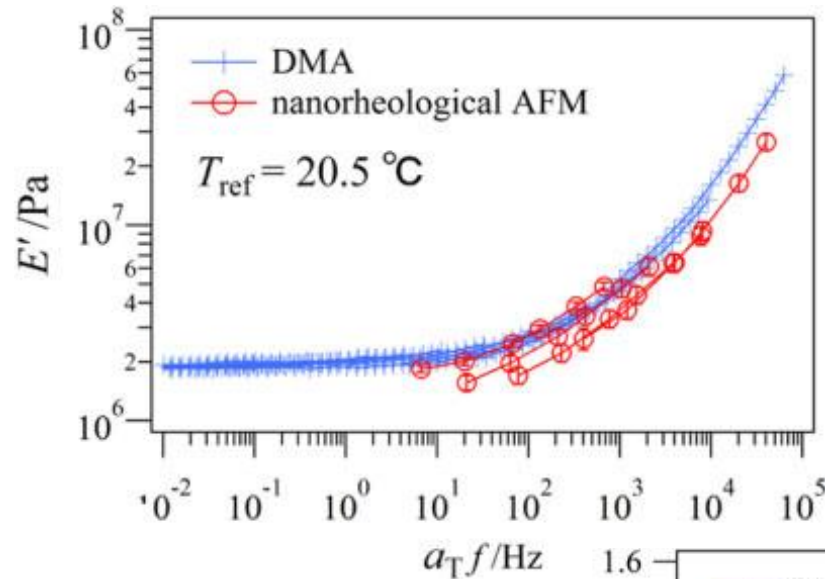
Japanese Journal of Applied Physics 57, 08NB08 (2018)



Nanorheological AFM

Vulcanized SBR

Japanese Journal of Applied Physics 57, 08NB08 (2018)



E' = Storage Modulus
 E'' = Loss Modulus
 $\tan \delta = E'/E''$

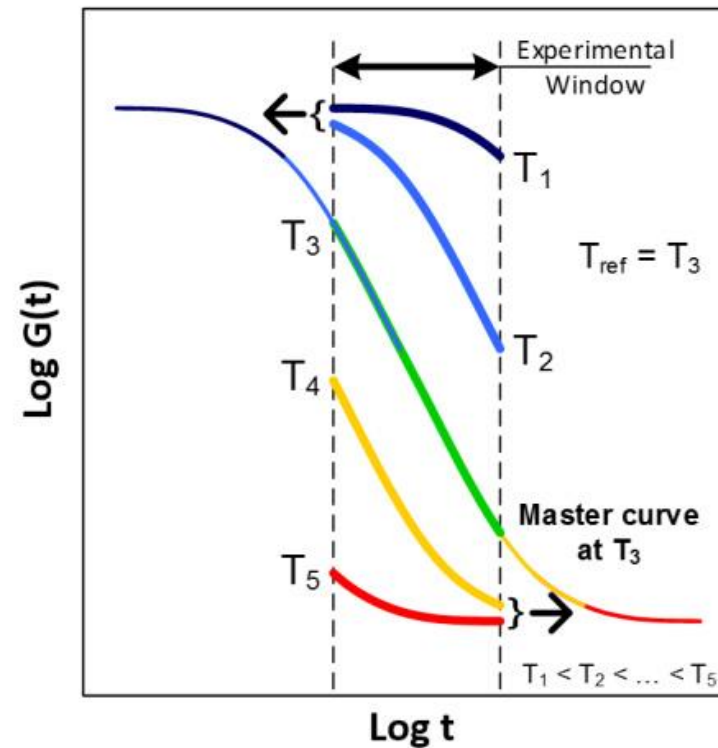
Take home message ...

Combined, AFM measurements with non-resonant modes and resonant modes can provide

FV based Contact Resonance for stiff samples at higher frequencies.

FV force nDMA, and FI/FQ curves (ImAFM) for soft samples at low frequencies.

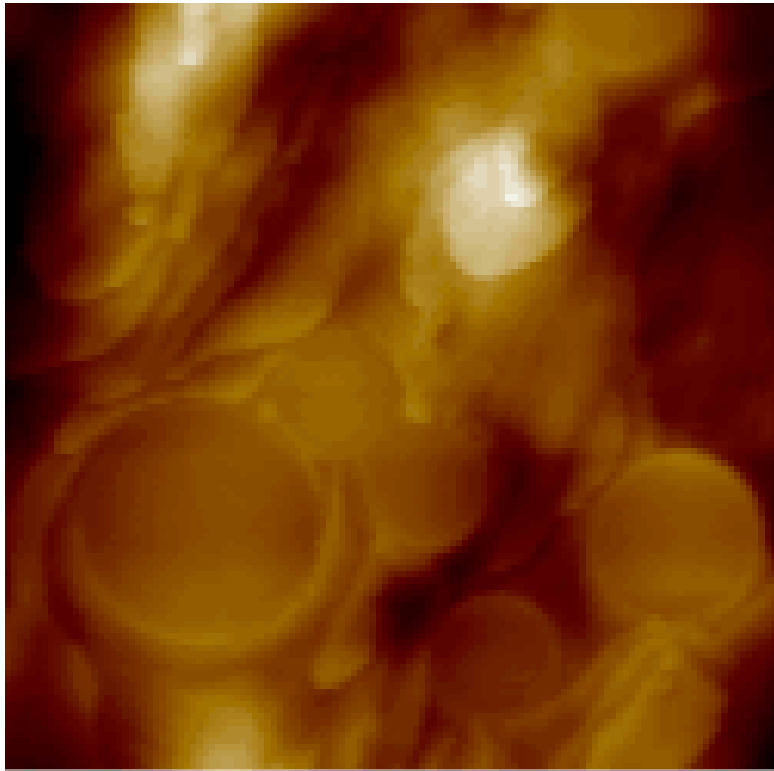
FV and PFT cover wide range of ramp rates for time-temperature studies.



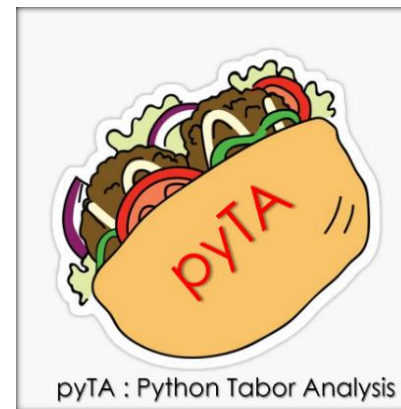
Conclusions

Multifrequency AFM methods are extremely promising but also need some (new) models to provide **quantitative parameters**.

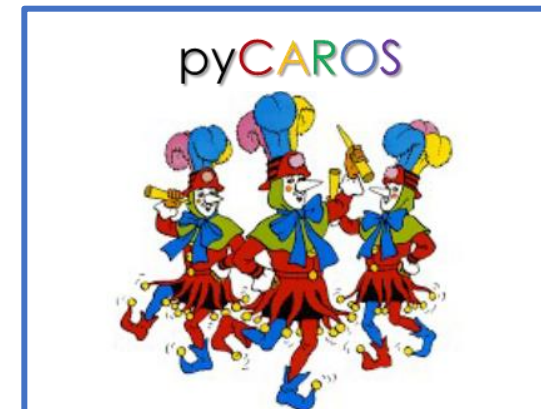
Data-driven materials development and design (machine learning, AI) are most probably the key issue to achieve this goal.



For instance, recording of stiffness, deformation, adhesion and viscoelastic (E' , E'' , $\tan \delta$) property maps in parallel to the topography image is now possible with **quantitative values** and using an appropriate **data clustering approach** and adapted **mechanical model(s)** and **preprocessing** of the force curves (deep learning)..



pyTA : Python Tabor Analysis



Questions ?

